Heating: furnace, laser, plasma, flame, arc

**Gas-Metal Rxn** 

 $3 \text{ Si} + 2 \text{ N}_2 \rightarrow \text{ Si}_3 \text{N}_4$ 



W + CH<sub>4</sub>  $\rightarrow$  WC + H<sub>2</sub> mp 2720 °C WC dissolved in Co = cemented carbides (Widia materials) Cementite steel + H<sub>2</sub>/CO + CH<sub>4</sub> + NH<sub>3</sub>  $\rightarrow$  Fe<sub>3</sub>C + nitrides

**Gas-Gas Rxn** 

#### Flame hydrolysis

volatile compounds are passed through an oxygen-hydrogen stationary flame, homogeneous nucleation from supersaturated vapor (nano):





SiCl<sub>4</sub> + H<sub>2</sub>O  $\rightarrow$  OSiCl<sub>2</sub> + 2 HCl OSiCl<sub>2</sub> + H<sub>2</sub>O  $\rightarrow$  SiClOOH + HCl

 $\mathrm{SiCIOOH} \rightarrow \rightarrow \rightarrow \rightarrow \mathrm{SiO}_2 + \mathrm{HCI}$ 







# Y<sub>2</sub>O<sub>3</sub> Particles by Flame Aerosol Process



#### **Flame Aerosol Process**

Calcium phosphate nanoparticles Ca/P molar ratios 1.43 to 1.67

Synthesized by simultaneous combustion of  $Ca(OAc)_2 + OP(O^nBu)_3$  in a flame spray reactor

Fluoro-apatite and zinc or magnesium doped calcium phosphates adding trifluoroacetic acid or metal carboxylates into the fuel

Nanoparticle morphology

At a molar ratio of Ca/P < 1.5 promoted the formation of dicalcium pyrophosphate  $(Ca_2P_2O_7)$ 

Phase pure tricalcium phosphate TCP -  $Ca_3(PO_4)_2$ obtained with a precursor Ca/P ratio of 1.52 after subsequent calcination at 900 °C

Micropores and the facile substitution of both anions and cations Possible application as a biomaterial



SEM micrographs of NiCo<sub>2</sub>O<sub>4</sub> particles obtained from different concentrations of Co(OAc)<sub>2</sub> and Ni(OAc)<sub>2</sub> precursor solutions – Lower concentration reduces particle size

#### **Spray Pyrolysis**

- (1) mass flow controller  $O_2$  1 L/min
- (2) ultrasonic nebulizer aqueous
- solution 2  $Co(OAc)_2$ : 1  $Ni(OAc)_2$
- (3) 3-zone heater 400 °C
- (4) temperature controller
- (5) electrostatic precipitator





(a) HAADF-STEM of a rutile@anatase core@shell microsphere; (b) titanium L2,3 core-loss EELS spectra acquired from the indicated areas compared to reference  $TiO_2$  polymorphs [rutile (green) and anatase (red)]

#### Rutile@Anatase core@shell microspheres (b)) fitting range (C) Ti-L<sub>2.3</sub> edge ~~Zone2 Zone1 Rutile C D Anatase 500 nm 458 466 470 500 nm 462 Energy Loss (eV) (d) (e) 500 nm 500 nm 500 nm R/A

(d-f) EELS maps: (d) rutile (green), (e) anatase (red), and (f) rutile and anatase overlaid color map. (c) 3D tomographic reconstruction of another typical rutile@anatase core-shell microsphere, together with the corresponding HAADF-STEM image (inset)



He atmosphere (100 torr), U = 10-20 V, I = 0-250 AFullerene C<sub>60</sub> extracted from the soot with toluene Yields 1 – 10 %

# **Vapor Phase Transport Syntheses**

Sealed glass tube reactors Solid reactant(s) A + gaseous transporting agent B (O<sub>2</sub>, Cl<sub>2</sub>, l<sub>2</sub>, CO.....) Temperature gradient furnace  $\Delta T \sim 50 - 1000 \ ^{\circ}C$ 

A + B react at  $T_2$  to form gaseous AB (g) Equilibrium established A (s) + B (g)  $\rightleftharpoons$  AB (g) Equilibrium constant K Gaseous transport of AB (g) to the other end Concentration gradient of AB (g) = driving force for gaseous diffusion AB (g) decomposes back to A (s) at  $T_1$ , crystals of pure A Temperature dependent K Equilibrium concentration of AB (g) changes with T, different at  $T_2$  and  $T_1$ 



# **Vapor Phase Transport Syntheses**

Whether T1 < T2 or T1 > T2 depends on the thermochemical balance of the reaction !

