

Learning and understanding thermodynamics: a struggle against obviousness

Dmitri V. Malakhov

Department of Materials Science

and Engineering

Outline

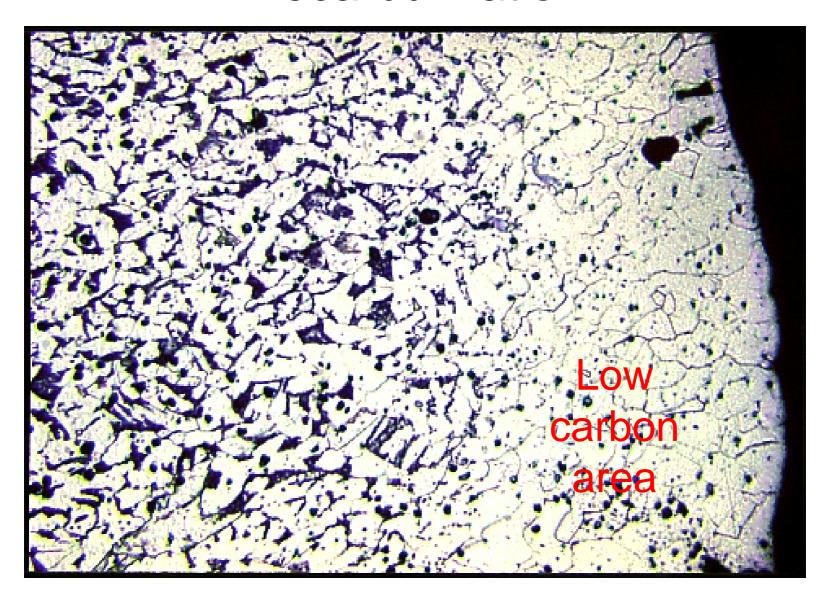
- Decarburization: let us warm up
- Maximal temperature of adiabatic combustion: do we calculate it correctly?
- Influence of pressure on the molar Gibbs energies: what does P do to G(x)?
- A choice of a reference frame in the compound energy formalism: is it unique?
- An advice: stay alert, be critical
- MSE: we are waiting for you

Combustion: a source of heat



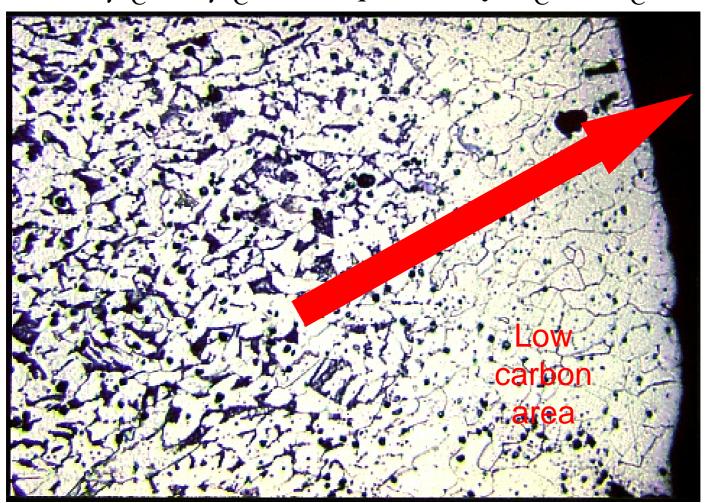
Component	Wt. %
С	0.4
Mn	0.7
Si	1.5
Cr	0.7
V	0.2

Decarburization

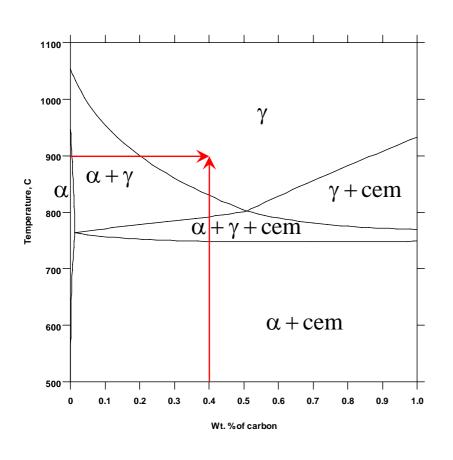


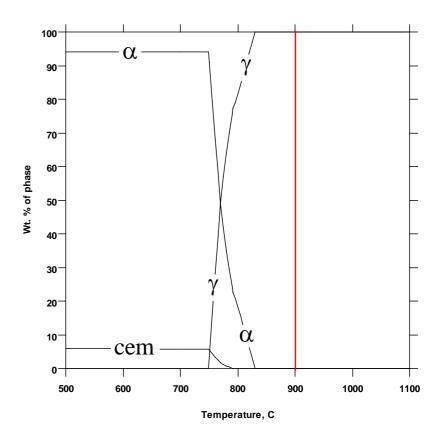
What causes the misfortune?

 $\mu_{\rm C}^{\rm gas} < \mu_{\rm C}^{\rm steel}$ or, equivalently, $a_{\rm C}^{\rm gas} < a_{\rm C}^{\rm steel}$



Austenitization: interior and near-surface region behave differently



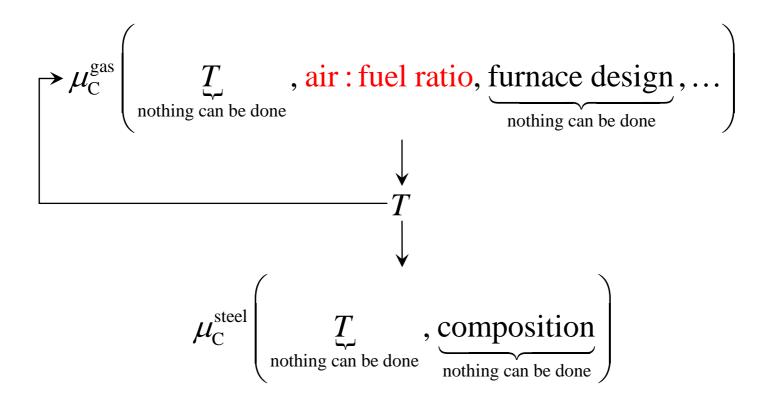


How can decarburization be suppressed or eliminated?

- Kinetics (induction heating: seconds instead of minutes)
- Chemistry (heating in vacuum: an absence of mediators such as H₂O, CO₂, H₂)
- Thermodynamics

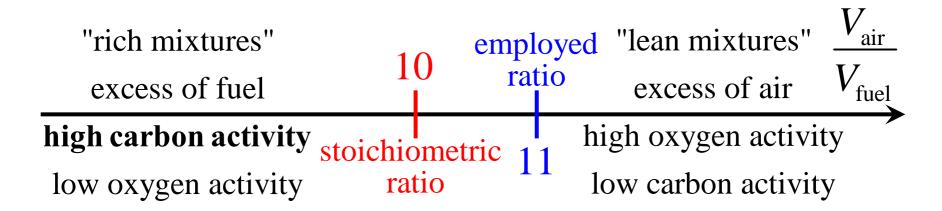
Can the adversity be avoided?

