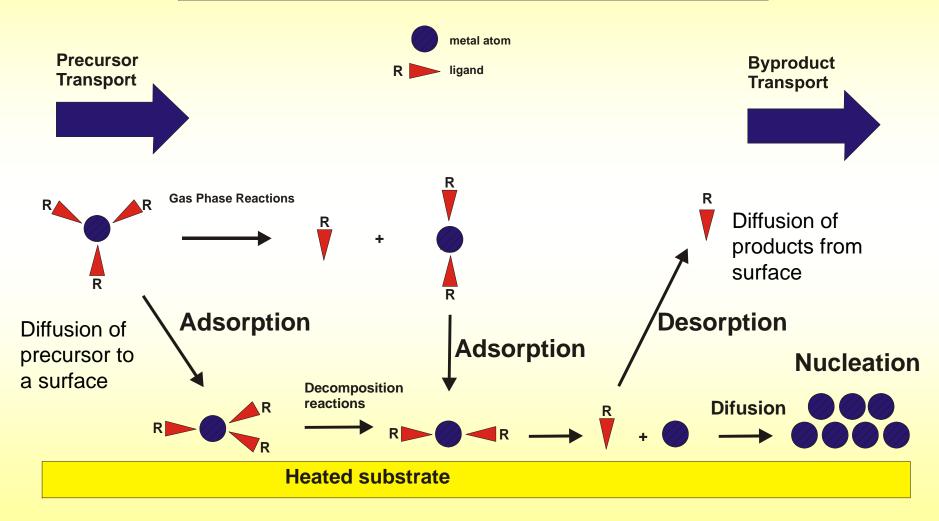
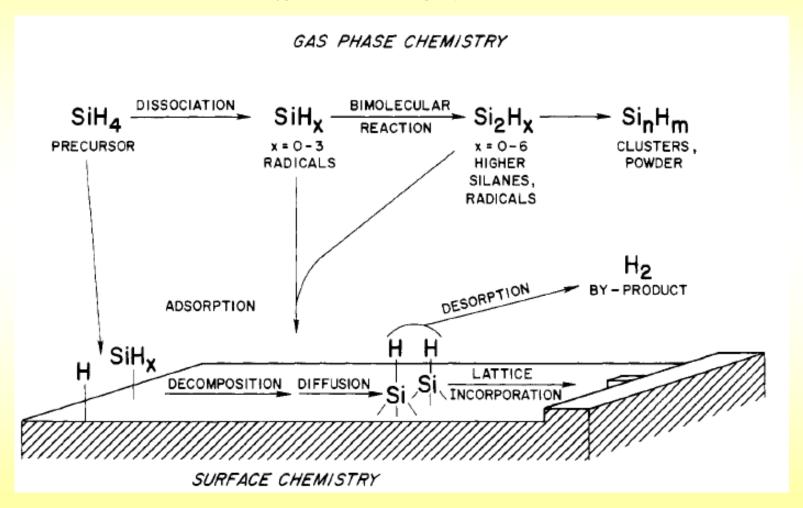
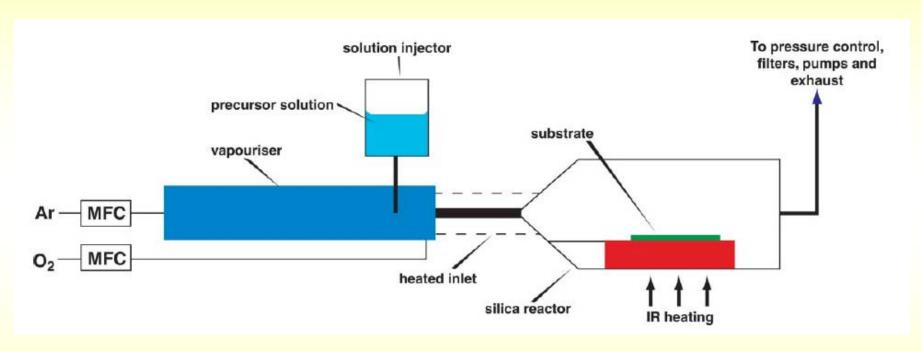
Basic steps in the CVD process



Silicon CVD

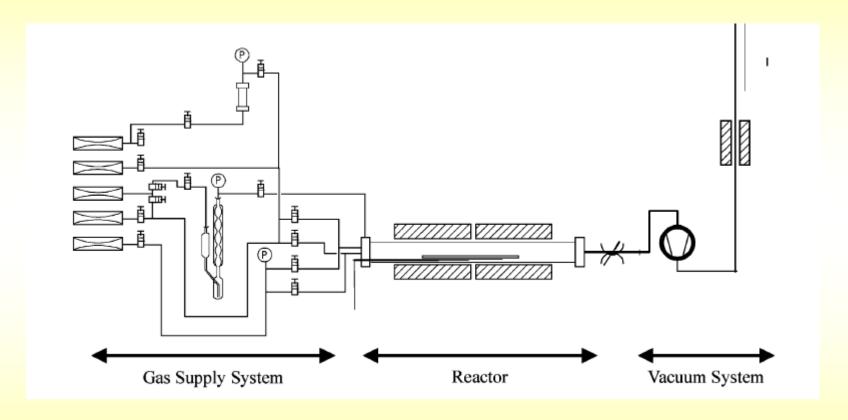


CVD Reactor



Cold-wall reactor

CVD Reactor



Hot-wall reactor

CVD_ALD_MLD

CVD Kinetics

Deposition depends on the sequence of events:

- (1) Diffusion of precursor to surface
- (2) Adsorption of precursor at surface
- (3) Chemical reaction at surface
- (4) Desorption of products from surface
- (5) Diffusion of products from surface
- The **slowest** event will be the rate-determining step

CVD Kinetics

Growth Rate Model

F1 = precursor flux from bulk of gas to substrate surface

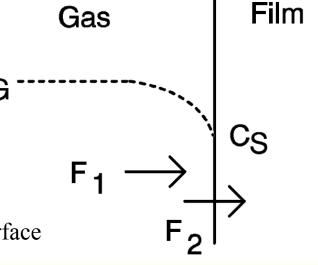
$$F1 = h_G \cdot (C_G - C_S)$$

 $h_G = \text{mass-transfer coefficient}$ $h_G = D / \delta$

 $D = gas diffusion constant D = D_0 T^{3/2} / P$

 δ = boundary layer thickness (related to gas velocity)