Transport can proceed from higher to lower or from lower to higher temperature

Example: Pt (s) +  $O_2(g) \rightleftharpoons PtO_2(g)$ 

Endothermic reaction, PtO<sub>2</sub> forms at hot end, diffuses to cool end, deposits well formed Pt crystals, observed in furnaces containing Pt heating elements or thermocouples (thermometers)

Chemical vapor transport,  $T_2 > T_1$ , provides concentration gradient and thermodynamic driving force for gaseous diffusion of vapor phase transport agent AB (g)

**Uses of VPT** 

- Synthesis of new solid state materials
- Growth of single crystals
- Purification of solids

# **Thermodynamics of VPT**

van't Hoff equation

 $A(s) + B(g) \rightleftharpoons AB(g)$ 

$$\ln K_2 - \ln K_1 = \ln \frac{K_2}{K_1} = \frac{\Delta H^0}{R} \left( \frac{1}{T_1} - \frac{1}{T_2} \right)$$

Reversible equilibrium needed:  $\Delta G^{\circ} = -RT \ln K_{eq} = \Delta H^{\circ} - T \Delta S^{\circ}$ 

\* Exothermic  $\Delta H^{\circ} < 0$ Smaller T implies larger K<sub>eq</sub> AB (g) forms at cooler end, decomposes at hotter end of reactor

W + 3 Cl<sub>2</sub>  $\rightleftarrows$  WCl<sub>6</sub> 400/1400 °C (exo) Ni + 4 CO  $\rightleftarrows$  Ni(CO)<sub>4</sub> 50/190 °C (exo)

\* Endothermic  $\Delta H^{\circ} > 0$ Larger T implies larger K<sub>eq</sub> AB (g) forms at hotter end, decomposes at cooler end of reactor

2 AI + AICI<sub>3</sub>  $\rightleftarrows$  3 AICI 1000/600 °C (endo) 4 AI + AI<sub>2</sub>S<sub>3</sub>  $\rightleftarrows$  3 AI<sub>2</sub>S 1000/900 °C (endo)

### **Vapor Phase Transport Syntheses**

Estimation of the thermochemical balance ( $\Delta$ H) of a transport reaction:

$$ZnS_{(s)} + I_{2(gas)} \rightleftharpoons ZnI_{2(gas)} + S_{(g)} \Delta H = ??$$

$$Zn_{(s)} + I_{2(g)} \rightleftharpoons ZnI_{2(gas)} \qquad \Delta H_{f} = -88 \text{ kJ mol}^{-1}$$

$$ZnS_{(s)} \rightleftharpoons Zn_{(s)} + S_{(g)} \qquad -(\Delta H_{f}) = +201 \text{ kJ mol}^{-1}$$

$$\Sigma \qquad ZnS_{(s)} + I_{2(gas)} \rightleftharpoons ZnI_{2(gas)} + S_{(g)} \quad \Delta H = +113 \text{ kJ mol}^{-1}$$

Endothermic reaction, transport from hot to cold end!

Purification/crystallization of metals: Van Arkel Method

 $Cr(s) + I_2(g) \rightleftharpoons CrI_2(g)$  Exothermic

Exothermic,  $Crl_2$  (g) forms at cold end, pure Cr (s) deposited at hot end Useful for Ti, Hf, V, Nb, Cu, Ta, Fe, Th Removes metals from carbide, nitride, oxide impurities



Double Transport involving opposing Exothermic-Endothermic reactions Endothermic:

 $WO_2(s) + I_2(g) (800 \ ^\circ C) \rightleftharpoons WO_2I_2(g) (1000 \ ^\circ C)$ 

**Exothermic:** 

W (s) + 2 H<sub>2</sub>O (g) + 3 I<sub>2</sub> (g) (1000 °C)  $\rightleftharpoons$  WO<sub>2</sub>I<sub>2</sub> (g) + 4 HI (g) (800 °C)

