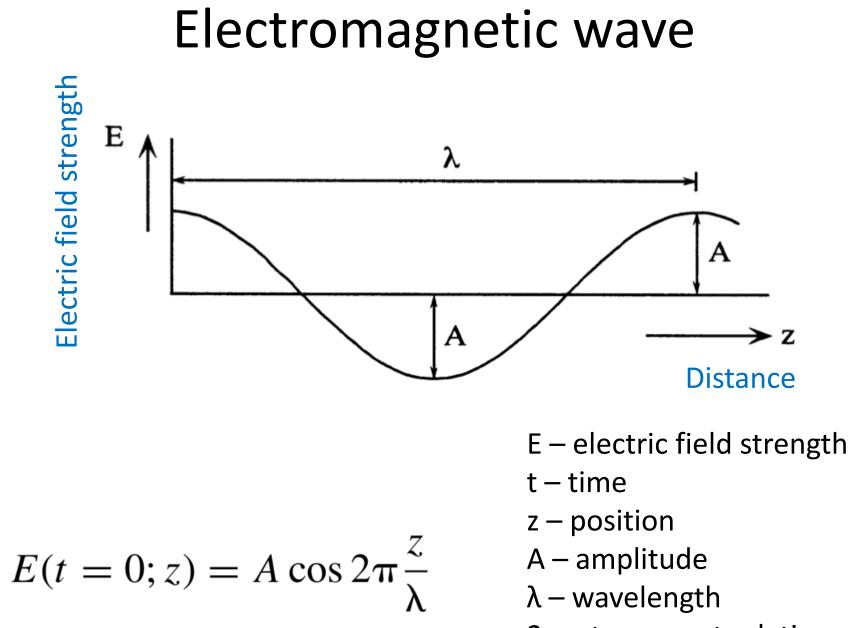
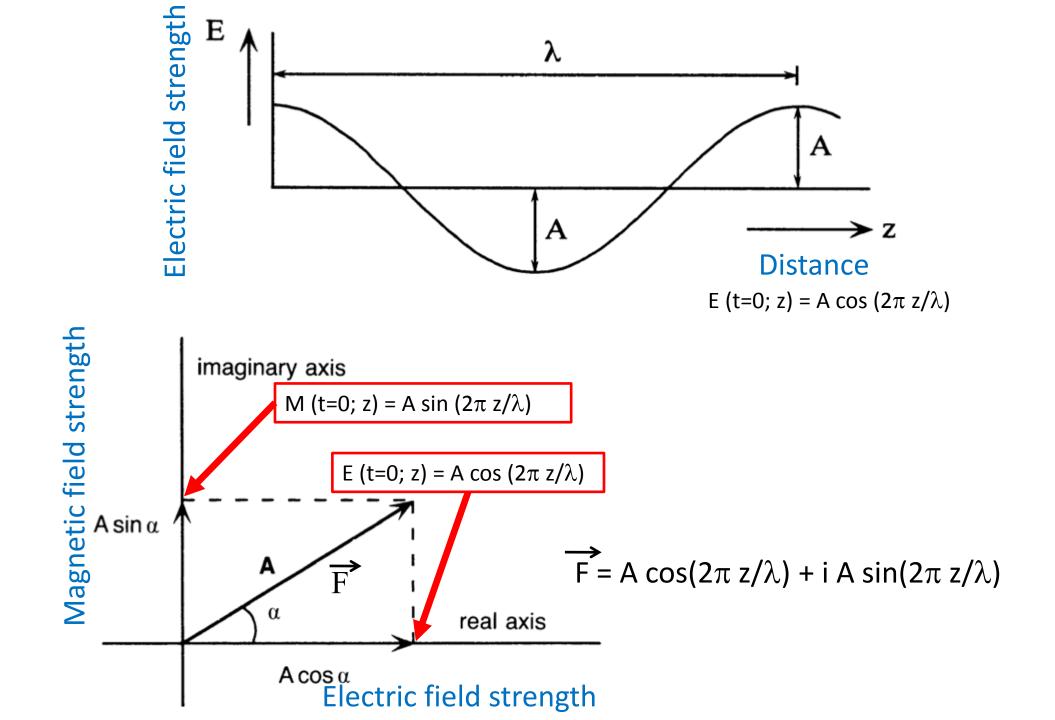
Structural Biology Methods

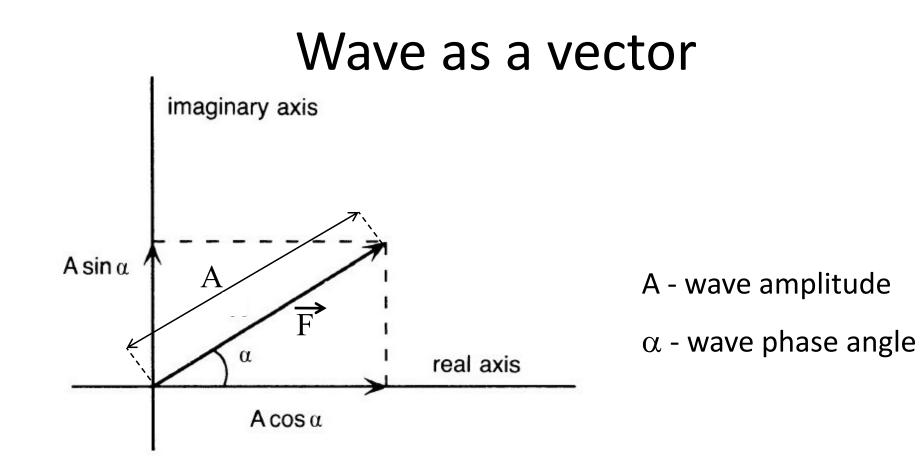
Fall 2022

Lecture #2



 2π – to convert relative distance to angles





 $\vec{F} = A \cos \alpha + i A \sin \alpha$ $\vec{F} = A \exp(i\alpha)$

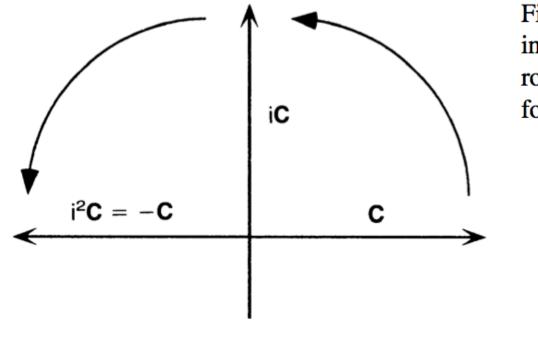


Figure 4.3. Multiplication of a vector **C** in the Argand diagram by *i* simply means rotating **C** 90° counterclockwise. Therefore, i^2 **C** = -**C**.

 $i = \sqrt{-1}$

 $A \cos \alpha + iA \sin \alpha$ $A \cos \alpha + iA \sin \alpha = A \exp[i\alpha]$

$A \cos \alpha + iA \sin \alpha = A \exp[i\alpha]$

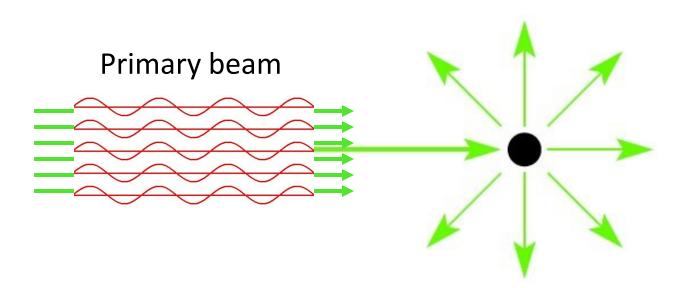
Properties of Exponential Terms

We will not prove that $A \cos \alpha + iA \sin \alpha = A \exp[i\alpha]$. You must know, however, the properties of exponential terms:

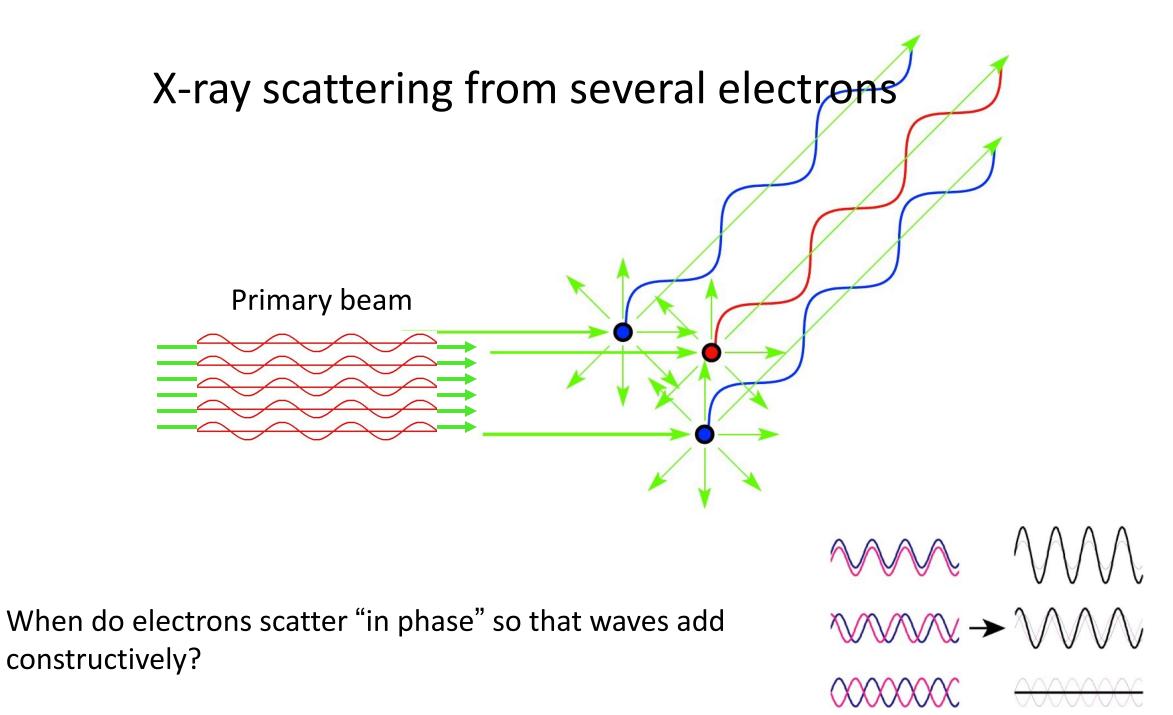
- -

$$\exp[a] \exp[b] = \exp[a + b]; \qquad \frac{\exp[a]}{\exp[b]} = \exp[a - b];$$
$$\exp[k \cdot a] = \{\exp[a]\}^k; \qquad \exp[0] = 1;$$
$$\exp[a] \to +\infty \text{ for } a \to +\infty;$$
$$\exp[a] \to 0 \text{ for } a \to -\infty.$$

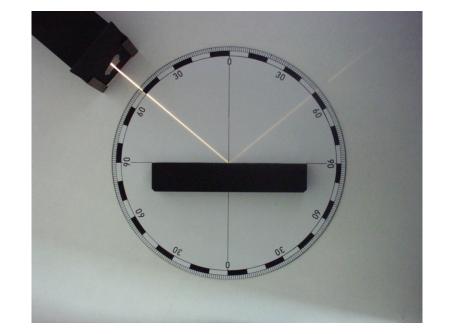
X-rays scatter from electrons in all directions

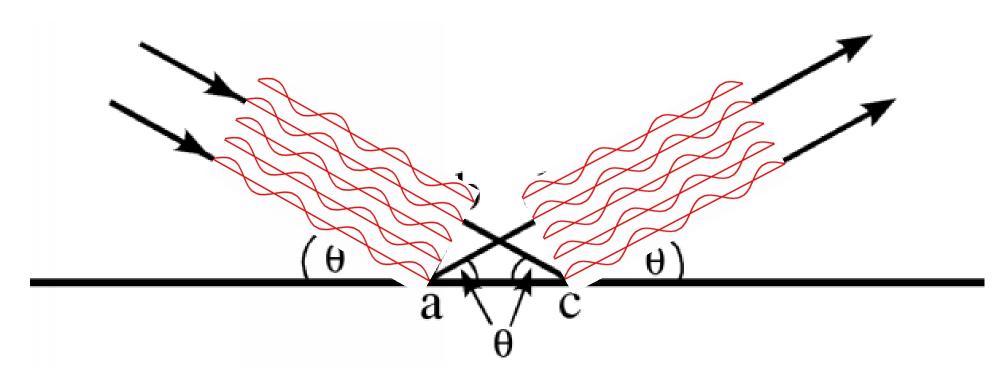


Secondary beams

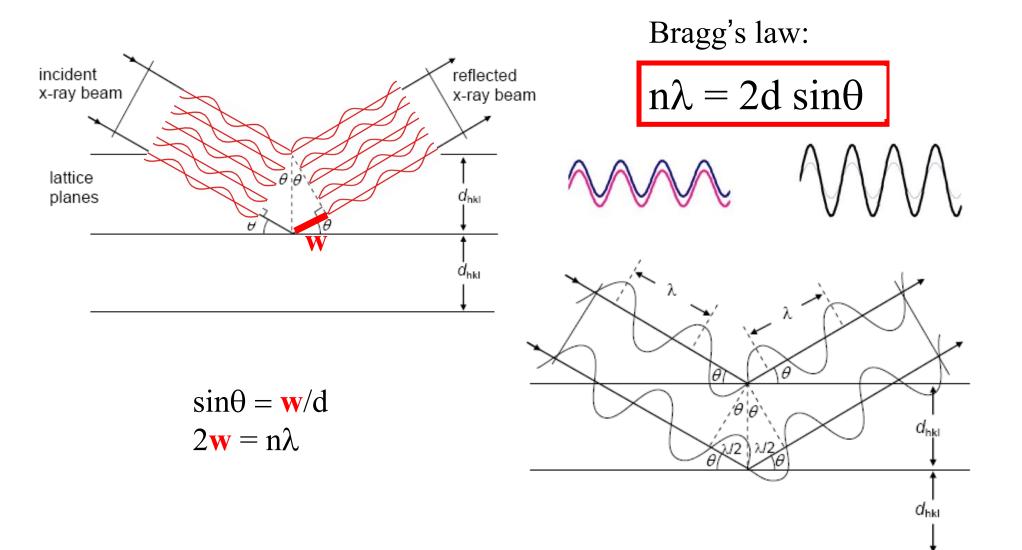


There is no path and PHASE DIFFERENCE when rays reflect from a plane

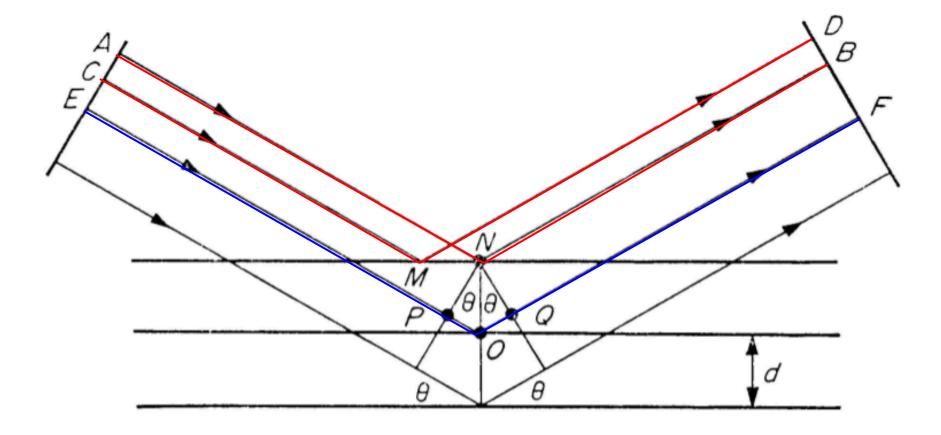




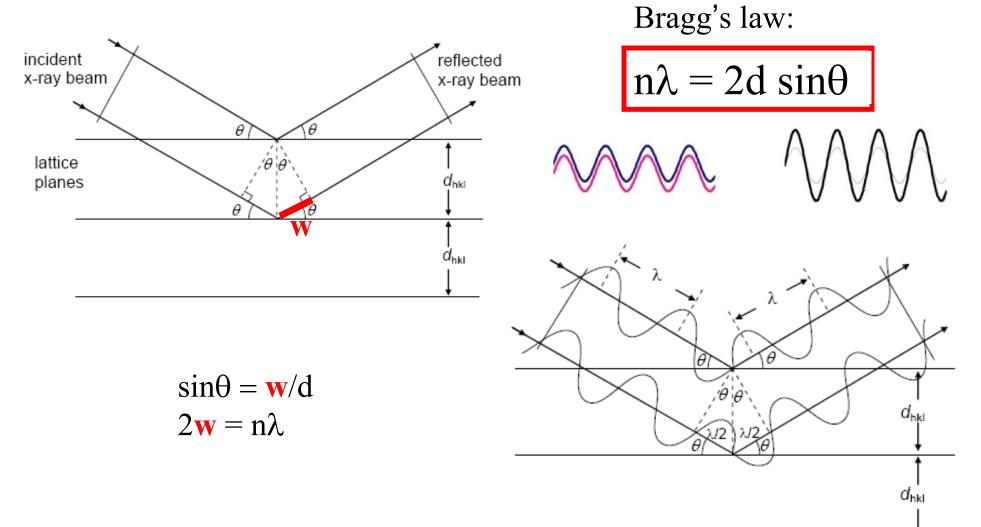
There is NO PHASE DIFFERENCE if the path differences are equal to whole number multiplies of wavelength



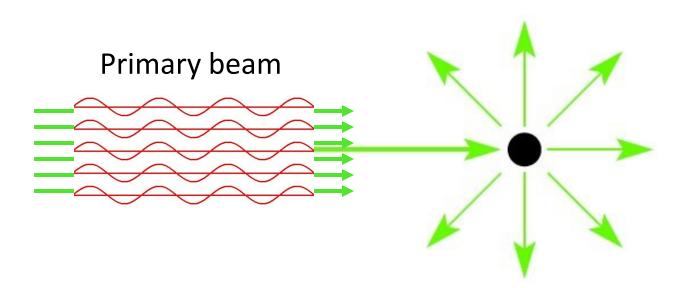
$n\lambda = 2d\sin\theta$



There is NO PHASE DIFFERENCE if the path differences are equal to prime number multiplies of wavelength (λ)

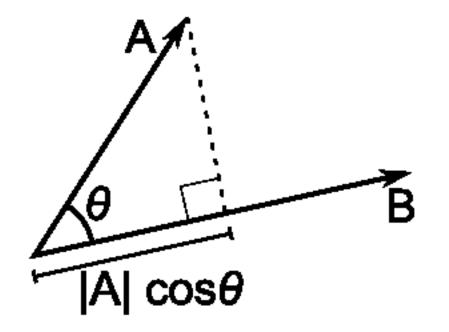


X-rays scatter from electrons in all directions



Secondary beams

Dot product



The Product of Two Vectors a and b

Let vectors **a** and **b**, with lengths |a| and |b|, be inclined at an angle θ . Scalar product: Their scalar product is the number $\mathbf{a} \cdot \mathbf{b} = ab \cos \theta$ and $\mathbf{a} \cdot \mathbf{b} = \mathbf{b} \cdot \mathbf{a}$.