$$\mu_{\mathrm{C}}^{\mathrm{gas}} < \mu_{\mathrm{C}}^{\mathrm{steel}}$$

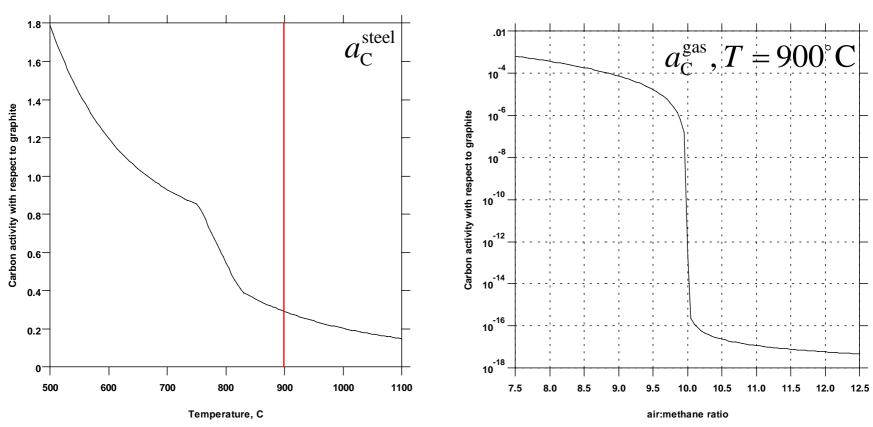


Terminology (fuel-dependent)

$$\underbrace{\text{CH}_{4}}_{\text{fuel}} + \underbrace{2\text{O}_{2} + 8\text{N}_{2}}_{\text{air}} \rightarrow \underbrace{\text{CO}_{2} + 2\text{H}_{2}\text{O} + 8\text{N}_{2}}_{\text{flue gas}}$$

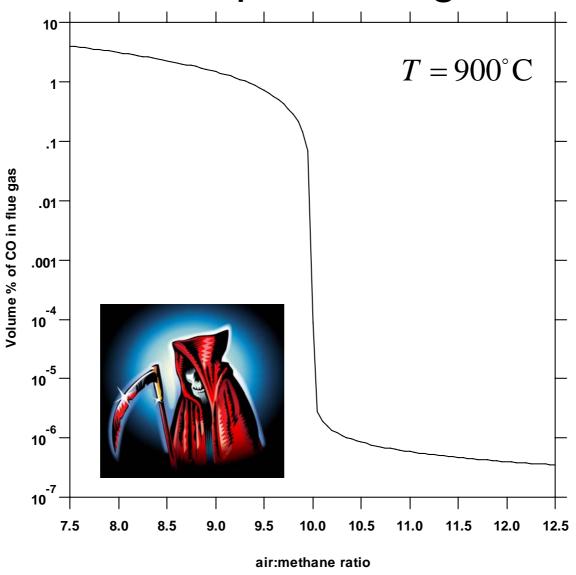


Can decarburization be defeated by changing the air:fuel ratio?



Thermodynamic verdict: no way!

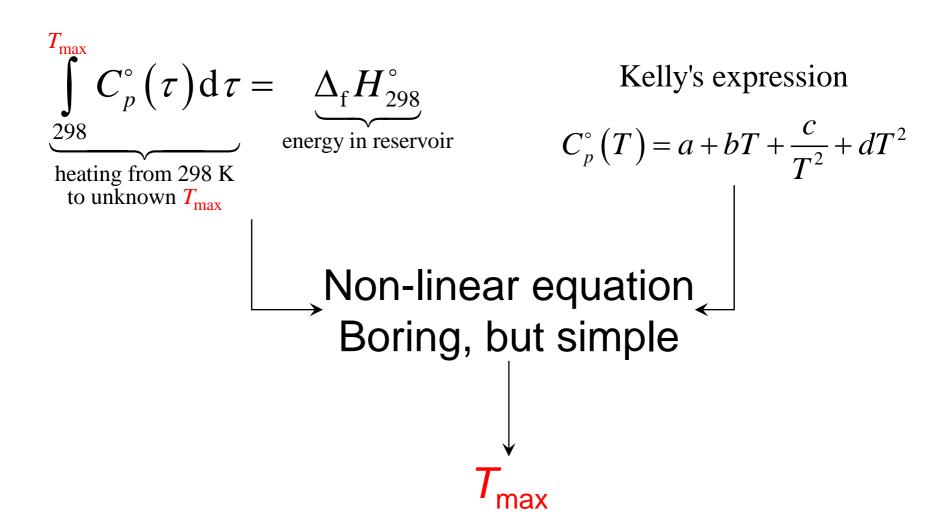
Another reason prohibiting low ratios



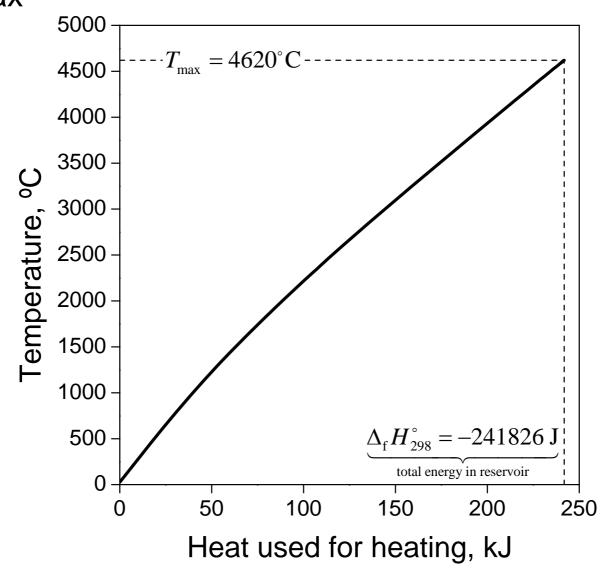
Temperature of adiabatic reaction between hydrogen and oxygen

- 1. Take 0.5 mole of $O_2(g)$ and 1 moles of $H_2(g)$ at 25°C and P = 1 atm. By definition, $H^\circ = 0$.
- 2. Make 1 mole of $H_2O(g)$ at 25°C and put $\Delta_f H_{298}^\circ$ released into a heat reservoir.
- 3. Maintain P = 1 atm and use all energy stored in the reservoir for heating 1 mole of gaseous H_2O .
- 4. Ask yourself a question: do I know how to calculate T_{max} ?

Of course you know, but just in case...



$T_{\text{max}} = 4620$ °C: an erroneous result



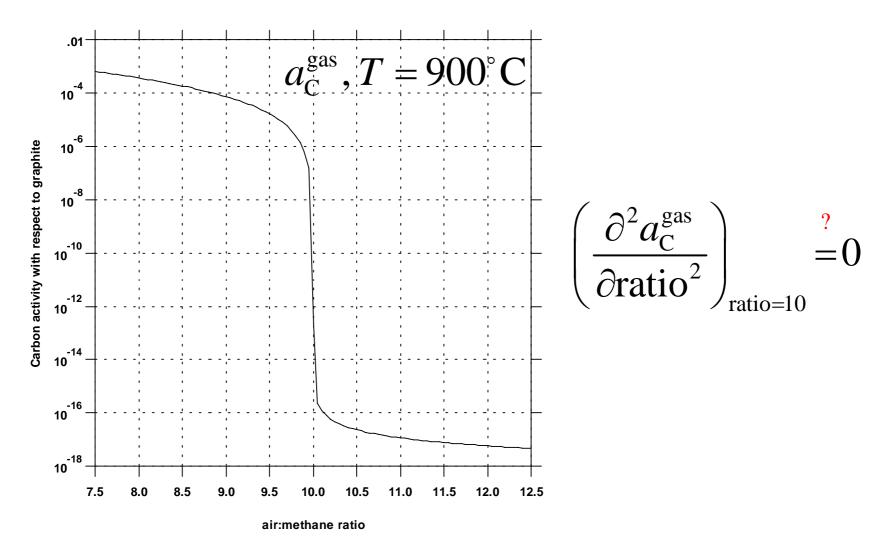
What happens at this temperature?

```
Output from POLY-3, equilibrium = 1, label A0 , database: SSUB3
Conditions:
N(H2)=1, N(O2)=0.5, T=4893.29, P=1E5
DEGREES OF FREEDOM 0
Temperature 4893.29 K ( 4620.14 C), Pressure 1.000000E+05
Number of moles of components 1.50000E+00, Mass in grams 1.80148E+01
Total Gibbs energy -1.87698E+06, Enthalpy 9.34464E+05, Volume 1.18691E+00
                       Moles M-Fraction Activity Potential Ref.stat
Component
                       1.0000E+00 6.6667E-01 1.7999E-12 -1.1003E+06 SER
H2
02
                       5.0000E-01 3.3333E-01 2.6171E-17 -1.5534E+06 SER
GAS
                          Status ENTERED
                                           Driving force 0.0000E+00
Moles 1.5000E+00, Mass 1.8015E+01, Volume fraction 1.0000E+00 Mole fractions:
H2 6.66667E-01 O2 3.33333E-01
Constitution:
H 6.47172E-01 H101 1.20372E-02 H102 8.06924E-07
O 3.24793E-01 O2 2.81954E-03 O3 5.47919E-09
H2 1.28658E-02 H201 3.11755E-04 H202 1.44469E-09
```