The antithetical nature of these two reactions allows  $W/WO_2$  mixtures to be separated at different ends of the gradient reactor using  $H_2O/I_2$  as the transporting VP reagents

**Vapor Phase Transport for Synthesis** 

 $A(s) + B(g)(T_1) \rightleftharpoons (T_2) AB(g)$ AB(g) + C(s)(T\_2)  $\rightleftharpoons (T_1) AC(s) + B(g)$ 

Concept: couple VPT with subsequent reaction to give overall reaction: A (s) + C (s) (T<sub>2</sub>)  $\overrightarrow{\leftarrow}$  (T<sub>1</sub>) AC (s)

Direct reaction sluggish even at high T

 $SnO_2(s) + 2 CaO(s) \rightarrow Ca_2SnO_4(s)$  Phosphor material for light-emitting diodes

The reaction speeded up with CO as VPT agent:

$$SnO_2(s) + CO(g) \rightleftharpoons SnO(g) + CO_2(g)$$
  
 $SnO(g) + CO_2(g) + 2 CaO(s) \rightleftharpoons Ca_2SnO_4(s) + CO(g)$ 

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**Direct reaction is sluggish:** 

 $Cr_2O_3(s) + NiO(s) \rightarrow NiCr_2O_4(s)$  Magnetic materials

 $Cr_2O_3(s) + 3/2O_2 \rightleftharpoons 2CrO_3(g)$  Greatly enhanced rate with  $O_2$ 

 $2 \operatorname{CrO}_3(g) + \operatorname{NiO}(s) \rightleftharpoons \operatorname{NiCr}_2O_4(s) + 3/2 O_2(g)$ 

**Overcoming Passivation Through VPT** 

2 Al (s) + 3 S (s)  $\rightarrow$  Al<sub>2</sub>S<sub>3</sub> (s) Passivating skin stops reaction

In presence of surface cleansing VPT agent I<sub>2</sub>:

**Endothermic:** 

 $AI_2S_3(s) + 3I_2(g) (700 \ ^{\circ}C) \rightleftharpoons 2AII_3(g) + 3/2S_2(g) (800 \ ^{\circ}C)$ 

**Vapor Phase Transport for Synthesis** 

 $Zn(s) + S(s) \rightarrow ZnS(s)$  passivation prevents reaction to completion

**Endothermic:** 

ZnS (s) +  $I_2$  (g) (800 °C)  $\rightleftharpoons$  Zn $I_2$  (g) +  $\frac{1}{2}$  S<sub>2</sub> (g) (900 °C)

VPT Synthesis of  $ZnWO_4$  from  $WO_3$  and ZnOa phosphor host crystal for Ag<sup>+</sup>, Cu<sup>+</sup>, Mn<sup>2+</sup>

WO<sub>3</sub> (s) + 2 Cl<sub>2</sub> (g) (980 °C)  $\overrightarrow{\leftarrow}$  WO<sub>2</sub>Cl<sub>2</sub> (g) + Cl<sub>2</sub>O (g) (1060 °C) ZnO (s) + WO<sub>2</sub>Cl<sub>2</sub> (g) + Cl<sub>2</sub>O (g) (1060 °C)  $\overrightarrow{\leftarrow}$  ZnWO<sub>4</sub> (s) + Cl<sub>2</sub> (g)

Growth of epitaxial GaAs films or single crystals by VPT

GaAs (s) + HCl (g)  $\rightleftharpoons$  GaCl (g) +  $\frac{1}{2}$  H<sub>2</sub> (g) +  $\frac{1}{4}$  As<sub>4</sub> (g) Endothermic



# **Laser-induced Homogeneous Pyrolysis**

#### Laser wavelength 10.60 ± 0.05 μm

- Overlap between the vertical reactant gas stream and the horizontal laser beam
- Reaction zone away from the chamber walls
- Nucleation of nanoparticles
- Less contamination
- Narrow size distribution



Excitation energy transferred to vibrationaltranslational modes ⇒ T increases



ENERGY —

# Iron-oxide Nanoparticles by Laser-induced Homogeneous Pyrolysis

#### $2 \operatorname{Fe}(\operatorname{CO})_5 + 3 \operatorname{N}_2 \operatorname{O} \rightarrow \operatorname{Fe}_2 \operatorname{O}_3 + 10 \operatorname{CO} + 3 \operatorname{N}_2$