System of two electrons

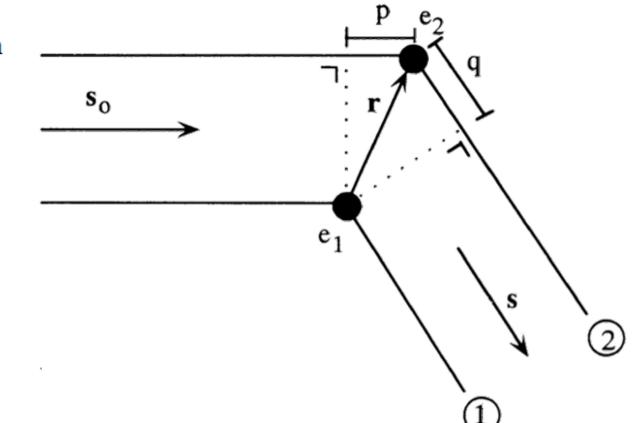
Figure 4.4. A system of two electrons: e_1 and e_2 . The path difference between the scattered waves 1 and 2 is p + q. **s**_0 **s**_0

$$p = \lambda \cdot \mathbf{r} \cdot \mathbf{s}_0$$
$$q = -\lambda \cdot \mathbf{r} \cdot \mathbf{s}$$

minus sign is due to the fact that the projection of \mathbf{r} on \mathbf{s} has a direction opposite to \mathbf{s}

 $p+q=\lambda\cdot\mathbf{r}\cdot(\mathbf{s}_0-\mathbf{s}).$

Figure 4.4. A system of two electrons: e_1 and e_2 . The path difference between the scattered waves 1 and 2 is p + q.



The wave along electron e_2 is lagging benind in phase compared with the wave along e_1 . With respect to wave 1, the phase of wave 2 is

$$-\frac{2\pi\mathbf{r}\cdot(\mathbf{s}_0-\mathbf{s})\cdot\boldsymbol{\lambda}}{\boldsymbol{\lambda}}=2\pi\mathbf{r}\cdot\mathbf{S},$$

where

$$\mathbf{S} = \mathbf{s} - \mathbf{s}_0 \tag{4.1}$$

It is interesting to note that the wave can be regarded as being reflected against a plane with θ as the reflecting angle and $|S| = 2(\sin \theta)/\lambda$ (Figure 4.5). The physical meaning of vector **S** is the following: Because $\mathbf{S} = \mathbf{s} - \mathbf{s}_0$, with $|s| = |s_0| = 1/\lambda$, **S** is perpendicular to the imaginary "reflecting plane," which makes equal angles with the incident and reflected beam.

Figure 4.5. The primary wave, represented by s_0 , can be regarded as being reflected against a plane. θ is the reflecting angle. Vector **S** is perpendicular to this plane.

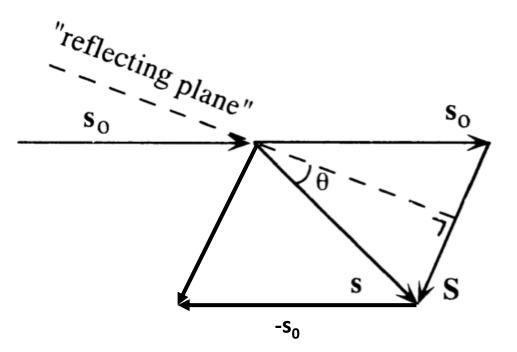
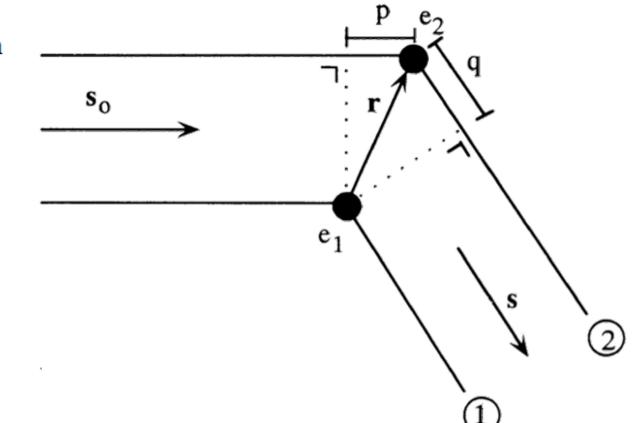


Figure 4.4. A system of two electrons: e_1 and e_2 . The path difference between the scattered waves 1 and 2 is p + q.



The wave along electron e_2 is lagging benind in phase compared with the wave along e_1 . With respect to wave 1, the phase of wave 2 is

$$-\frac{2\pi\mathbf{r}\cdot(\mathbf{s}_0-\mathbf{s})\cdot\boldsymbol{\lambda}}{\boldsymbol{\lambda}}=2\pi\mathbf{r}\cdot\mathbf{S},$$

where

$$\mathbf{S} = \mathbf{s} - \mathbf{s}_0 \tag{4.1}$$

If we add the waves 1 and 2 in Figure 4.4, the Argand diagram shows two vectors, **1** and **2**, with equal length (amplitude) and a phase of $2\pi \mathbf{r} \cdot \mathbf{S}$ for wave **2** with respect to wave **1** (Figure 4.6). Vector **T** represents the sum of the two waves. In mathematical form: $\mathbf{T} = 1 + 2 = 1 + 1 \exp[2\pi i \mathbf{r} \cdot \mathbf{S}]$ if the length of the vectors equals 1. So far we had the origin of this two-electron system in e_1 .

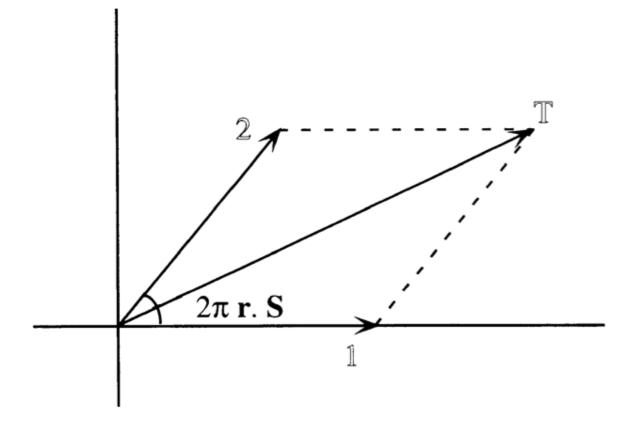


Figure 4.6. The summation of the two scattered waves in Figure 4.4 with the origin in electron e_1 .

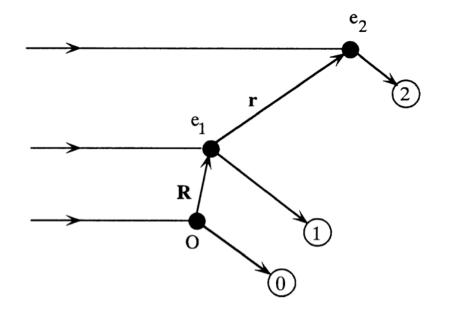


Figure 4.7. The origin, or reference point, for the scattered waves of the two-electron system is now located at O.

Suppose we

move the origin over $-\mathbf{R}$ from e_1 to point O (Figure 4.7). Then we obtain the following: With respect to a wave **0**, wave **1** has a phase of $2\pi \mathbf{R} \cdot \mathbf{S}$, and wave **2** has a phase of $2\pi (\mathbf{r} + \mathbf{R}) \cdot \mathbf{S}$ (Figure 4.8)

$$\mathbf{T} = 1 + 2 = \exp[2\pi i\mathbf{R} \cdot \mathbf{S}] + \exp[2\pi i(\mathbf{r} + \mathbf{R}) \cdot \mathbf{S}]$$
$$= \exp[2\pi i\mathbf{R} \cdot \mathbf{S}]\{1 + \exp[2\pi i\mathbf{r} \cdot \mathbf{S}]\}$$

Conclusion: A shift of the origin by $-\mathbf{R}$ causes an increase of all phase angles by $2\pi\mathbf{R} \cdot \mathbf{S}$. The amplitude and intensity (which is proportional to the square of the amplitude) of wave \mathbf{T} do not change.

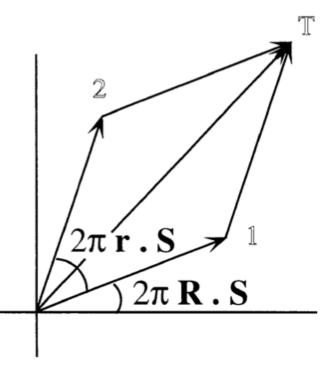


Figure 4.8. The summation of waves 1 and 2 with the origin of the two-electron system in position O.