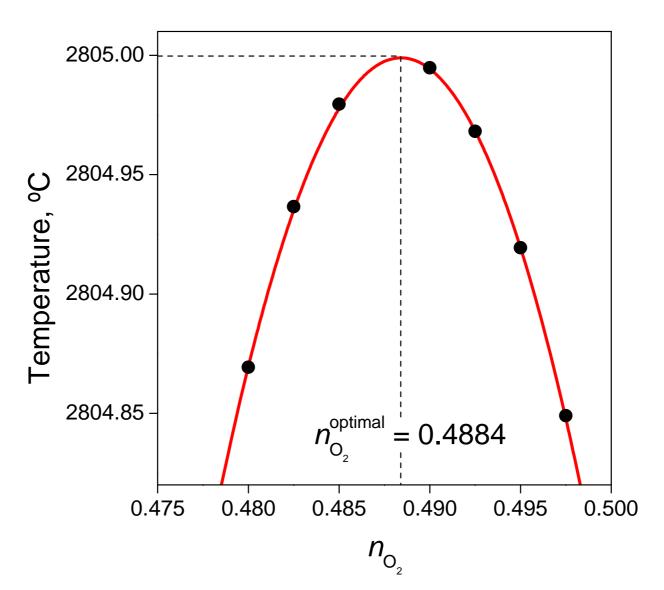
A dramatic difference

```
Output from POLY-3, equilibrium = 1, label A0 , database: SSUB3
Conditions:
P=1E5, N(H2)=1, N(O2)=0.5, H=0
DEGREES OF FREEDOM 0
Temperature 3077.91 K ( 2804.76 C), Pressure 1.000000E+05
Number of moles of components 1.50000E+00, Mass in grams 1.80148E+01
Total Gibbs energy -1.01149E+06, Enthalpy 4.40821E-10, Volume 3.10200E-01
Component
                          Moles M-Fraction Activity Potential Ref.stat
H2
                          1.0000E+00 6.6667E-01 1.2037E-10 -5.8452E+05 SER
02
                          5.0000E-01 3.3333E-01 3.2235E-15 -8.5394E+05 SER
GAS
                             Status ENTERED
                                                 Driving force 0.0000E+00
Moles 1.5000E+00, Mass 1.8015E+01, Volume fraction 1.0000E+00 Mole fractions:
H2 6.66667E-01 O2 3.33333E-01
Constitution:
H2O1 5.85041E-01 H 7.70546E-02 H1O2 4.50911E-05
H2 1.48661E-01 02 5.06614E-02 H2O2 2.49697E-06
H101 1.05477E-01 O
                          3.30571E-02 O3 1.90507E-08
                    T_{\text{max}} = 4620^{\circ}\text{C} \rightarrow T_{\text{max}} = 2805^{\circ}\text{C}
only H<sub>2</sub>O is considered all species are considered
```

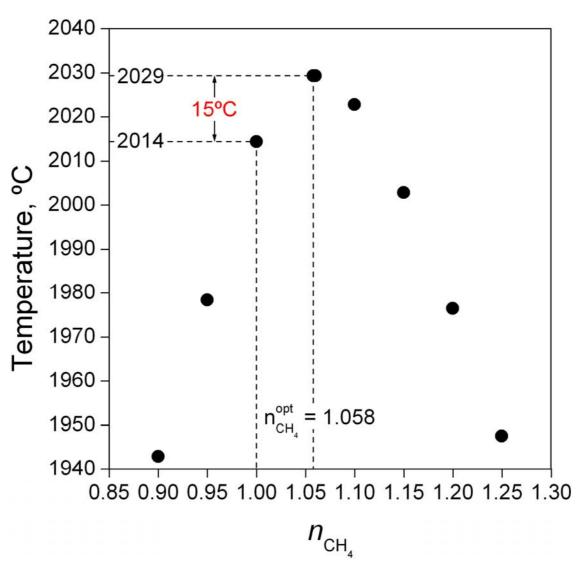
A less obvious fault in our calculations



Who said that 2:1 was the best ratio?



$2O_2 + 8N_2 + nCH_4$



Fuel & oxidizer? Yes, but ratio as well!



I taught many thermodynamics-related courses



Shewmon "Transformations in metals"

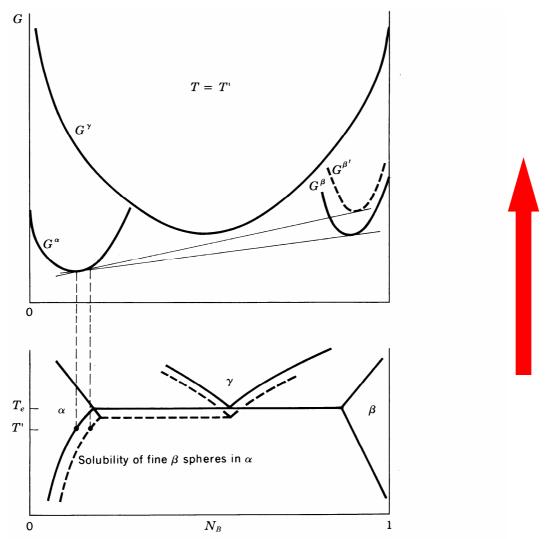


figure 4-10. Free-energy diagram and phase diagram indicating change in solubility of β , and eutectoid temperature when β is present as fine spheres (labeled β').

Porter, Easterling "Phase transformations in metals and alloys"

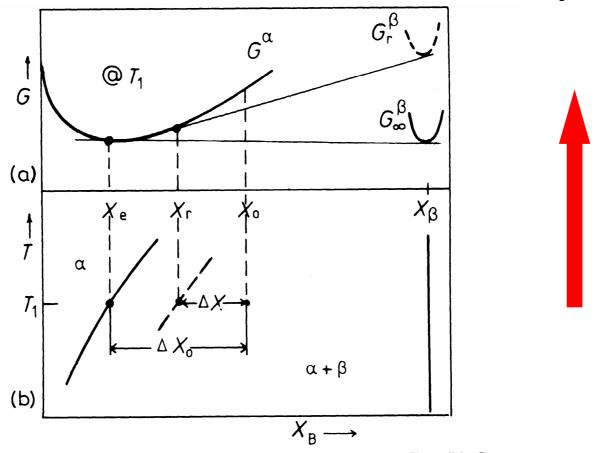


Fig. 5.20 The Gibbs-Thomson effect. (a) Free energy curves at T_1 . (b) Corresponding phase diagram.