 C_G , C_S = precursor conc. at bulk of gas and at substrate surface (conc. gradient – driving force for diffusion)



F2 = flux consumed in film-growth reaction (rate of chemical reaction)

$$F2 = k_S \cdot C_S$$

 k_S = surface-reaction rate constant: k_S = A exp (-E_a/kT)

Steady state

$$F1 = F2 = F$$

CVD Kinetics

Steady state F1 = F2 = F

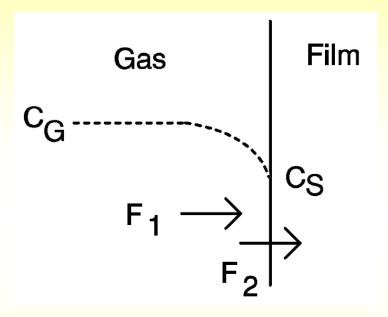
Growth Rate Model

F1 = F2 (rate of transport = rate of reaction)

$$\mathbf{h}_{\mathbf{G}} \cdot (\mathbf{C}_{\mathbf{G}} - \mathbf{C}_{\mathbf{S}}) = \mathbf{k}_{\mathbf{S}} \cdot \mathbf{C}_{\mathbf{S}}$$

$$C_S = C_G / (1 + k_S / h_G)$$

$$F = k_S h_G C_G / (k_S + h_G)$$



Growth rate (thickness growth rate)

$$dy / dt = F / \rho$$

$$y = film thickness$$

 $\rho = atomic density of film$

$$\frac{dy}{dt} = C_G \frac{1}{\rho} \frac{1}{\frac{1}{k_S} + \frac{1}{h_G}}$$

CVD_ALD_MLD

Growth Rate

$$\frac{dy}{dt} = C_G \frac{1}{\rho} \frac{1}{\frac{1}{k_S} + \frac{1}{h_G}}$$

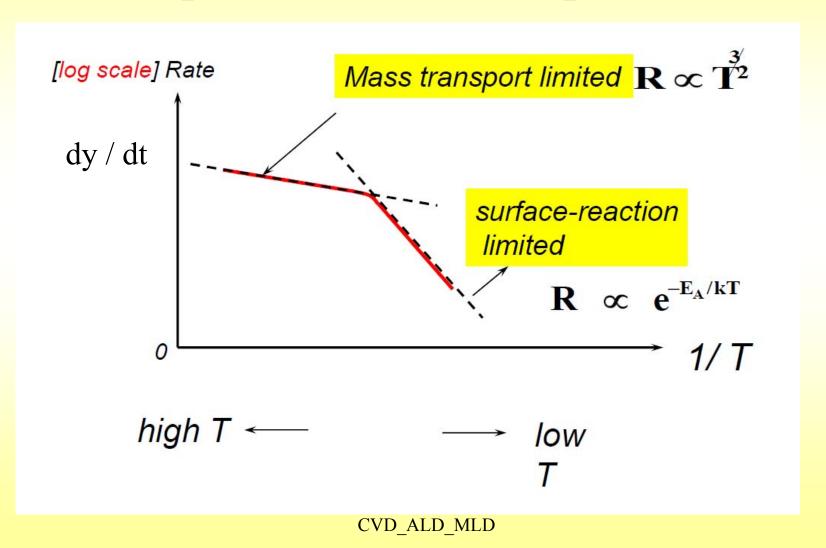
Growth rate is determined by:

- a) Concentration of a precursor in bulk of gas mixture
- b) By the smaller of h_G and k_S

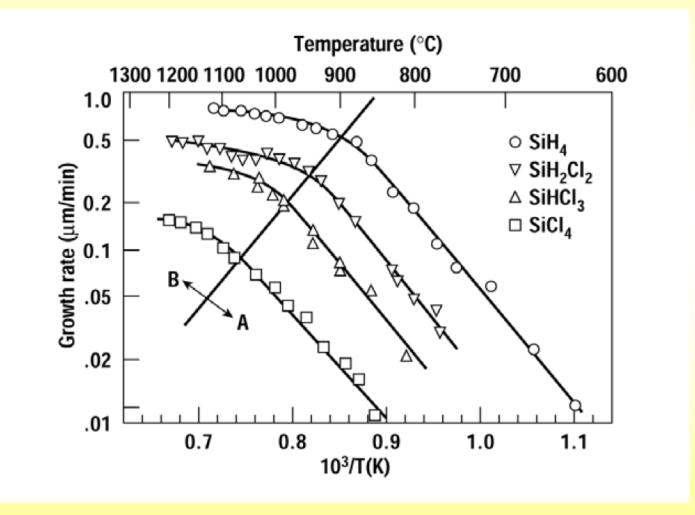
$$k_S \ll h_G$$
 = Surface reaction limited dy/dt ~ exp(-E_a/kT)