Suppose we

move the origin over $-\mathbf{R}$ from e_1 to point O (Figure 4.7). Then we obtain the following: With respect to a wave **0**, wave **1** has a phase of $2\pi \mathbf{R} \cdot \mathbf{S}$, and wave **2** has a phase of $2\pi (\mathbf{r} + \mathbf{R}) \cdot \mathbf{S}$ (Figure 4.8)

$$\mathbf{T} = 1 + 2 = \exp[2\pi i\mathbf{R} \cdot \mathbf{S}] + \exp[2\pi i(\mathbf{r} + \mathbf{R}) \cdot \mathbf{S}]$$
$$= \exp[2\pi i\mathbf{R} \cdot \mathbf{S}]\{1 + \exp[2\pi i\mathbf{r} \cdot \mathbf{S}]\}$$

Conclusion: A shift of the origin by $-\mathbf{R}$ causes an increase of all phase angles by $2\pi\mathbf{R} \cdot \mathbf{S}$. The amplitude and intensity (which is proportional to the square of the amplitude) of wave \mathbf{T} do not change.

Scattering by an atom

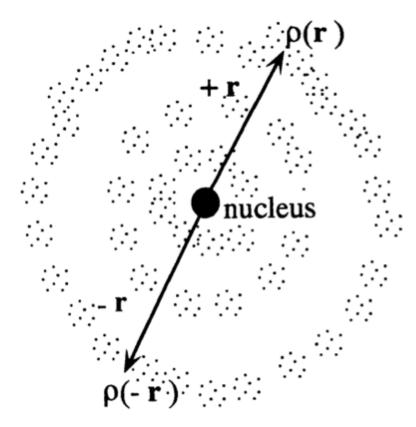
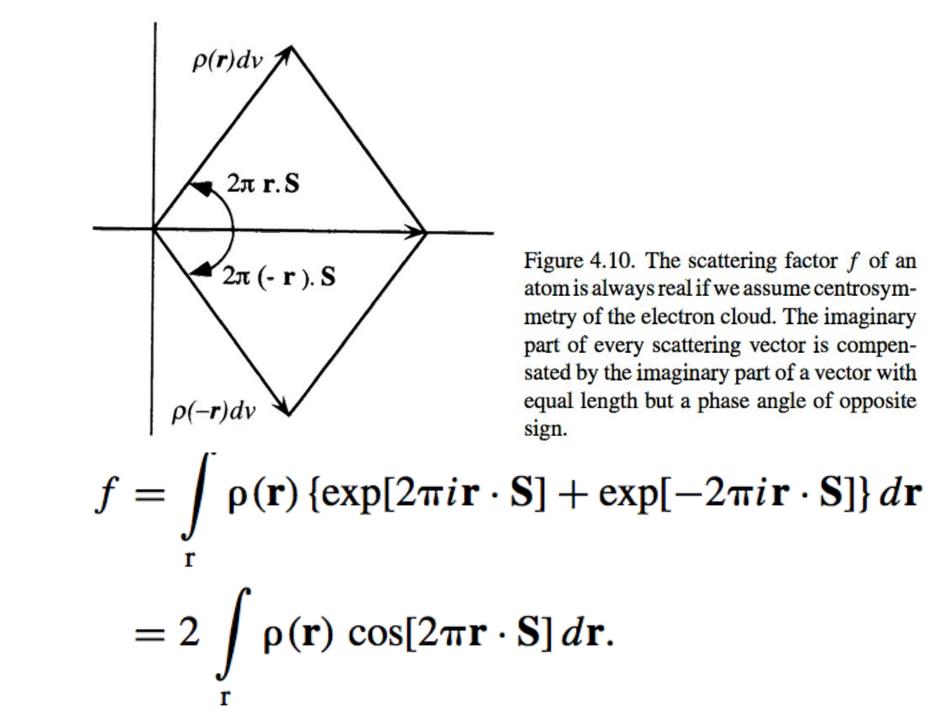


Figure 4.9. The electron cloud of an atom. $\rho(\mathbf{r})$ is the electron density. Because of the centrosymmetry, $\rho(\mathbf{r}) = \rho(-\mathbf{r})$.

$$f = \int_{\mathbf{r}} \rho(\mathbf{r}) \exp[2\pi i \mathbf{r} \cdot \mathbf{S}] d\mathbf{r},$$



Scattering by an atom depends of the length of |S| (resolution)

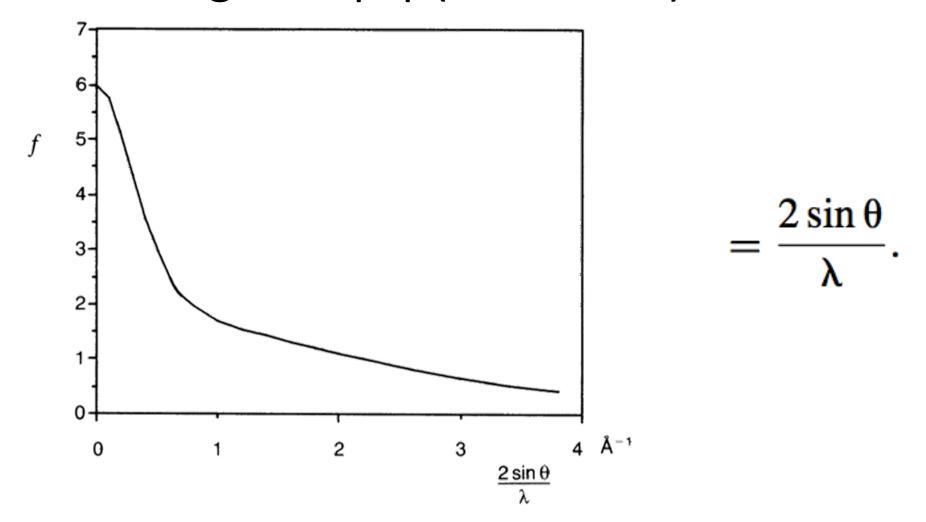
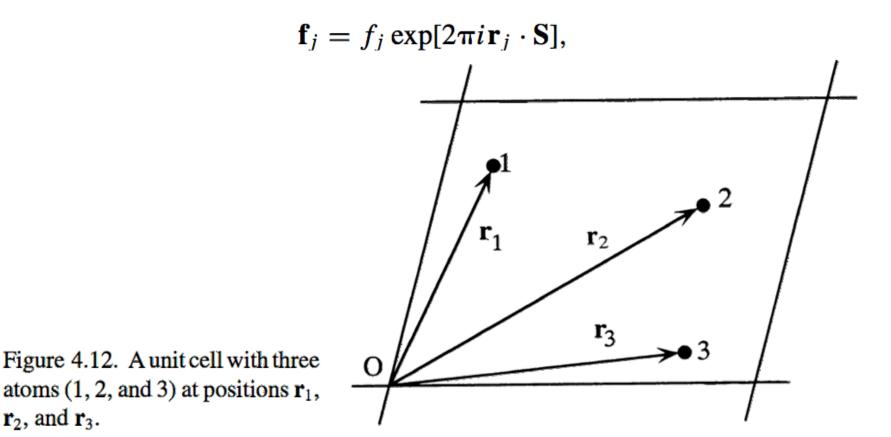


Figure 4.11. The scattering factor f for a carbon atom as a function of $2(\sin \theta)/\lambda$. f is expressed as electron number, and for the beam with $\theta = 0$, f = 6.

Scattering by a unit cell

Suppose a unit cell has *n* atoms at positions \mathbf{r}_j (j = 1, 2, 3, ..., n) with respect to the origin of the unit cell (Figure 4.12). With their own nuclei as origins, the atoms diffract according to their atomic scattering factor *f*. If the origin is now transferred to the origin of the unit cell, the phase angles change by $2\pi \mathbf{r}_j \cdot \mathbf{S}$. With respect to the new origin, the scattering is given by



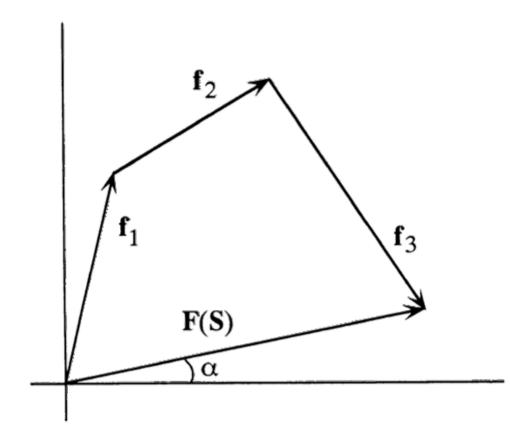


Figure 4.13. The structure factor F(S) is the sum of the scattering by the separate atoms in the unit cell.

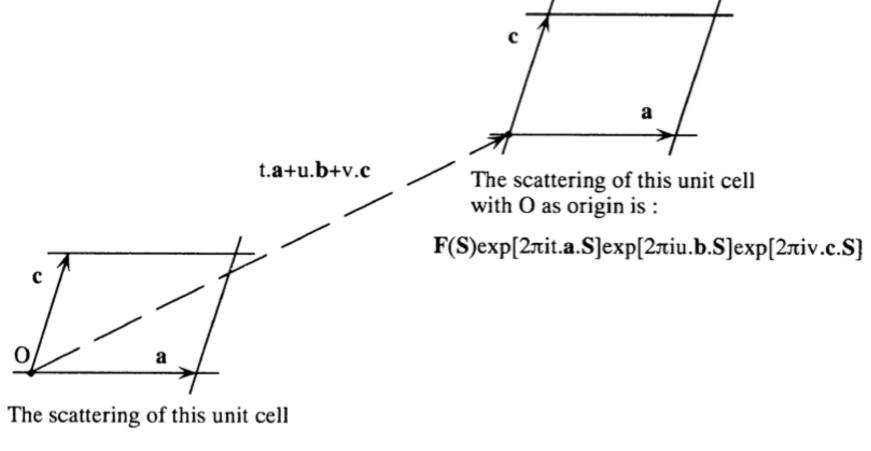
unit cell is

$$\mathbf{F}(\mathbf{S}) = \sum_{j=1}^{n} f_j \, \exp[2\pi i \mathbf{r}_j \cdot \mathbf{S}]. \tag{4.3}$$

F(S) is called the *structure factor* because it depends on the arrangement (structure) of the atoms in the unit cell (Figure 4.13).