Hillert "Applications of Gibbs energy-composition diagrams"

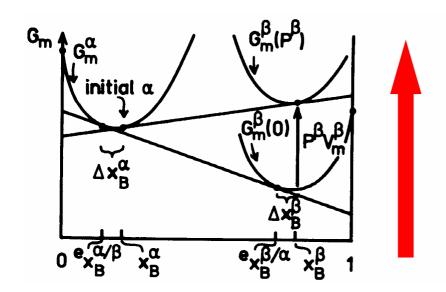


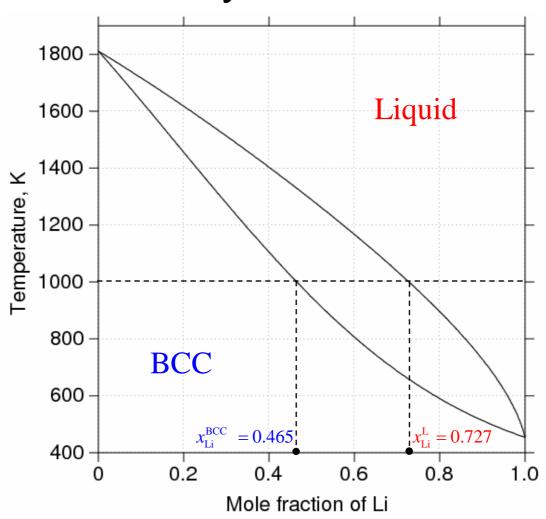
Fig. 27. Change in compositions when a pressure is applied to one of the phases in a two-phase equilibrium.

Rationale

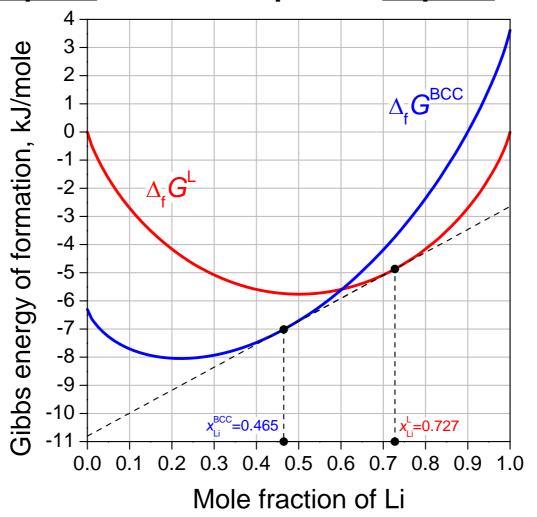
$$\left(\frac{\partial G}{\partial P}\right)_T = \bigvee_{\text{always positive}}$$

If P increases, then G moves upward

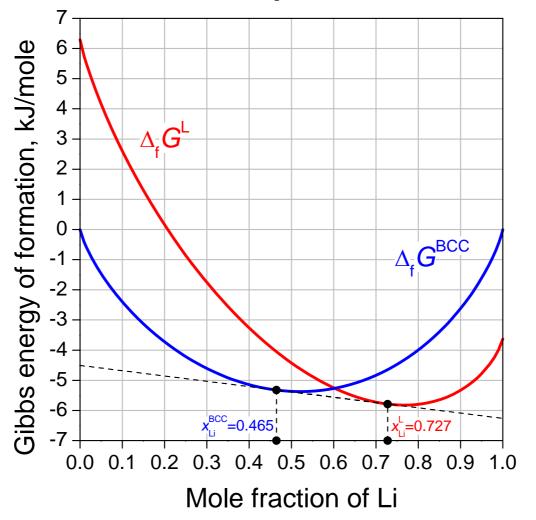
Liquid/BCC equilibrium in the Fe–Li system



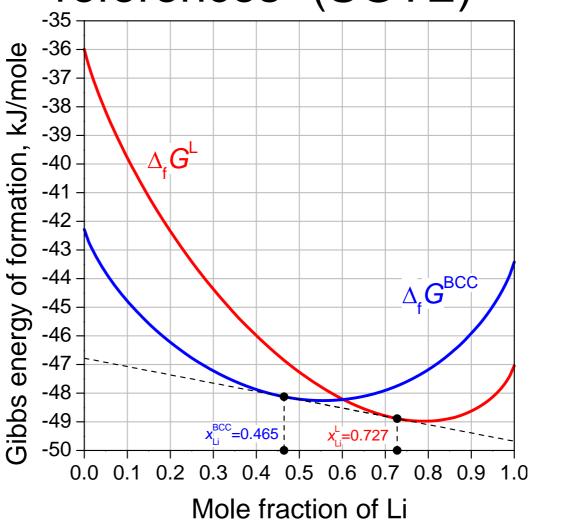
T = 1000 K, reference states are pure liquid Fe and pure liquid Li



T = 1000 K, reference states are pure BCC Fe and pure BCC Li



T = 1000 K, "standard element references" (SGTE)

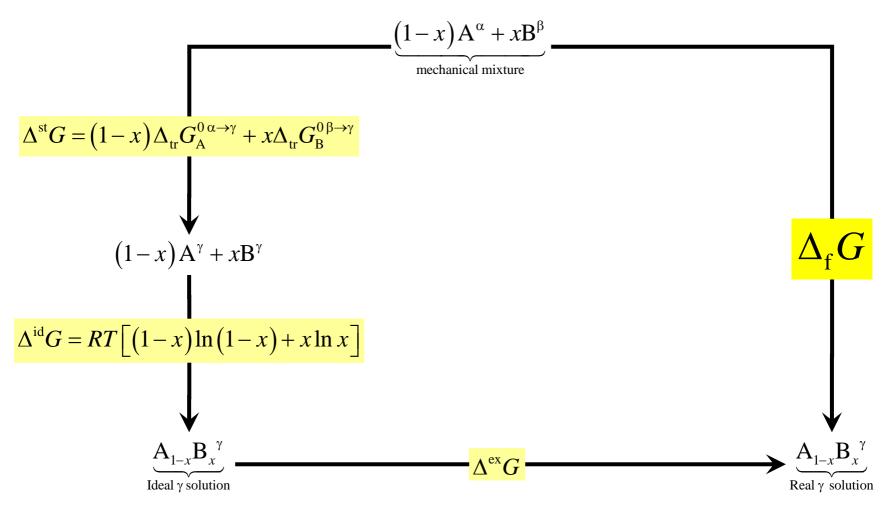


Not G per se, but $\Delta_f G!$

$$\left(\frac{\partial G}{\partial P}\right)_T = V_{\text{always positive}}$$

$$\left(\frac{\partial \Delta_{\mathbf{f}} G}{\partial P}\right)_{T} = \Delta_{\mathbf{f}} V$$