$$h_G \ll k_S$$
 = Mass transport limited dy/dt ~ $T^{3/2}$

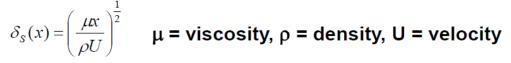
Deposition rate vs. Temperature

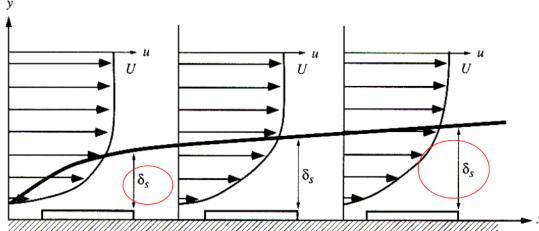


Deposition rate vs. Temperature



Growth Rate Dependence on Flow Velocity





$$F1 = h_G \cdot (C_G - C_S)$$

 $h_G = mass-transfer coefficient$

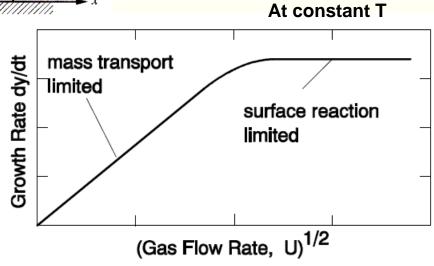
$$h_G = D / \delta$$

 δ = boundary layer thickness

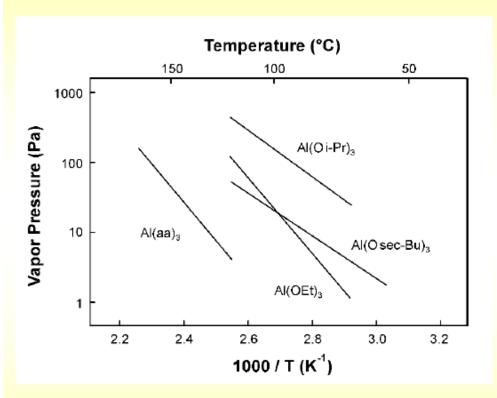
Low flow rate U

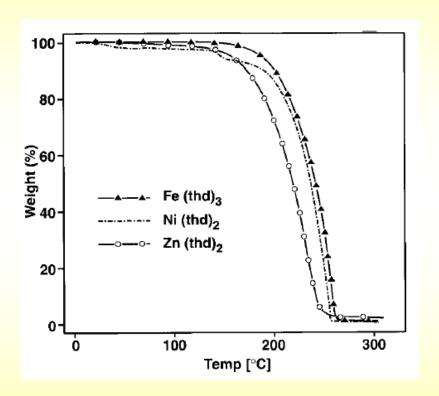
~ large boundary layer thickness δ ~ slow mass-transfer

CVD_AI



Precursor Volatility





$$\ln \frac{p_2}{p_1} = \frac{-\Delta H_{subl}^0}{R} \left(\frac{1}{T_2} - \frac{1}{T_1} \right)$$

CVD_ALD_MLD

Aluminum

2.27 μΩcm, easily etched, Al dissolves in Si,
GaAs + Al → AlAs + Ga
Gas diffusion barriers, Al on polypropylene, food packaging = chip

bags, party balloons, high optical reflectivity

TIBA β-Hydride Elimination

H
CH₃
H

β-Methyl Elimination
$$H \to CH_3$$

Al

CVD_ALD_MLD

Al deposits selectively on Al surfaces, not on SiO₂
Laser-induced nucleation
248 nm only surface adsorbates pyrolysed
193 nm gas phase reactions, loss of spatial selectivity control

TMA

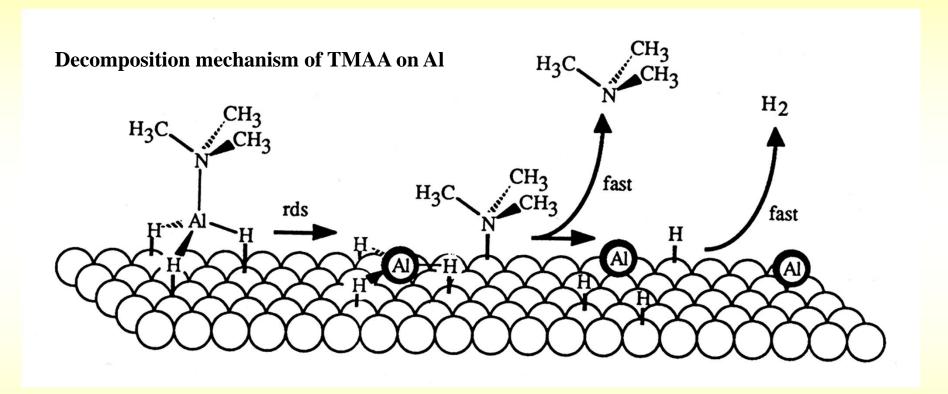
large carbon incorporation, Al₄C₃, RF plasma, laser

$$Al_2(CH_3)_6 \longrightarrow 1/2 Al_4C_3 + 9/2 CH_4$$
 under N_2

$$Al_2(CH_3)_6 + 3H_2 \longrightarrow 2Al + 6CH_4$$
 under H_2

TMAA

 $(CH_3)_3N-AlH_3$ \longrightarrow $Al + (CH_3)_3N + 3/2 H_2$ below 100 °C



$$(CH_3)_3N-AlH_3$$
 \longrightarrow $Al + (CH_3)_3N + 3/2 H_2$ below 100 °C

Aluminoboranes

$$H_{3}C$$
 CH_{3}
 C

DMAH

ligand redistribution

$$[(CH_3)_2AlH]_3 \longrightarrow (CH_3)_3Al \uparrow + AlH_3 \longrightarrow Al + H_2$$

at 280 °C, low carbon incorporation

Tungsten

5.6 $\mu\Omega$ cm, a high resistance to electromigration, the highest mp of all metals 3410 °C.

$$2 WF_6 + 3 Si \rightarrow 2 W + 3 SiF_4$$

$$WF_6 + 3H_2 \rightarrow W + 6HF$$

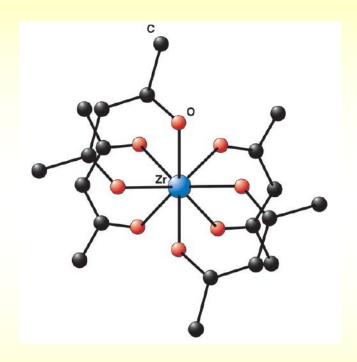
$$WF_6 + 3/2 SiH_4 \rightarrow W + 3 H_2 + 3/2 SiF_4$$