Scattering by a crystal

Suppose that the crystal has translation vectors **a**, **b**, and **c** and contains a large number of unit cells: n_1 in the **a** direction, n_2 in the **b** direction, and n_3 in the **c** di



with O as origin is F(S)

Figure 4.14. A crystal contains a large number of identical unit cells. Only two of them are drawn in this figure.

To obtain the scattering by the crystal, we must add the scattering by all unit cells with respect to a single origin. We choose the origin O in Figure 4.14. For a unit cell with its own origin at position $t \cdot \mathbf{a} + u \cdot \mathbf{b} + v \cdot \mathbf{c}$, in which t, u, and v are whole numbers, the scattering is

 $\mathbf{F}(\mathbf{S}) \times \exp[2\pi i t \mathbf{a} \cdot \mathbf{S}] \times \exp[2\pi i u \mathbf{b} \cdot \mathbf{S}] \times \exp[2\pi i v \mathbf{c} \cdot \mathbf{S}].$

The total wave K(S) scattered by the crystal is obtained by a summation over all unit cells:

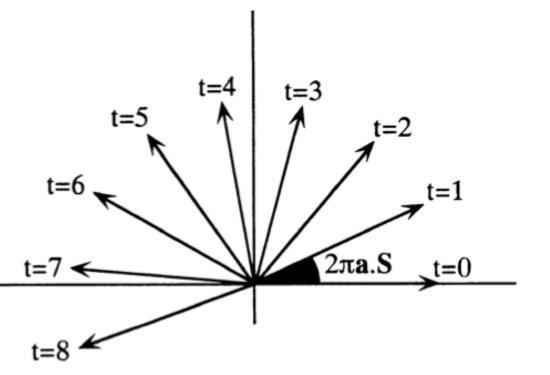
$$\mathbf{K}(\mathbf{S}) = \mathbf{F}(\mathbf{S}) \times \sum_{t=0}^{n_1} \exp[2\pi i t \mathbf{a} \cdot \mathbf{S}] \times \sum_{u=0}^{n_2} \exp[2\pi i u \mathbf{b} \cdot \mathbf{S}] \times \sum_{\nu=0}^{n_3} \exp[2\pi i \nu \mathbf{c} \cdot \mathbf{S}].$$

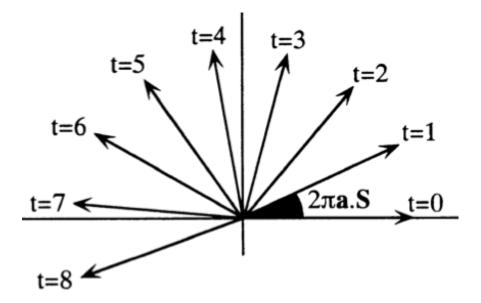
The total wave K(S) scattered by the crystal is obtained by a summation over all unit cells:

$$\mathbf{K}(\mathbf{S}) = \mathbf{F}(\mathbf{S}) \times \sum_{t=0}^{n_1} \exp[2\pi i t \mathbf{a} \cdot \mathbf{S}] \times \sum_{u=0}^{n_2} \exp[2\pi i u \mathbf{b} \cdot \mathbf{S}] \times \sum_{\nu=0}^{n_3} \exp[2\pi i \nu \mathbf{c} \cdot \mathbf{S}].$$

Because n_1 , n_2 , and n_3 are very large, the summation $\sum_{t=0}^{n_1} \exp[2\pi i t \mathbf{a} \cdot \mathbf{S}]$ and the other two over u and v are almost always equal to zero unless $\mathbf{a} \cdot \mathbf{S}$ is an integer h, $\mathbf{b} \cdot \mathbf{S}$ is an integer k, and $\mathbf{c} \cdot \mathbf{S}$ is an integer l. This is easy to understand if we regard $\exp[2\pi i t \mathbf{a} \cdot \mathbf{S}]$ as a vector in the Argand diagram with a length of 1 and a phase angle $2\pi t \mathbf{a} \cdot \mathbf{S}$ (see Figure 4.15).

Figure 4.15. Each arrow represents the scattering by one unit cell in the crystal. Because of the huge number of unit cells and because their scattering vectors are pointing in different directions, the scattering by a crystal is, in general, zero. However, in the special case that $\mathbf{a} \cdot \mathbf{S}$ is an integer h, all vectors point to the right and the scattering by the crystal can be of appreciable intensity.





Conclusion: A crystal does not scatter X-rays, unless

$$\mathbf{a} \cdot \mathbf{S} = h,$$

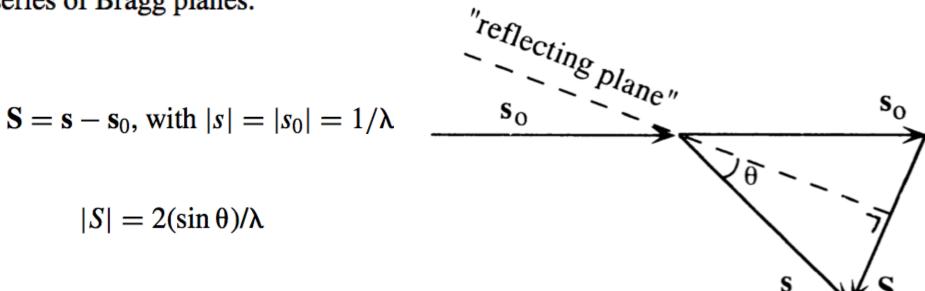
$$\mathbf{b} \cdot \mathbf{S} = k,$$

$$\mathbf{c} \cdot \mathbf{S} = l.$$
(4.4)

These are known as the Laue conditions. h, k, and l are whole numbers, either positive, negative, or zero. The amplitude of the total scattered wave is proportional to the amplitude of the structure factor F(S) and the number of unit cells in the crystal.

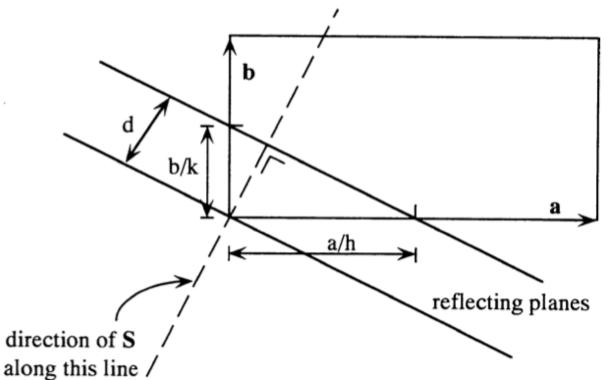
Diffraction Conditions

In Section 4.3, we noted that vector **S** is perpendicular to a "reflecting" plane. With a chosen origin for the system, $\mathbf{r} \cdot \mathbf{S}$ is the same for all points in the reflecting plane. This is true because the projection of each \mathbf{r} on **S** has the same length. Because $\mathbf{r} \cdot \mathbf{S}$ determines the phase angle, the waves from all points in a reflecting plane reflect in phase. Choose the origin of the system in the origin *O* of the unit cell. The waves from a reflecting plane through the origin have phase angle 0 ($\mathbf{r} \cdot \mathbf{S} = 0$). For a parallel plane with $\mathbf{r} \cdot \mathbf{S} = 1$, they are shifted by $1 \times 2\pi$, and so forth. All parallel planes with $\mathbf{r} \cdot \mathbf{S}$ equal to an integer are reflecting in phase and form a series of Bragg planes.



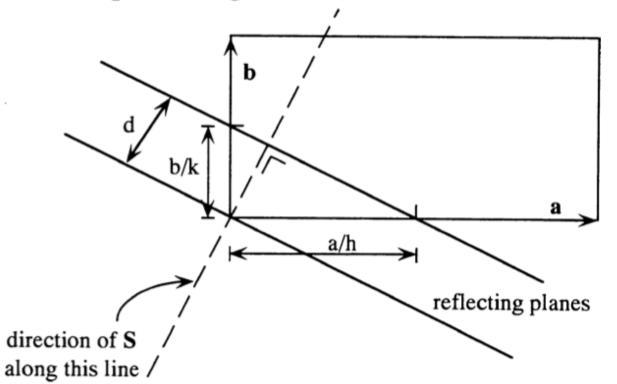
Bragg planes are identical to lattice planes

The plane with $\mathbf{r} \cdot \mathbf{S} = 1$ cuts the **a**-axis at position $\mathbf{r} = \frac{\mathbf{a}}{\xi}$. Thus $\frac{\mathbf{a}}{\xi} \cdot \mathbf{S} = 1$. But from the Laue conditions we know that $\frac{\mathbf{a}}{h} \cdot \mathbf{S} = 1$. Therefore, $\xi = h$ and in the same way the reflecting plane cuts the **b**-axis at **b**/*k* and the **c**-axis at **c**/*l*. The result is that the reflecting planes are the lattice planes.