Let us make a phase γ from pure components



Reference states are pure liquid components

$$\Delta G^{L} = (1 - x) \underbrace{\Delta_{tr} G_{A}^{0 L \to L}}_{\equiv 0} + x \underbrace{\Delta_{tr} G_{B}^{0 L \to L}}_{\equiv 0} + \underbrace{\Delta^{id} G^{L}}_{\neq f(P)} + \underbrace{\Delta^{ex} G^{L}}_{\neq f(P)}$$

$$\Delta G^{\alpha} = (1 - x) \Delta_{tr} G_{A}^{0 L \to \alpha} + x \Delta_{tr} G_{B}^{0 L \to \alpha} + \underbrace{\Delta^{id} G^{\alpha}}_{\neq f(P)} + \underbrace{\Delta^{ex} G^{\alpha}}_{\neq f(P)}$$

$$\left(\frac{\partial \Delta G^{\mathrm{L}}}{\partial P}\right)_{T} = \mathbf{0}$$

$$\left(\frac{\partial \Delta G^{\alpha}}{\partial P}\right)_{T} = (1-x) \left(\frac{\partial \Delta_{\text{tr}} G_{A}^{0 \text{ L} \to \alpha}}{\partial P}\right)_{T} + x \left(\frac{\partial \Delta_{\text{tr}} G_{B}^{0 \text{ L} \to \alpha}}{\partial P}\right)_{T} \\
= (1-x) \left(V_{A}^{\alpha} - V_{A}^{L}\right) + x \left(V_{B}^{\alpha} - V_{B}^{L}\right) < 0$$
usually negative

$$V_{\rm m} = \frac{A}{\rho}$$

Why "usually"? Because there are rare exceptions such as H₂O, Bi, Sb, cast iron

What does this mean?!



 ΔG^{L} does not change its position

$$\Delta G^{\alpha}$$
 shifts downward by $\left[(1-x)(V_{A}^{L}-V_{A}^{\alpha}) + x(V_{B}^{L}-V_{B}^{\alpha}) \right] \times P$

The solid phase α is stabilized by pressure applied

Now reference states are pure solid components

$$\Delta G^{L} = (1 - x) \Delta_{tr} G_{A}^{0 \alpha \to L} + x \Delta_{tr} G_{B}^{0 \alpha \to L} + \underbrace{\Delta^{id} G^{L}}_{\neq f(P)} + \underbrace{\Delta^{ex} G^{L}}_{\neq f(P)}$$

$$\Delta G^{\alpha} = (1 - x) \underbrace{\Delta_{\text{tr}} G_{A}^{0 \alpha \to \alpha}}_{\equiv 0} + x \underbrace{\Delta_{\text{tr}} G_{B}^{0 \alpha \to \alpha}}_{\equiv 0} + \underbrace{\Delta^{\text{id}} G^{\alpha}}_{\neq f(P)} + \underbrace{\Delta^{\text{ex}} G^{\alpha}}_{\neq f(P)}$$

$$\left(\frac{\partial \Delta G^{\alpha}}{\partial P}\right)_{T} = 0$$

$$\left(\frac{\partial \Delta G^{L}}{\partial P}\right)_{T} = (1-x)\left(\frac{\partial \Delta_{tr}G_{A}^{0\,\alpha \to L}}{\partial P}\right)_{T} + x\left(\frac{\partial \Delta_{tr}G_{B}^{0\,\alpha \to L}}{\partial P}\right)_{T}$$

$$= (1-x)\left(\frac{V_{A}^{L} - V_{A}^{\alpha}}{\partial P}\right) + x\left(\frac{V_{B}^{L} - V_{B}^{\alpha}}{\partial P}\right) > 0$$
usually positive

$$V_{\rm m} = \frac{A}{\rho}$$

What's going on?!



 ΔG^{α} does not change its position

$$\Delta G^{L}$$
 shifts upward by $\left[(1-x)(V_{A}^{L}-V_{A}^{\alpha}) + x(V_{B}^{L}-V_{B}^{\alpha}) \right] \times P$

The liquid phase is destabilized by pressure applied

Reference states are pure liquid components

 $\Delta G^{\rm L}$ does not change its position,

 ΔG^{α} shifts downward by

$$\underbrace{\left[(1-x) \left(V_{\rm A}^{\rm L} - V_{\rm A}^{\alpha} \right) + x \left(V_{\rm B}^{\rm L} - V_{\rm B}^{\alpha} \right) \right] \times P}_{\text{our result}}$$

 ΔG^{α} does not change its position,

 ΔG^{L} shifts upward by

$$\left[(1-x) \left(V_{\mathbf{A}}^{\mathbf{L}} - V_{\mathbf{A}}^{\alpha} \right) + x \left(V_{\mathbf{B}}^{\mathbf{L}} - V_{\mathbf{B}}^{\alpha} \right) \right] \times P$$

In terms of relative positions, these 3 situations are identical

$$\Delta G^{\alpha}$$
 shifts upward by $\left[(1-x)V_{A}^{\alpha} + xV_{B}^{\alpha} \right] \times P$

$$\Delta G^{L}$$
 shifts upward by $\left[(1-x)V_{A}^{L} + xV_{B}^{L} \right] \times P$

Reference states are pure solid components

 ΔG^{α} does not change its position,

 ΔG^{L} shifts upward by

$$\underbrace{\left[(1-x) \left(V_{\mathbf{A}}^{\mathbf{L}} - V_{\mathbf{A}}^{\alpha} \right) + x \left(V_{\mathbf{B}}^{\mathbf{L}} - V_{\mathbf{B}}^{\alpha} \right) \right] \times P}_{\text{our result}}$$

 ΔG^{L} does not change its position,

 ΔG^{α} shifts downward by

$$\left[(1-x) \left(V_{\mathbf{A}}^{\mathbf{L}} - V_{\mathbf{A}}^{\alpha} \right) + x \left(V_{\mathbf{B}}^{\mathbf{L}} - V_{\mathbf{B}}^{\alpha} \right) \right] \times P$$

In terms of relative positions, these 3 situations are identical

$$\Delta G^{\alpha}$$
 shifts upward by $\left[(1-x)V_{A}^{\alpha} + xV_{B}^{\alpha} \right] \times P$

$$\Delta G^{L}$$
 shifts upward by $\left[(1-x)V_{A}^{L} + xV_{B}^{L} \right] \times P$

Shewmon "Transformations in metals"

4-6 Consider a system in which four phases exist with a $G(N_{\rm B})$ diagram as shown in Fig. 4-13. Show the phase diagram that results if the free energy of the α , β , and γ phases decreases relative to that of the liquid as the temperature is decreased. Do this by first showing the $G(N_{\rm B})$ diagram for several lower temperatures.

