Basic steps in the CVD process

metal atom

R ▶ ligand

Precursor Transport

Byproduct Transport

Gas Phase Reactions

R + R

Diffusion of products from surface

Diffusion of precursor to a surface

Adsorption

Adsorption

Desorption

Nucleation

Decomposition reactions

R + R

R +

Difusion

Heated substrate
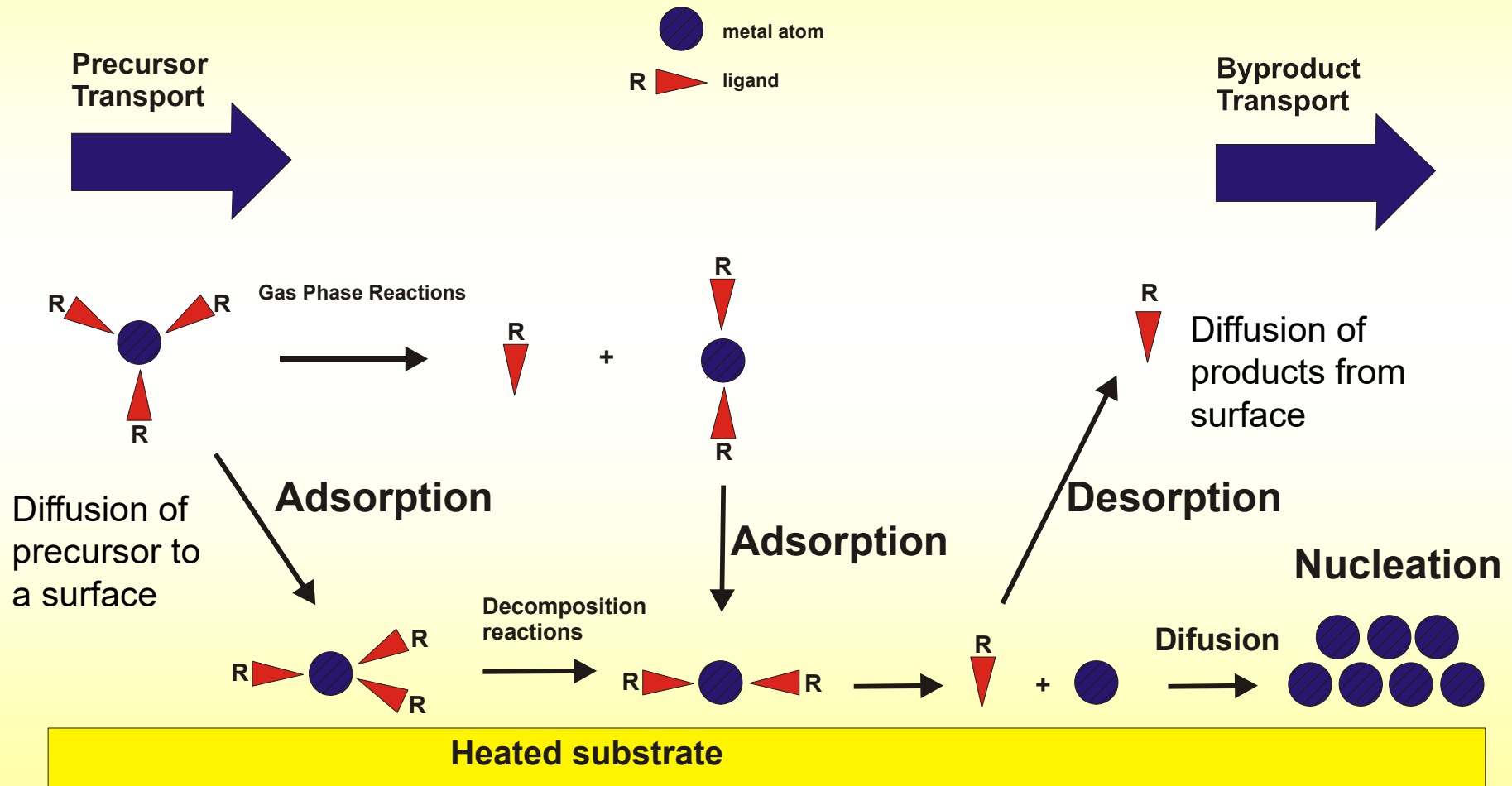
Cold-Wall CVD Reactor

Hot-Wall CVD Reactor
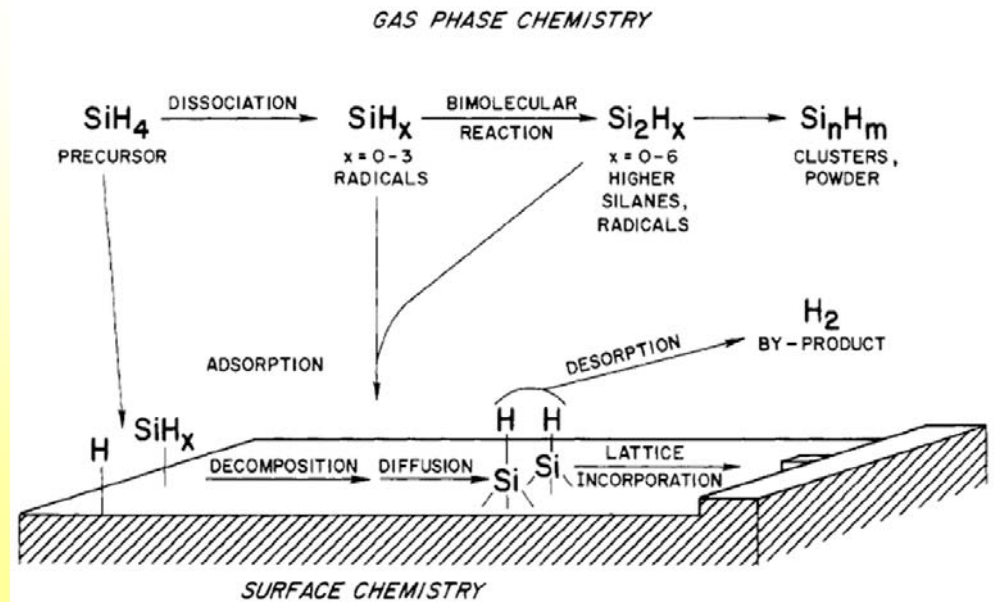

Gas Supply System   Reactor   Vacuum System