The end points of the vectors \mathbf{a}/h , \mathbf{b}/k , and \mathbf{c}/l form a lattice plane perpendicular to vector **S** (see the text). *d* is the distance between these lattice planes.

The projection of \mathbf{a}/h on **S** has a length 1/|S|, but this projection is also equal to the distance d between the lattice planes (Figure 4.16).



The end points of the

vectors \mathbf{a}/h , \mathbf{b}/k , and \mathbf{c}/l form a lattice plane perpendicular to vector **S** (see the text). *d* is the distance between these lattice planes.

$$1/|S| = d \text{ and } |S| = 2(\sin \theta)/\lambda \implies \frac{2d \sin \theta}{\lambda} = 1$$

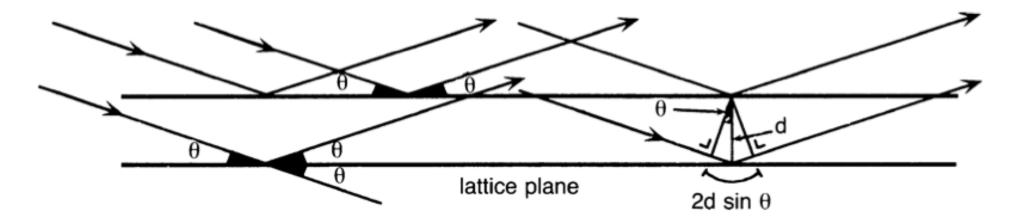


Figure 4.17. Two lattice planes are drawn separated by a distance d. The incident and the reflected beams make an angle θ with the lattice planes. Note that the beam is thus deflected through an angle of 2θ relative to its incident direction.

The incident and reflected beam make an equal angle with the plane (Figure 4.17). In a series of parallel reflecting planes (Bragg planes), the phase difference between the radiation from successive planes is 2π .

The advantage of lattice plane

reflection and Bragg's law is that it offers a visual picture of the scattering process.

Reciprocal lattice and Ewald construction

There is a crystal lattice and a reciprocal lattice. The crystal lattice is real, but the reciprocal lattice is an imaginary lattice.

Question: What is the advantage of the reciprocal lattice? Answer: With the reciprocal lattice, the directions of scattering can easily be constructed.

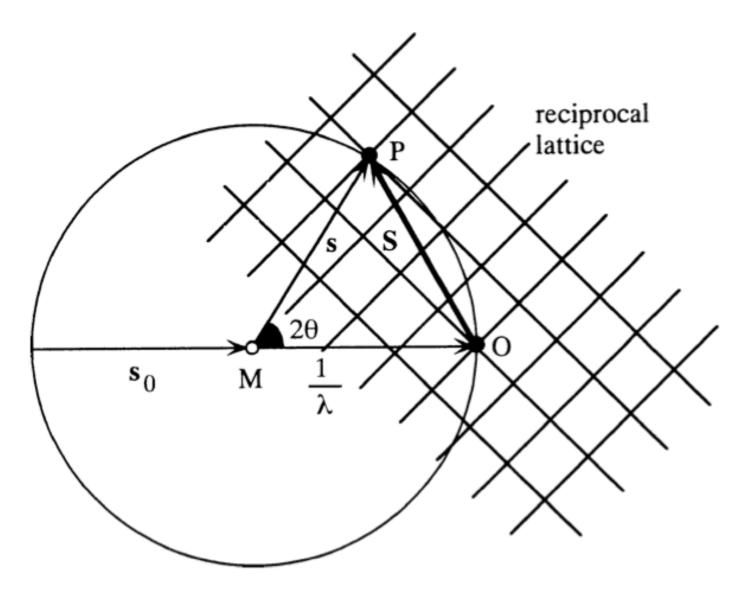


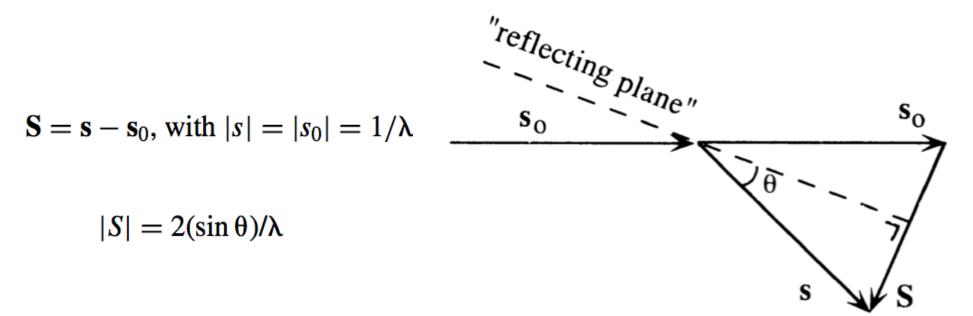
Figure 4.19. The Ewald sphere as a tool to construct the direction of the scattered beam. The sphere has radius $1/\lambda$. The origin of the reciprocal lattice is at O. \mathbf{s}_0 indicates the direction of the incident beam; \mathbf{s} indicates the direction of the scattered beam.

Expected end of lecture #2

In Section 4.3 we noted that vector S is related to the direction and angle of the reflected beam. Properties of S are as follows:

 $\mathbf{S} \perp$ reflecting plane; $|S| = \frac{2\sin\theta}{\lambda}$ because $\mathbf{S} = \mathbf{s} - \mathbf{s}_0$; and $|s_0| = |s| = \frac{1}{\lambda}$.

We also noted (Section 4.7) that the reflecting planes are in fact the lattice planes and that these lattice planes divide the **b** and **c** areas into an integral number (h, k, and l) of equal pieces. Moreover, we found that |S| = 1/d, where d is the distance between the lattice planes in one set of planes.



We now pay attention to the special planes (100), (010), and (001). For (100), the indices are h = 1, k = 0, and l = 0. **S**(100) is perpendicular to this plane and has a length of 1/d(100); we call this vector \mathbf{a}^* . In the same way, $\mathbf{S}(010) \perp$ plane (010) with length 1/d(010); we call this vector \mathbf{b}^* . Additionally, $\mathbf{S}(001) \perp$ plane (001) with length 1/d(001); we call this vector \mathbf{c}^* . Because $\mathbf{a}^* \perp$ plane (100), it is perpendicular to the **b**-axis and the **c**-axis, and, therefore, $\mathbf{a}^* \cdot \mathbf{b} = \mathbf{a}^* \cdot \mathbf{c} = 0$, but $\mathbf{a} \cdot \mathbf{a}^* = \mathbf{a} \cdot \mathbf{S}(100) = h = 1$. In the same way, it can be shown that $\mathbf{b} \cdot \mathbf{b}^* =$ $\mathbf{b} \cdot \mathbf{S}(010) = k = 1$ and $\mathbf{c} \cdot \mathbf{c}^* = \mathbf{c} \cdot \mathbf{S}(001) = l = 1$.

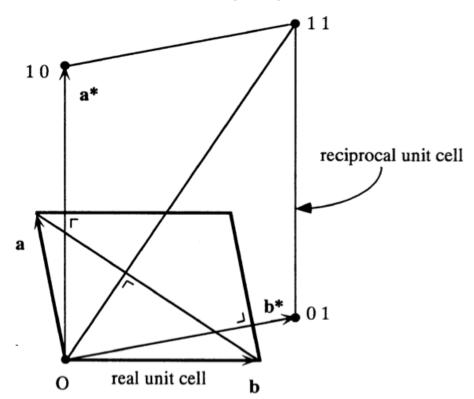


Figure 4.18. The relation between a real unit cell and the corresponding reciprocal unit cell.

Why did we introduce the vectors \mathbf{a}^* , \mathbf{b}^* , and \mathbf{c}^* ? The answer is that the end points of the vectors $\mathbf{S}(hkl)$ are located in the lattice points of a lattice constructed with the unit vectors \mathbf{a}^* , \mathbf{b}^* , and \mathbf{c}^* .

Proof: S can always be written as $S = X \cdot a^* + Y \cdot b^* + Z \cdot c^*$. Multiply by a:

$$\mathbf{a} \cdot \mathbf{S} = X \times \mathbf{a} \cdot \mathbf{a}^* + Y \times \mathbf{a} \cdot \mathbf{b}^* + Z \times \mathbf{a} \cdot \mathbf{c}^*$$

....
= $h = X \times 1 = 0 = 0$

It follows that X = h, and by the same token Y = k and Z = l. Therefore, $S = h \cdot a^* + k \cdot b^* + l \cdot c^*$. The crystal lattice based on a, b, and c is called the *direct* lattice and that based on a^* , b^* , and c^* is called the *reciprocal* lattice. Each reflection (h k l) is denoted by a point (h k l) in the reciprocal lattice. The relation between direct and reciprocal unit cells is drawn schematically in Figure 4.18.