$$W(CO)_6 \rightarrow W + 6 CO$$

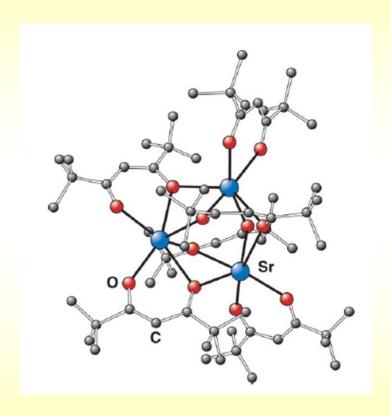
Diketonate Ligands

R ₁	R ₂	Name	Abbreviation
CH ₃	CH ₃	Pentane-2,4-dionate (acetylacetonate)	acac
CH ₃	CF ₃	1,1,1-trifluoropentane-2,4-dionate (trifluoroacetylacetonate)	tfac
CF ₃	CF ₃	1,1,1,5,5,5-hexafluoropentane-2,4-dionate (hexafluoroacetylacetonate)	hfac
CH ₃	C(CH ₃) ₃	1,1-dimethylhexane-3,5-dionate	dhd
C(CH ₃) ₃	C(CH ₃) ₃	2,2,6,6-tetramethylheptane-3,5-dionate	thd
CH ₃	CH ₂ CH(CH ₃) ₂	6-methylheptane-2,4-dionate	mhd
C(CH ₃) ₃	CH ₂ CH(CH ₃) ₂	2,2,7-trimethyloctane-3,5-dionate	tmod
C ₆ H ₅	C_6H_5	1,3-diphenylpropane-1,3-dionate (dibenyzoylmethanate)	dbm

Diketonate Precursors



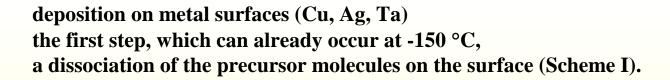
Mononuclear



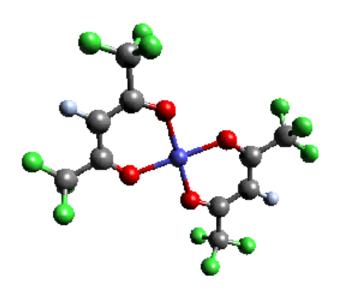
Polynuclear

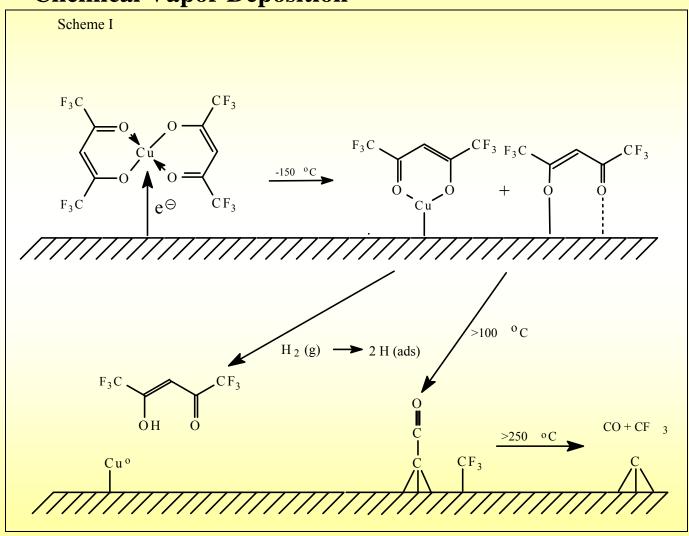
Copper(II) hexafluoroacetylacetonate

excellent volatility (a vapor pressure of 0.06 Torr at r. t.), low decomposition temperature, stability in air, low toxicity, commercial availability

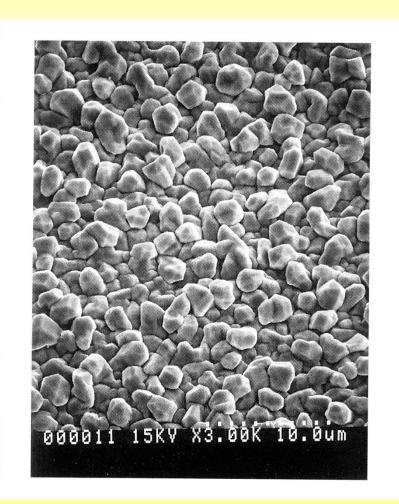


An electron transfer from a metal substrate to the single occupied HOMO which has an anti-bonding character with respect to copper d_{xy} and oxygen p orbitals weakens the Cu-O bonds and facilitates their fission.

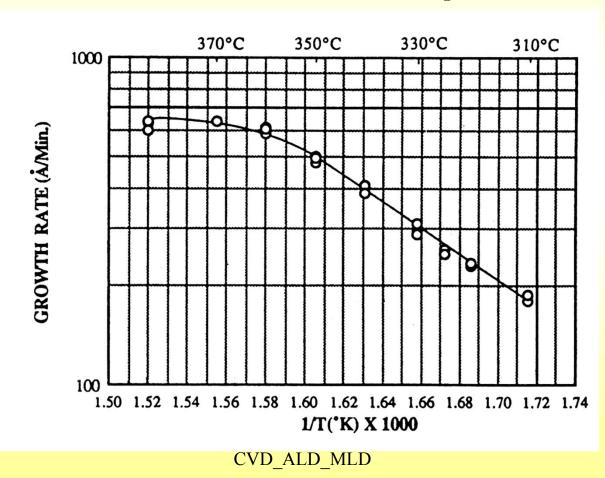




SEM of Cu film, coarse grain, high resistivity



Growth rate of Cu films deposited from Cu(hfacac)₂ with 10 torr of H₂



Cu(I) precursors

Disproportionation to Cu(0) and Cu(II)

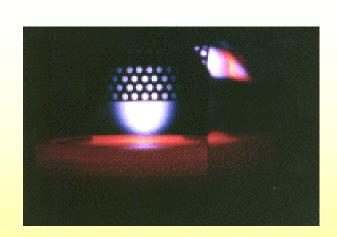
 $2 Cu(diketonate)L_n \rightarrow Cu + Cu(diketonate)_2 + n L$

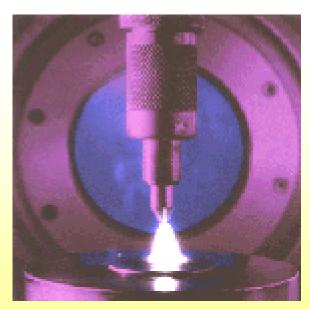
$$\begin{array}{c} R \\ O \\ C \\ U \end{array}$$

Diamond films

activating gas-phase carbon-containing precursor molecules:

- •thermal (e.g. hot filament)
- •plasma (D.C., R.F., or microwave)
- •combustion flame (oxyacetylene or plasma torches)





Experimental conditions:

temperature 1000-1400 K

the precursor gas diluted in an excess of hydrogen (typical CH₄ mixing ratio ~1-2vol%)

Deposited films are polycrystalline

Film quality:

- •the ratio of sp³ (diamond) to sp²-bonded (graphite) carbon
- •the composition (e.g. C-C versus C-H bond content)
- the crystallinity

Combustion methods: high rates (100-1000 µm/hr), small, localised areas, poor quality films.