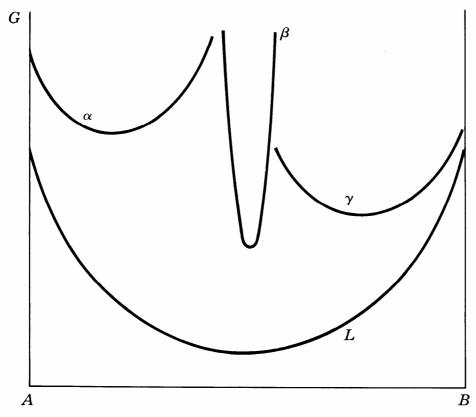
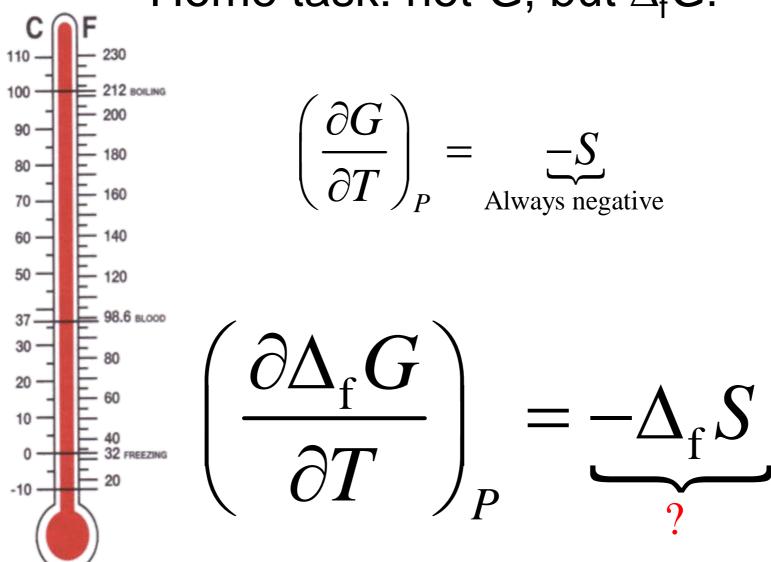
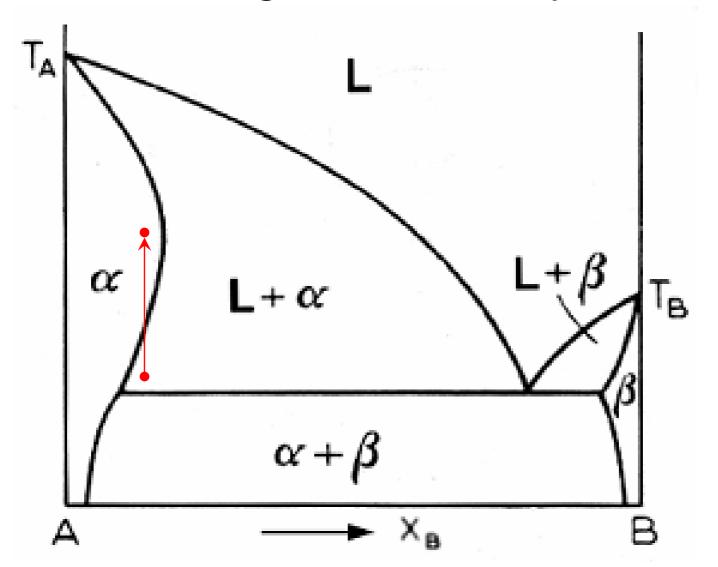


figure 4-13

Home task: not G, but $\Delta_f G$!

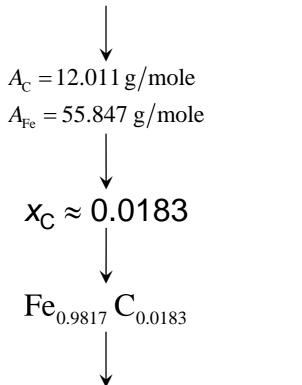


Retrograde solubility



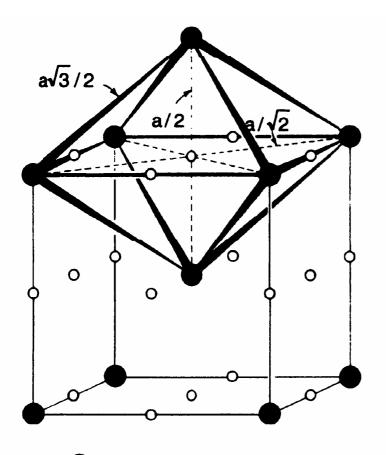
Making an excusable mistake

Steel contains 0.4 wt.% of C

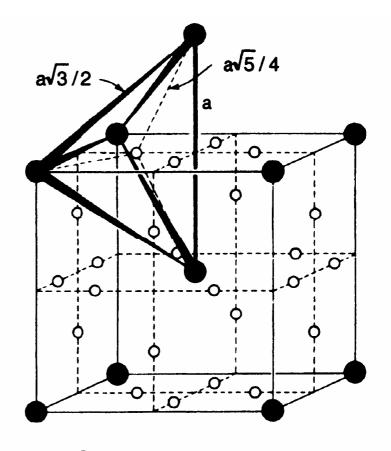


$$S^{\text{conf}} = -R(0.9817 \ln 0.9817 + 0.0183 \ln 0.0183) \approx 0.761 \frac{J}{\text{K} \times \text{mole}}$$
a wrong result

α -Fe (ferrite) and δ -Fe

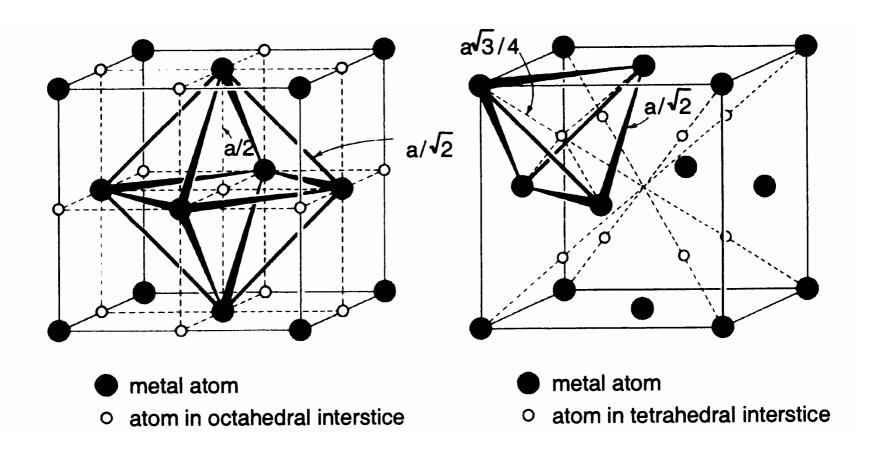


- metal-Atom
- o atom in octahedral interstice



- metal-Atom
- o atom in tetrahedral interstice

γ -Fe (austenite)

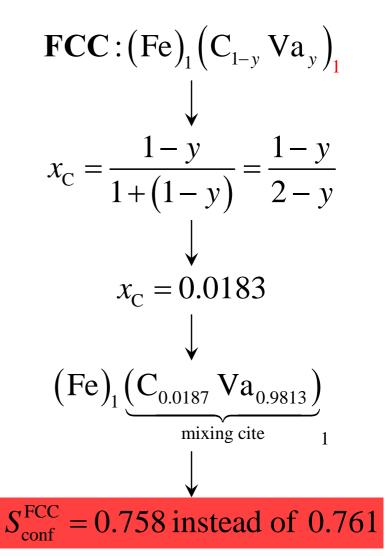


Correcting the mistake

BCC:
$$(Fe)_1 (C_{1-y} Va_y)_3$$

$$\downarrow x_C = \frac{3(1-y)}{1+3(1-y)} = \frac{3-3y}{4-3y}$$

$$\downarrow x_C = 0.0183$$



Sublattice model (CEF)

$$(A,B)_a(C,D)_c$$
, $\underbrace{a+c=1}_{\text{beneficial}}$ For the sake of simplicity $(A,B)(C,D)$

$$(A_{1-y}, B_y)(C_{1-z}, D_z)$$

$$\underbrace{\left(Na_{1-y}^{+}, K_{y}^{+}\right)}_{\text{cation site}}\underbrace{\left(Cl_{1-z}^{-}, Br_{z}^{-}\right)}_{\text{anion cite}}$$