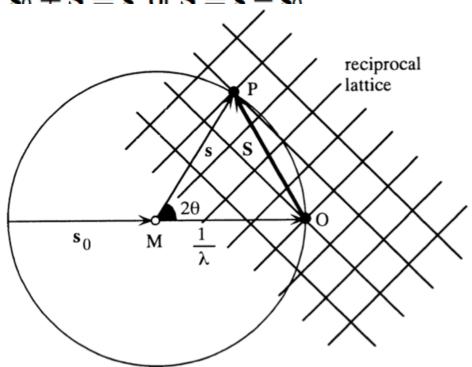
Table 4.1. Relationship Between the Axes and Angles in the Direct and the Reciprocal Lattices in a Triclinic Space Group

$a^* = rac{bc\sinlpha}{V}$	$a = \frac{b^* c^* \sin \alpha^*}{V^*}$
$b^* = \frac{ac\sin\beta}{V}$	$b = \frac{a^* c^* \sin \beta^*}{V^*}$
$c^* = \frac{ab\sin\gamma}{V}$	$c = \frac{a^* b^* \sin \gamma^*}{V^*}$
$V = \frac{1}{V*} = abc\sqrt{1 - \cos^2\alpha - \cos^2\beta - \cos^2\gamma + 2\cos\alpha\cos\beta\cos\gamma}$	
$V^* = \frac{1}{V} = a^* b^* c^* \sqrt{1 - \cos^2 \alpha^* - \cos^2 \beta^* - \cos^2 \gamma^* + 2\cos \alpha^* \cos \beta^* \cos \gamma^*}$	
$\cos \alpha^* = \frac{\cos \beta \cos \gamma - \cos \alpha}{\sin \beta \sin \gamma}$	$\cos \alpha = \frac{\cos \beta^* \cos \gamma^* - \cos \alpha^*}{\sin \beta^* \sin \gamma^*}$
$\cos\beta^* = \frac{\cos\alpha\cos\gamma - \cos\beta}{\sin\alpha\sin\gamma}$	$\cos\beta = \frac{\cos\alpha^*\cos\gamma^* - \cos\beta^*}{\sin\alpha^*\sin\gamma^*}$
$\cos \gamma^* = \frac{\cos \alpha \cos \beta - \cos \gamma}{\sin \alpha \sin \beta}$	$\cos \gamma = \frac{\cos \alpha^* \cos \beta^* - \cos \gamma^*}{\sin \alpha^* \sin \beta^*}$
$\sin \alpha \sin \beta$	$\sin \alpha^* \sin \beta^*$

If the magnitude of scattering G(S), corresponding to each vector S, is plotted at the tip of S in reciprocal space, a so-called *weighted reciprocal lattice* is obtained. For crystals, G(S) has nonzero values only at the lattice points and then G(S) = F(S). However, for nonperiodic objects, G(S) can have a nonzero value anywhere in reciprocal space.

With the reciprocal lattice, the diffraction directions can easily be constructed. The following procedure is applied:

- Step 1: Direct the incoming (primary) X-ray beam (s_0) toward the origin *O* of reciprocal space. Take the length of s_0 equal to $1/\lambda$. (Figure 4.19).
- Step 2: Construct a sphere with the origin O of reciprocal space on its surface, with center M on the line s_0 , and radius $MO = 1/\lambda$. This sphere is called the Ewald sphere.
- Step 3: If a wave vector S has its end point on the Ewald sphere (e.g., at point P), then MP is the scattered beam s. This is true because, as shown in Figure 4.3, $s_0 + S = s$ or $S = s s_0$



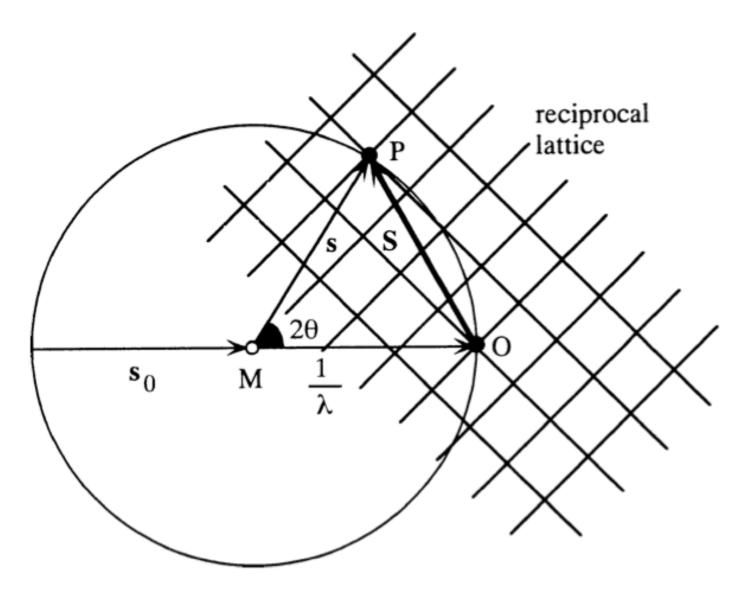
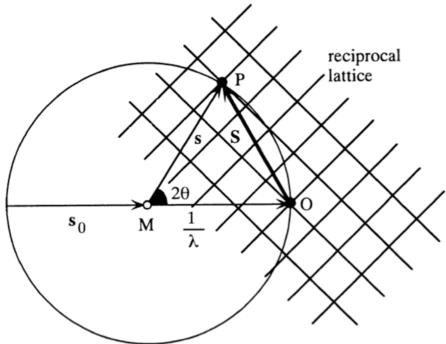


Figure 4.19. The Ewald sphere as a tool to construct the direction of the scattered beam. The sphere has radius $1/\lambda$. The origin of the reciprocal lattice is at O. \mathbf{s}_0 indicates the direction of the incident beam; \mathbf{s} indicates the direction of the scattered beam.

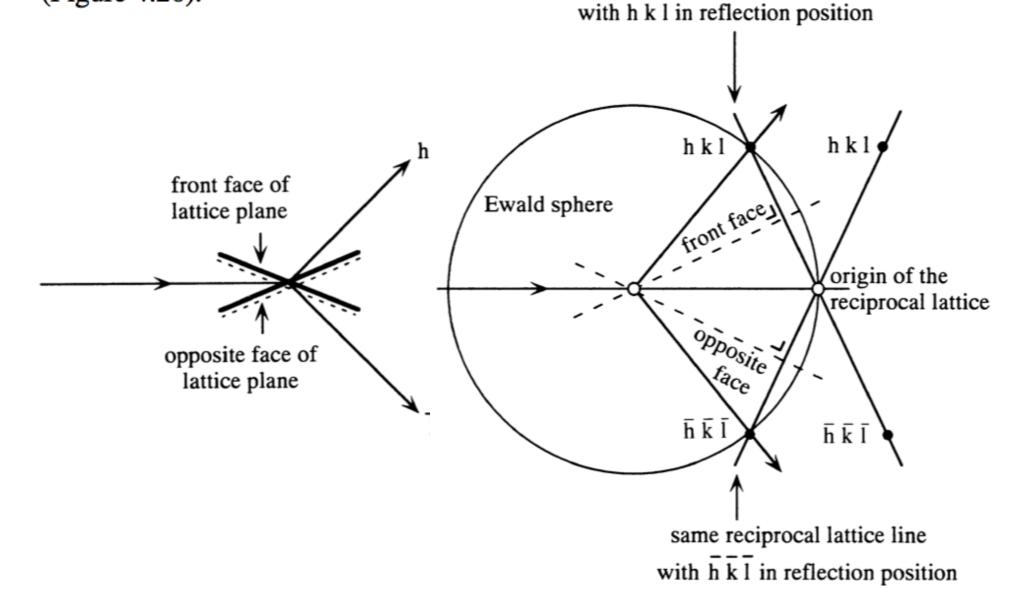
- The reciprocal lattice rotates exactly as the crystal does.
- The direction of the beam diffracted from the crystal is parallel to *MP* for the orientation of the crystal, which corresponds to the orientation of the reciprocal lattice.

From Figure 4.19, two properties of S(h k l) can easily be derived:

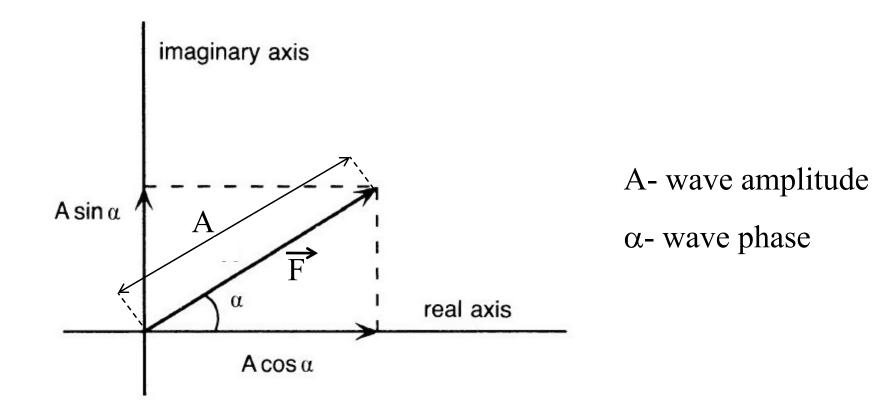
- The reciprocal space vector S(h k l) = OP(h k l) is perpendicular to the reflecting plane h k l, which is in agreement with the definition of S in Section 4.3.
 S(h h l) = 2(sin 0) () = 1 (d and Praggia law is fulfilled
- 2. $|S(h k l)| = 2(\sin \theta)/\lambda = 1/d$ and Bragg's law is fulfilled.



One more comment on lattice planes: If the beam h k l corresponds to reflection against one face (let us say the front) of a lattice plane, then $(\bar{h}\bar{k}\,\bar{l})$ [or (-h, -k, -l)corresponds to the reflection against the opposite face (the back) of the plane (Figure 4.20).