Hot filament and plasma methods: slower growth rates (0.1-10 µm/hr), high quality films.

Hydrogen atoms generated by activation (thermally or via electron bombardment)
H-atoms play a number of crucial roles in the CVD process:

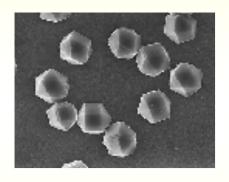
H abstraction reactions with hydrocarbons, highly reactive radicals: CH₃ (stable hydrocarbon molecules do not react to cause diamond growth) radicals diffuse to the substrate surface and form C-C bonds to propagate the diamond lattice.

H-atoms terminate the 'dangling' carbon bonds on the growing diamond surface, prevent cross-linking and reconstructing to a graphite-like surface.

Atomic hydrogen etches both diamond and graphite but, under typical CVD conditions, the rate of diamond growth exceeds its etch rate whilst for graphite the converse is true. This is the basis for the preferential deposition of diamond rather than graphite.

CVD_ALD_MLD

Diamond initially nucleates as individual microcrystals,
which then grow larger until they coalesce into a continuous film





Enhanced nucleation by ion bombardment:
damage the surface - more nucleation sites
implant ions into the lattice
form a carbide interlayer - glue, promotes diamond growth, aids adhesion

Substrates: metals, alloys, and pure elements:

Little or no C Solubility or Reaction: Cu, Sn, Pb, Ag, and Au, Ge, sapphire, diamond, graphite

C Diffusion: Pt, Pd, Rh, Fe, Ni, and Ti

the substrate acts as a carbon sink, deposited carbon dissolves into the metal surface, large amounts of C transported into the bulk,

a temporary decrease in the surface C concentration, delaying the onset of nucleation

Carbide Formation: Ti, Zr, Hf, V, Nb, Ta, Cr, Mo, W, Fe, Co, Ni, Y, Al

B, Si, SiO₂, quartz, Si₃N₄ also form carbide layers.

SiC, WC, and TiC

Applications of diamond films:

Thermal management - a heat sink for laser diodes, microwave integrated circuits active devices mounted on diamond can be packed more tightly without overheating

Cutting tools - an abrasive, a coating on cutting tool inserts

CVD diamond-coated tools have a longer life, cut faster and provide a better finish
than conventional WC tool bits

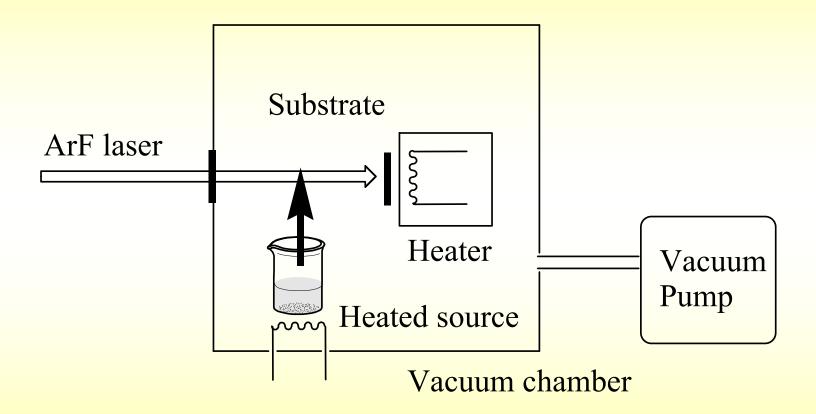
Wear Resistant Coatings -protect mechanical parts, reduce lubrication gearboxes, engines, and transmissions

Optics - protective coatings for infrared optics in harsh environments,
ZnS, ZnSe, Ge: excellent IR transmission but brittle

the flatness of the surface, roughness causes attenuation and scattering of the IR signal

Electronic devices - doping, an insulator into a semiconductor p-doping: B_2H_6 incorporates B into the lattice doping with atoms larger than C very difficult, n-dopants such as P or As, cannot be used for diamond, alternative dopants, such as Li