$$\underbrace{\left(\mathbf{Sr}_{a}^{2+},\mathbf{Ba}_{b}^{2+},\mathbf{La}_{1-a-b-c}^{3+},\mathbf{Va}_{c}^{0}\right)}_{\text{A-cite}}\underbrace{\left(\mathbf{Ti}_{1-y}^{4+},\mathbf{Va}_{y}^{0}\right)}_{\text{B-cite}}\mathbf{O}_{3}^{2-}$$

$$(1-y)A + yB + (1-z)C + zD \rightarrow (A_{1-y}, B_y)(C_{1-z}, D_z)$$

Another way to synthesize the phase

$$(1-y)A + yB + (1-z)C + zD$$

$$\rightarrow \alpha AC + \beta AD + \gamma BC + \delta BD$$

$$(A_{1-y}, B_y)(C_{1-z}, D_z)$$

Playing field

$$(A,B)_a(C,D)_c$$

Compounds a.k.a. end-members

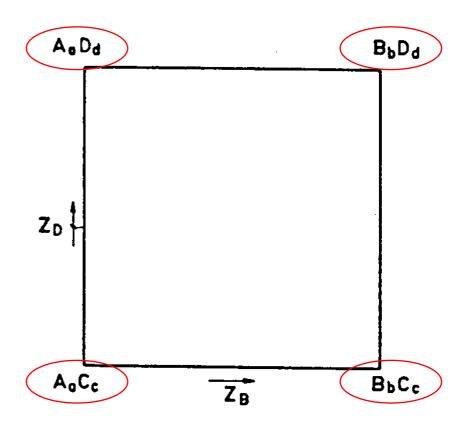


Fig. 1. Representation of composition in a quaternary system where the components mix with each other, two and two.

Hillert's suggestion was based on a powerful KISS principle

$$\alpha AC + \beta AD + \gamma BC + \delta BD \rightarrow (A_{1-y}, B_y)(C_{1-z}, D_z)$$

$$(1-y)(1-z)AC + (1-y)zAD$$

$$+y(1-z)BC + yzBD \rightarrow (A_{1-y}, B_y)(C_{1-z}, D_z)$$

$$\mathbf{A} \quad (1-y)(1-z) + (1-y)z$$

$$= 1 + \sqrt{z} + \sqrt{z} + \sqrt{z} = 1-y$$

Reference surface, not reference line or plane or hyperplane

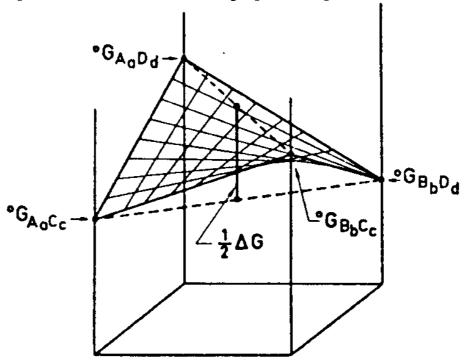


Fig. 2. Suggested surface of reference for the free energy in a quaternary system where the components mix with each other two and two.

What's about uniqueness?

$$\alpha AC + \beta AD + \gamma BC + \delta BD \rightarrow (A_{1-y}, B_y)(C_{1-z}, D_z)$$

$$\begin{cases} \mathbf{A} & \alpha + \beta = 1 - y \\ \mathbf{B} & \gamma + \delta = y \\ \mathbf{C} & \alpha + \gamma = 1 - z \\ \mathbf{D} & \beta + \delta = z \end{cases}$$

$$\begin{cases} \mathbf{A} & \alpha + \beta = 1 - y \\ 2 & 0 & 0 & 1 & 1 \\ 1 & 0 & 1 & 0 \\ 4 & 0 & 1 & 0 & 1 \end{pmatrix}$$

$$\begin{cases} \mathbf{C} & \alpha + \gamma = 1 - z \\ 1 - 3 = (0 & 1 & -1 & 0) \\ 0 & 1 & -1 & 0) + 2 = (0 & 1 & 0 & 1) \end{cases}$$

$$rank = 3$$
$$4 - 3 = 1$$

Hillert's choice was the simplest and most convenient one

Tikhonov regularization

$$\begin{cases} x_1 + x_2 = 1 \\ x_1 + x_2 = 1 \end{cases}$$

$$\underbrace{\begin{pmatrix} 1 & 1 \\ 1 & 1 \end{pmatrix}}_{A} \underbrace{\begin{pmatrix} x_1 \\ x_2 \end{pmatrix}}_{\mathbf{x}} = \underbrace{\begin{pmatrix} 1 \\ 1 \end{pmatrix}}_{\mathbf{b}} \qquad \|A\mathbf{x} - \mathbf{b}\|^2 \to \min \qquad \mathbf{x} = \left(A^{\mathsf{T}}A\right)^{-1} A^{\mathsf{T}}\mathbf{b}$$

$$\|A\mathbf{x} - \mathbf{b}\|^2 \to \min \quad \mathbf{x} = (A^{\mathrm{T}}A)^{-1} A^{\mathrm{T}}\mathbf{b}$$

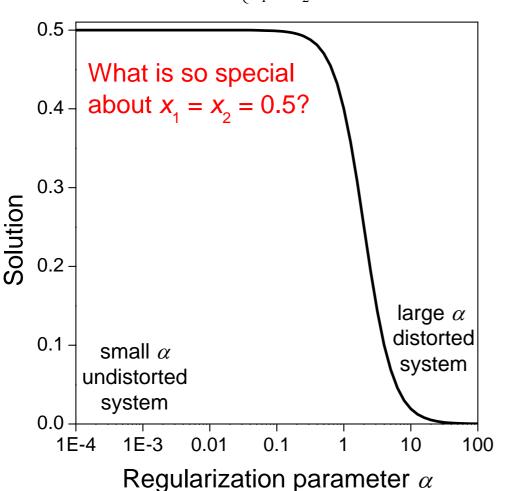
But what if A is ill-conditioned or singular?

$$\|A\mathbf{x} - \mathbf{b}\|^2 + \|\mathbf{\Gamma}\mathbf{x}\|^2 \to \min, \ \mathbf{\Gamma} = \underbrace{\alpha}_{\text{regularization parameter}} \times \mathbf{I}_{\text{matrix}}$$

$$\mathbf{x}(\alpha) = \left(A^{\mathrm{T}}A + \mathbf{\Gamma}^{\mathrm{T}}\mathbf{\Gamma}\right)^{-1}A^{\mathrm{T}}\mathbf{b}$$

How does it work?

$$\begin{cases} x_1 + x_2 = 1 \\ x_1 + x_2 = 1 \end{cases} \mathbf{x} (\alpha) = (A^{\mathsf{T}} A + \mathbf{\Gamma}^{\mathsf{T}} \mathbf{\Gamma})^{-1} A^{\mathsf{T}} \mathbf{b}$$



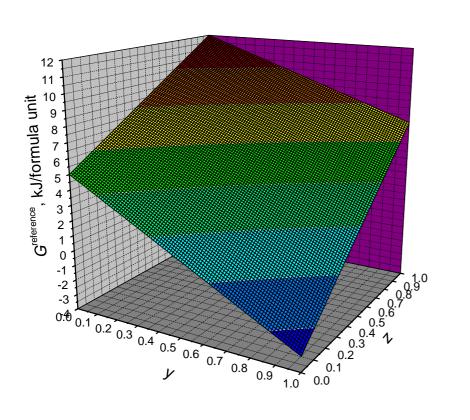
$$\begin{cases} x_1 + x_2 = 1 \\ x_1 + x_2 = 1 \\ x_1^2 + x_2^2 \to \min \end{cases}$$