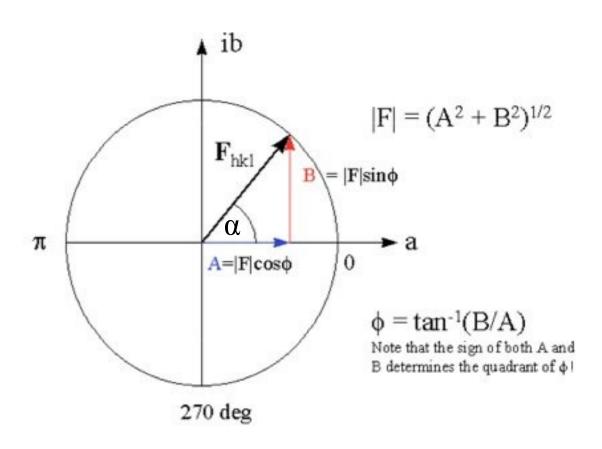
Wave as a vector



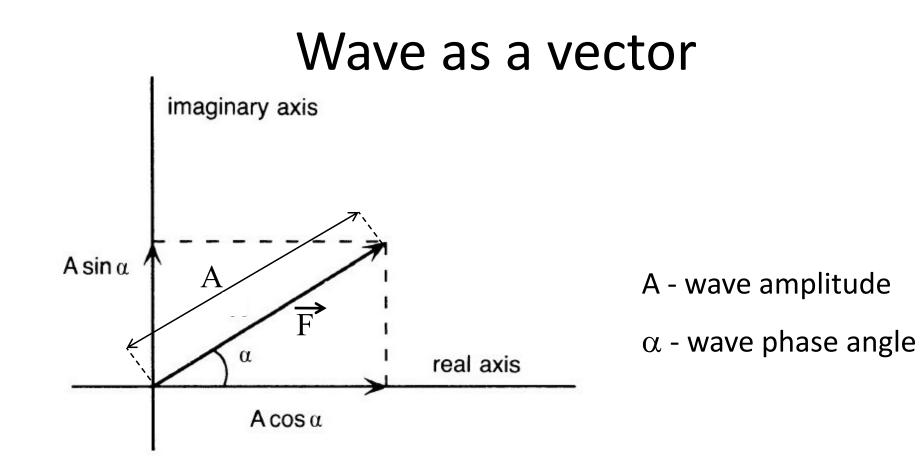
 \vec{F} =Acos α +iAsin α \vec{F} =Aexp(i α)

Phase problem

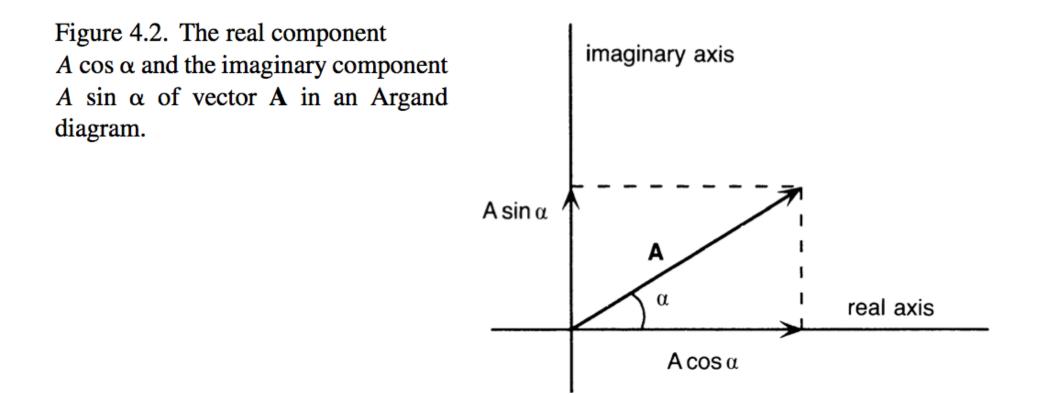
- *F_{hkl}* is complex and can be represented with an Argand diagram.
- $F_{hkl} = A + iB$
- We measured $|F_{hkl}|$ in the experiment but we still need



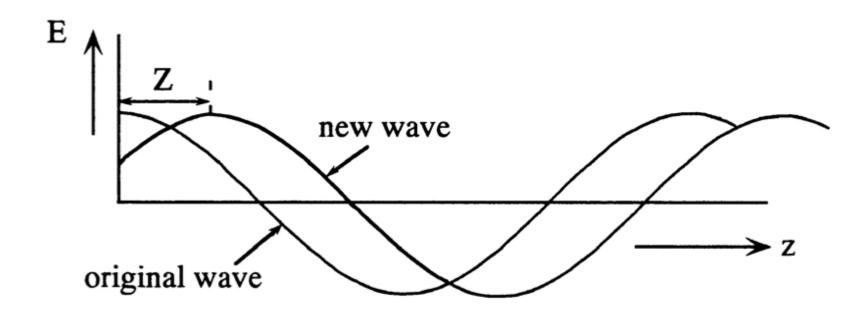
 α_{hkl} .



 $\vec{F} = A \cos \alpha + i A \sin \alpha$ $\vec{F} = A \exp(i\alpha)$



 $A \cos(\omega t + \alpha) = A \cos \alpha \cos \omega t + A \sin \alpha \sin \omega t$ $= A \cos \alpha \cos \omega t + A \sin \alpha \cos(\omega t + 90^{\circ})$ $A \cos \alpha + iA \sin \alpha$

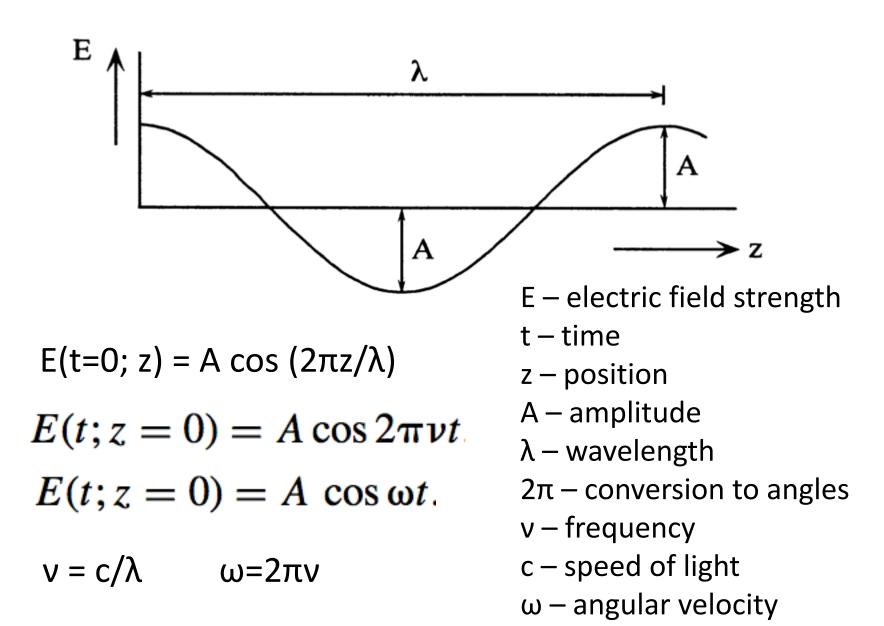


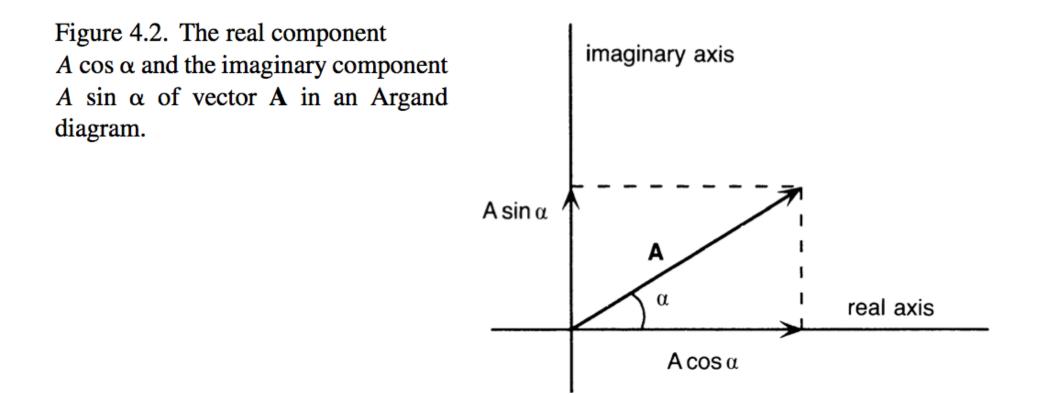
 $(Z/\lambda) \times 2\pi = \alpha$

Original wave at z = 0 and time $t : E_{orig}(t; z = 0) = A \cos \omega t$ New wave at z = 0 and time $t : E_{new}(t; z = 0) = A \cos(\omega t + \alpha)$

 $A \cos(\omega t + \alpha) = A \cos \alpha \cos \omega t + A \sin \alpha \sin \omega t$ $= A \cos \alpha \cos \omega t + A \sin \alpha \cos(\omega t + 90^{\circ})$

Wave description





 $A \cos(\omega t + \alpha) = A \cos \alpha \cos \omega t + A \sin \alpha \sin \omega t$ $= A \cos \alpha \cos \omega t + A \sin \alpha \cos(\omega t + 90^{\circ})$ $A \cos \alpha + iA \sin \alpha$