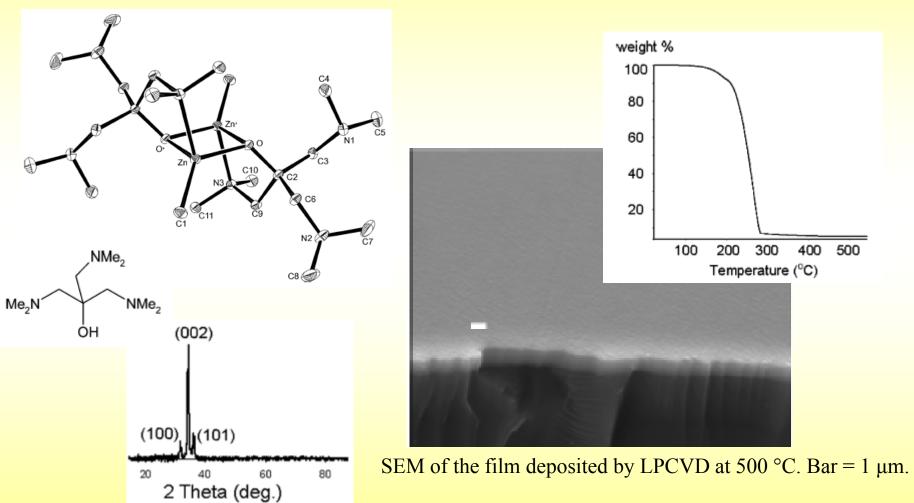
Laser-Enhaced CVD



$$Si(O_2CCH_3)_4 \rightarrow SiO_2 + 2 O(OCCH_3)_2$$

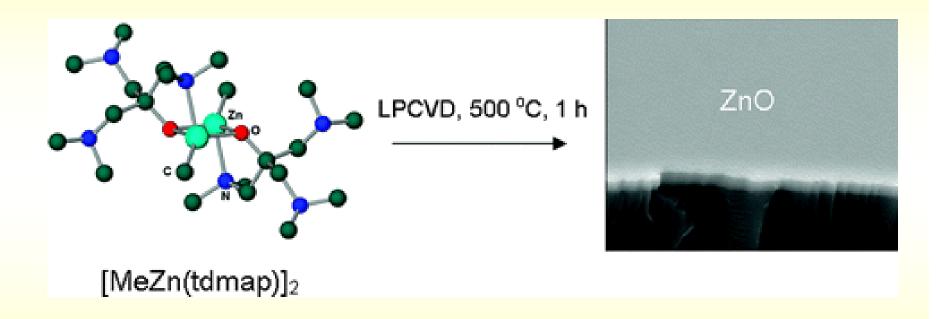
CVD_ALD_MLD

LPCVD of ZnO from Aminoalcoholates

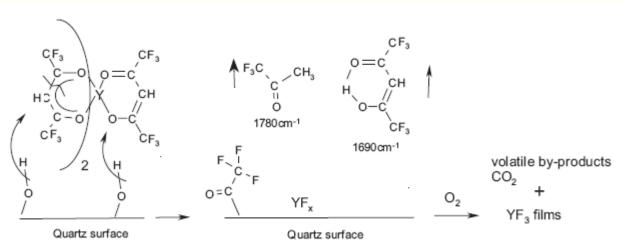


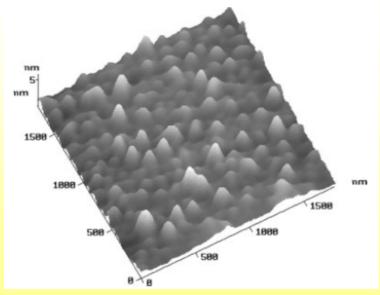
Hexagonal ZnO PDF 79-0208 CVD_ALD_MLD

LPCVD of ZnO from Aminoalcoholates



CVD of YF₃ from hfacac Complex





CVD_ALD_MLD

ALD Atomic Layer Deposition

Special modification of CVD

Method for the deposition of thin films

Film growth by cyclic process

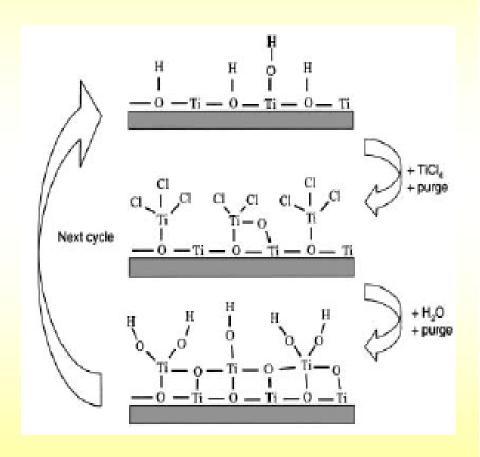
4 steps:

1/ exposition by1st precursor

2/ cleaning of the reaction chamber

3/ exposition by 2nd precursor

4/ cleaning of the reaction chamber



ALD Atomic Layer Deposition

Cycle repetitions until desired film thickness is reached

1 cycle: 0.5 s – several sec. thickness 0.1- 3 Å

Self-Limiting Growth Mechanism High reactivity Formation of a monolayer

Control of film thickness and composition

Deposition on large surface area

ALD vs. CVD Comparison

ALD Carried out at room temperature

Control over number of deposited layers = film thickness

Reactor walls inactive – no reactive layer

Separate loading of reactive precursors

Self-limiting growth

Precursor transport to the reaction zone does not have to be highly uniform (as in CVD)

Solid precursors

ALD vs. CVD Comparison

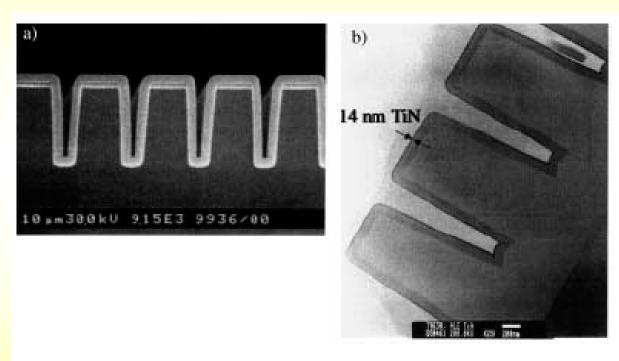


Figure 2. Cross-sectional SEM images for a 300-nm Al_2O_3 film (a) and a 14-nm TiN film (b) deposited on a patterned silicon substrate.