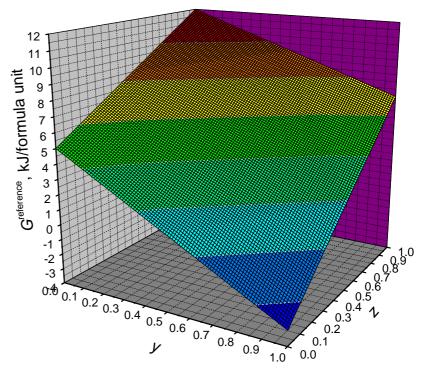
Let us choose a particular solution possessing the minimal norm

$$\alpha AC + \beta AD + \gamma BC + \delta BD \rightarrow (A_{1-y}, B_y)(C_{1-z}, D_z)$$

$$\begin{cases} \alpha + \beta = 1 - y \\ \gamma + \delta = y \\ \alpha + \gamma = 1 - z \end{cases}$$
$$\beta + \delta = z$$
$$\alpha^{2} + \beta^{2} + \gamma^{2} + \delta^{2} \rightarrow \min$$
$$\alpha \ge 0, \beta \ge 0, \gamma \ge 0, \delta \ge 0$$

Comparison

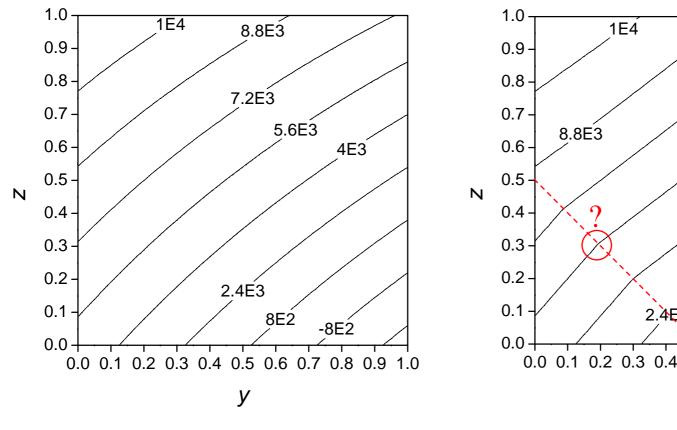


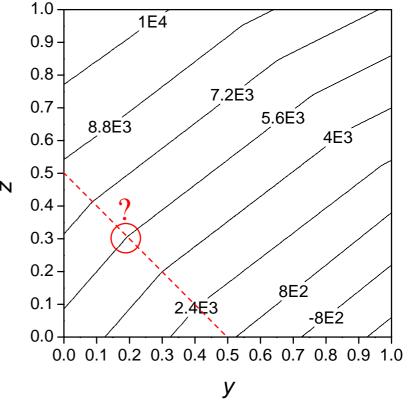


Hillert

Euclid

Comparison

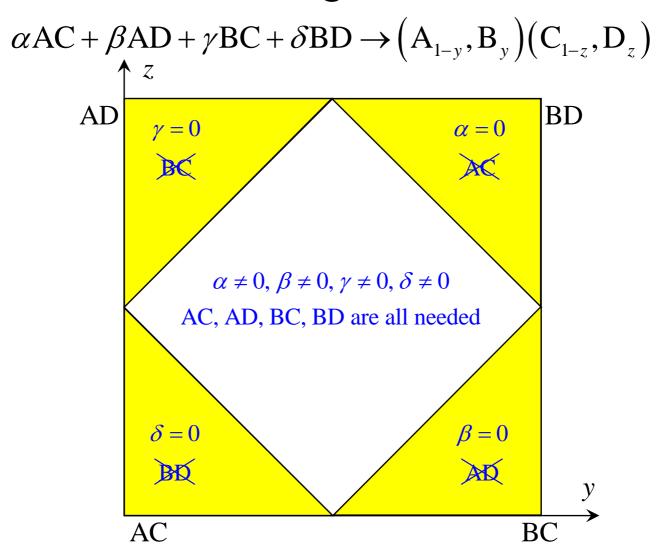




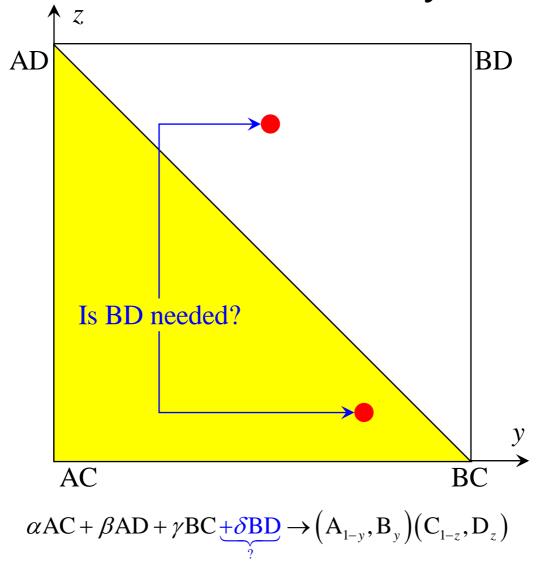
Hillert

Euclid

An interesting "side effect"



Are all end-members always needed?



Only three compounds are required

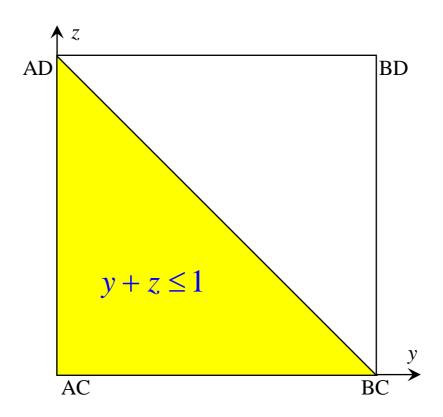
$$\alpha AC + \beta AD + \gamma BC \rightarrow (A_{1-y}, B_y)(C_{1-z}, D_z)$$

$$\begin{cases} \mathbf{A} & \alpha + \beta = 1 - y \\ \mathbf{B} & \gamma = y \\ \mathbf{C} & \alpha + \gamma = 1 - z \\ \mathbf{D} & \beta = z \end{cases}$$

$$(1-y-z)AC+zAD+yBC \rightarrow (A_{1-y},B_y)(C_{1-z},D_z)$$

Non-negativity condition

$$(1-y-z)AC + zAD + yBC \rightarrow (A_{1-y}, B_y)(C_{1-z}, D_z)$$
$$1-y-z \ge 0 \Rightarrow y+z \le 1$$



Another particular solution

$$\alpha AC + \beta AD + \gamma BC + \delta BD \rightarrow (A_{1-y}, B_y)(C_{1-z}, D_z)$$

$$\begin{cases} \alpha + \beta = 1 - y \\ \gamma + \delta = y \\ \alpha + \gamma = 1 - z \end{cases}$$

$$\beta + \delta = z$$

$$\alpha^{2} + \beta^{2} + \gamma^{2} + \delta^{2} \rightarrow \min$$

$$\alpha \ge 0, \ \beta \ge 0, \ \gamma \ge 0, \ \delta \ge 0$$

$$\begin{cases} \alpha + \beta = 1 - y \\ \gamma + \delta = y \\ \alpha + \gamma = 1 - z \end{cases}$$
$$\beta + \delta = z$$
$$\min(\alpha, \beta, \gamma, \delta) \rightarrow \min$$
$$\alpha \ge 0, \beta \ge 0, \gamma \ge 0, \delta \ge 0$$

http://mse.mcmaster.ca/