Precursor Properties

Selection of suitable combination of precursors

Molecular size influences film thickness

Gases, volatile liquids, solids with high vapor pressure

Typical precursors:

Metallic - halogenides (chlorides), alkyls, alkoxides, organometallics (cyclopentadienyl complexes), alkyl amides

Nonmetallic - water, hydrogen peroxide, ozone, hydrides, ammonia, hydrazine, amines

Precursor Properties

Thermally stable

Must react with surface centers (hydroxyl groups on oxide surface)

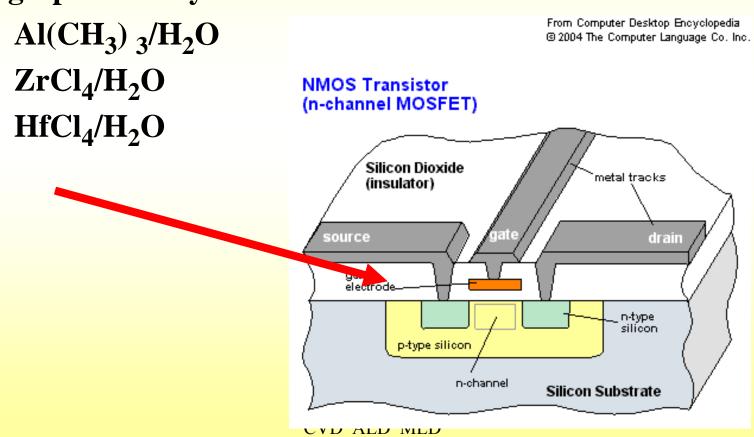
Thermodynamics

Kinetics

Mechanisms

Examples of ALD

High-permitivity Oxides



Examples of ALD

DRAM capacitors

(Ba,Sr)TiO₃ – Sr and Ba cyclopentadienyl compounds and water as precursors

Nitrides of transition metals

TiN - TiCl₄ and NH₃

TaN - TaCl₅/Zn/NH₃

WN - WF₆ and NH₃

WC_xN_y

Examples of ALD

Metallic films

Difficult by ALD: metal surface has no reaction sites, low reactivity with reducing agents

W - WF₆ and Si₂H₆

Ru, Pt - organometallic precursors and oxygen applies to all precious metals capable of catalytic dissociation of $\rm O_2$

Ni, Cu – metal oxide reduction by hydrogen radicals formed in plasma

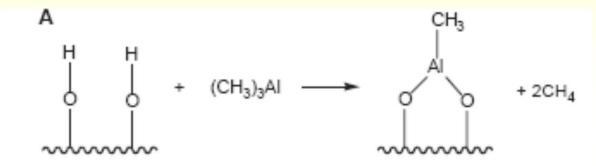
Al – direct reduction of AlMe₃ by H radicals from plasma

CVD_ALD_MLD

Precursors: trimethylalane, tris(tert-butoxy)silanol Deposition of amorphous SiO_2 and nanolaminates of Al_2O_3 32 monolayers in 1 cycle

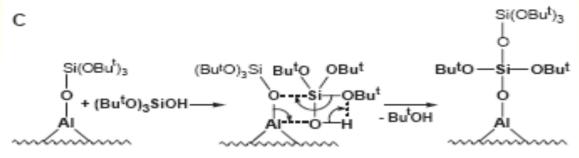
Applications:
microelectronics
optical filters
protective layers (against diffusion, oxidation, corrosion)

Step A

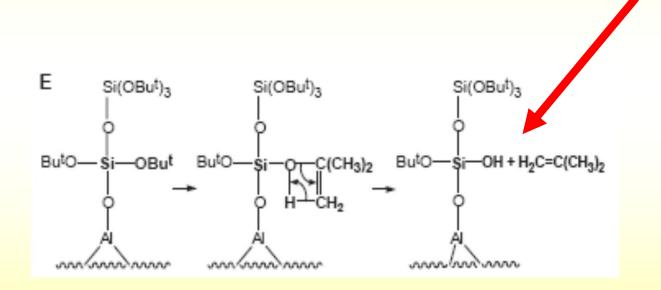


Step B

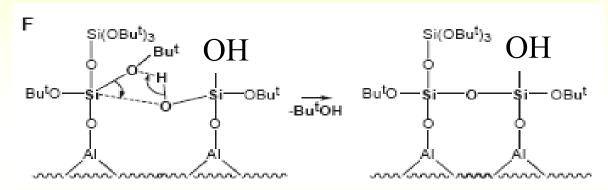
C, D: alkoxide - siloxide exchange



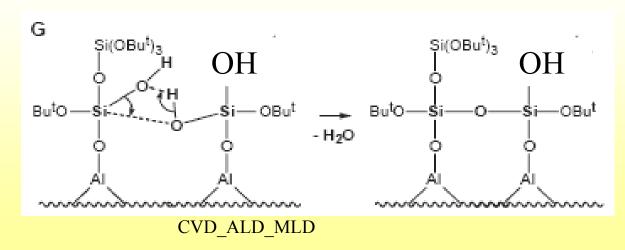
E: elimination of isobutene = formation of -OH



F: elimination of butanol = condensation

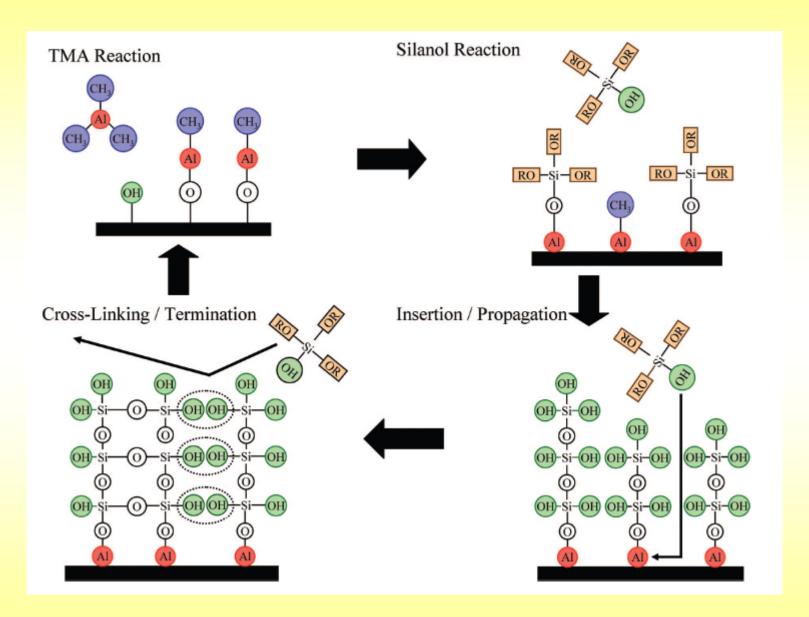


G: elimination of water = condensation

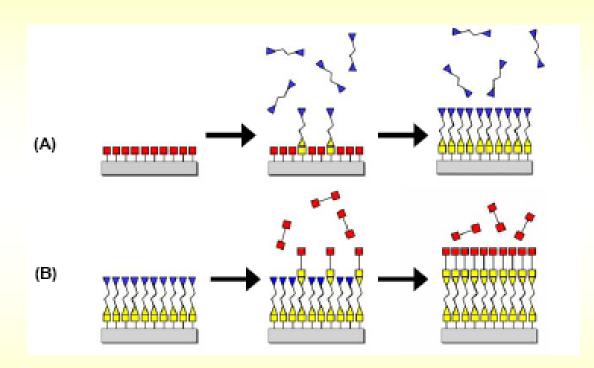


50

Repeat Step A

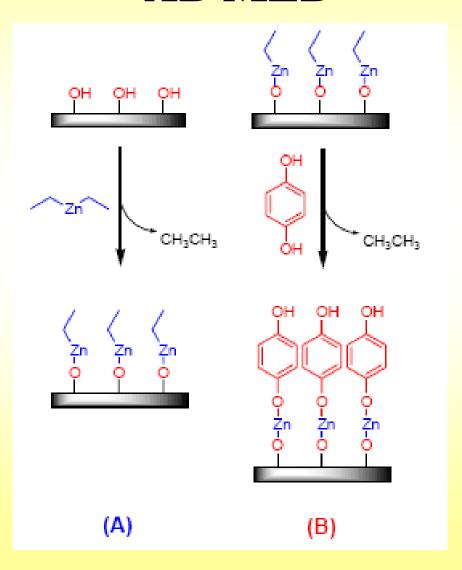


MLD - Molecular Layer Deposition

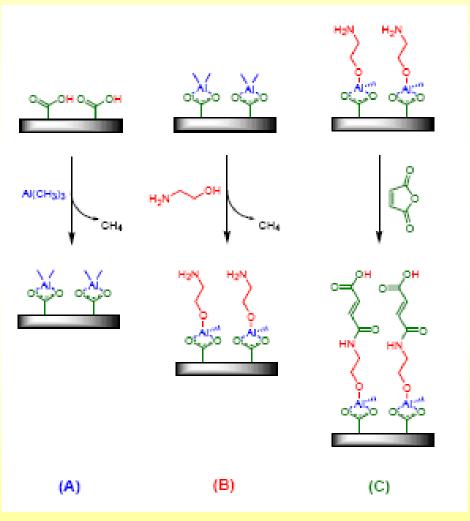


Sequential, self-limiting reactions A and B for MLD growth using two homobifunctional reactants

AB MLD

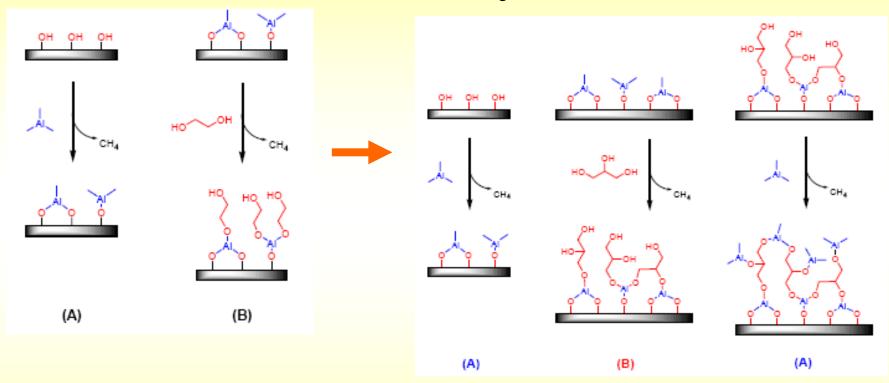


ABC MLD



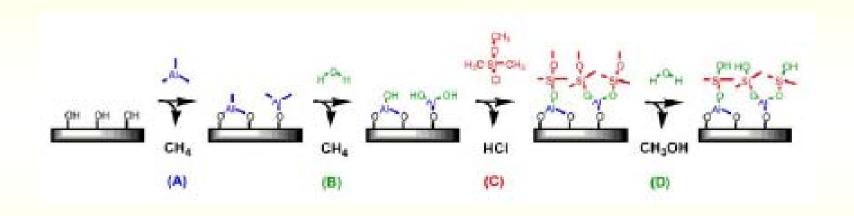
CVD_ALD_MLD

Diols vs. Polyols

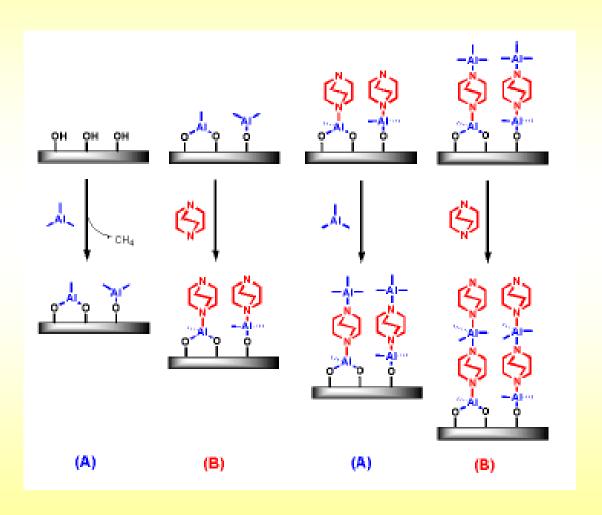


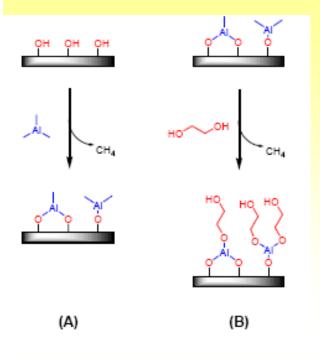
homobifunctional precursors can react twice with the AlCH₃* surface species, double reactions lead to a loss of reactive surface sites and decreasing growth rate

ABCD MLD growth of an alumina-siloxane



AB Lewis Acid-Lewis Base Reactions





Alucone