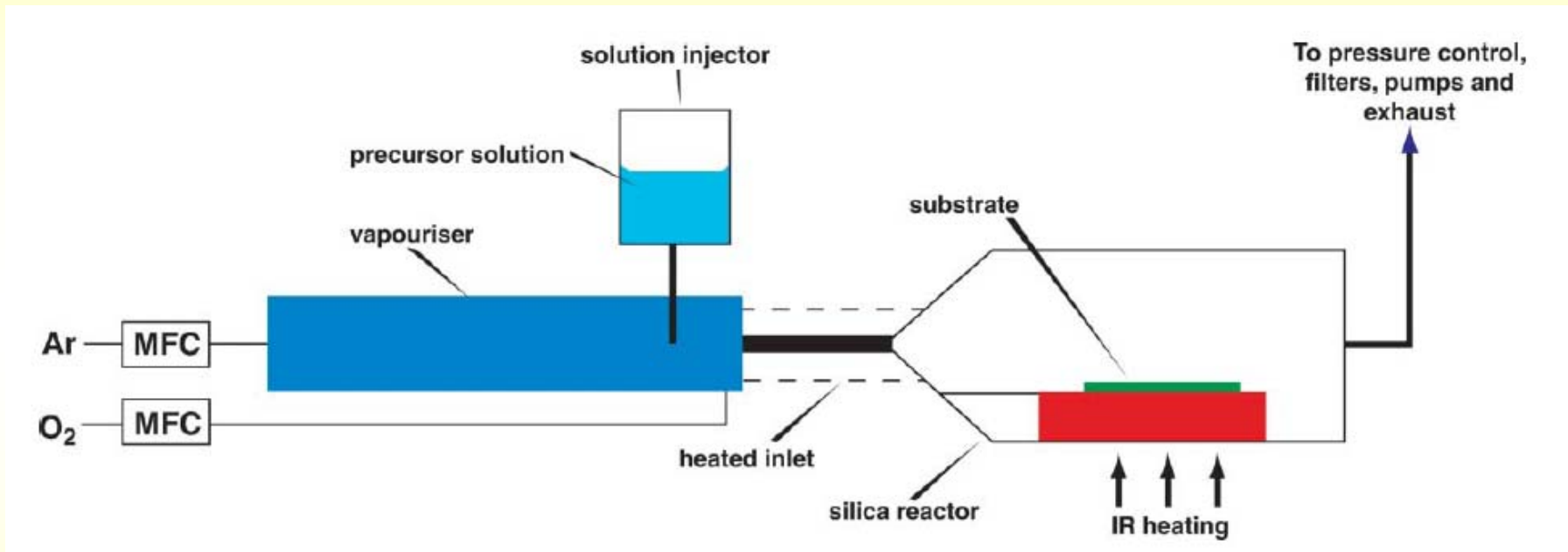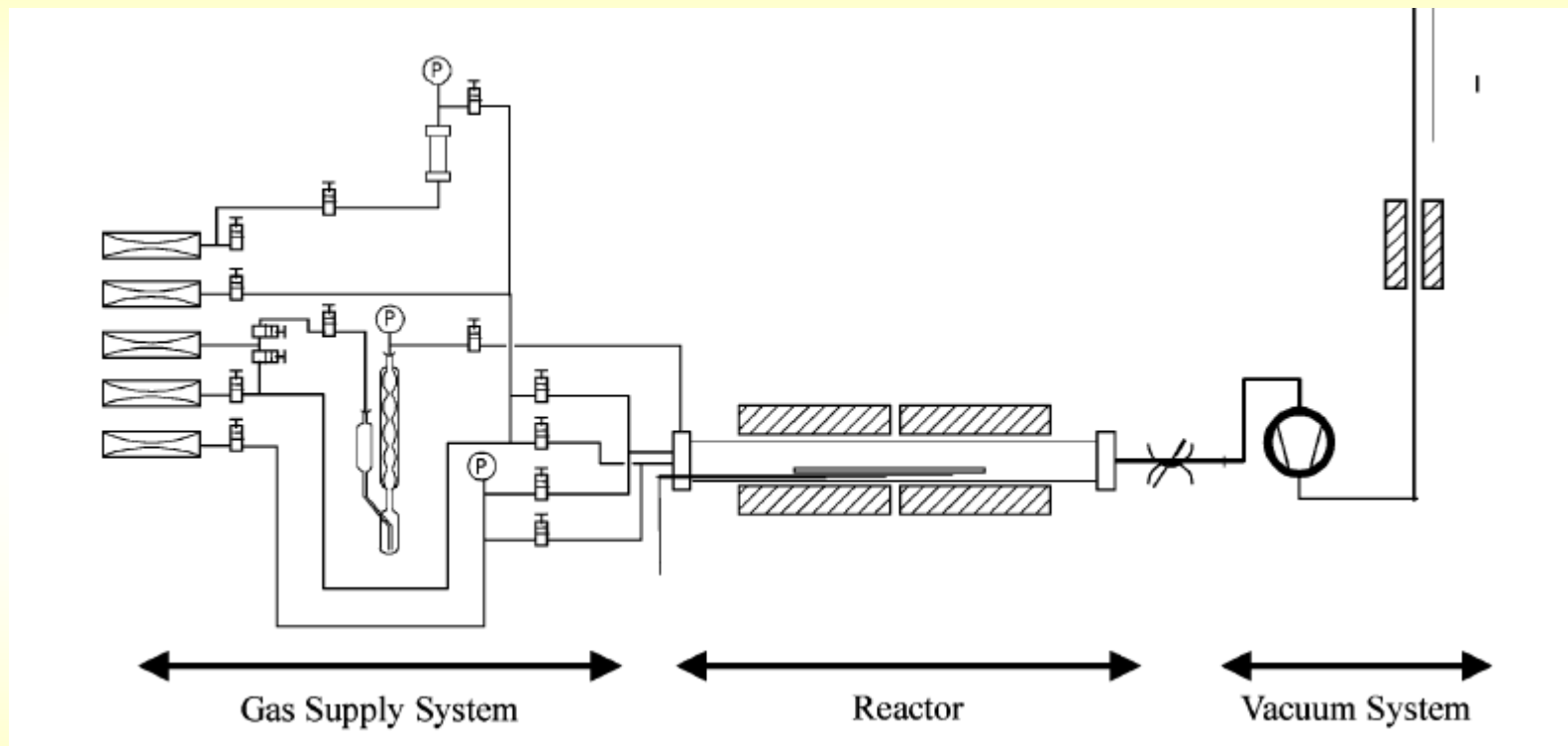
CVD Kinetics

Deposition depends on the sequence of events:
(1) Diffusion of precursor to surface
(2) Adsorption of precursor at surface
(3) Chemical reaction at surface
(4) Desorption of byproducts from surface
(5) Diffusion of byproducts from surface

The *slowest* event will be the rate-determining step

GAS PHASE CHEMISTRY

SiH_4 PRECURSOR — DISSOCIATION → SiH_x x = 0-3 RADICALS — BIMOLECULAR REACTION → Si_2H_x x = 0-6 HIGHER SILANES, RADICALS → Si_nH_m CLUSTERS, POWDER

ADSORPTION

DESORPTION → H_2 BY-PRODUCT

H SiH_x H H

DECOMPOSITION DIFFUSION → Si Si LATTICE INCORPORATION

SURFACE CHEMISTRY

CVD Kinetics

Growth Rate Model

F$_1$ = precursor flux from bulk of gas to substrate surface

F$_1$ = h$_G$ · (C$_G$ - C$_S$)

h$_G$ = mass-transfer coefficient   h$_G$ = D / δ
D = gas diffusion constant   D = D$_o$ T$^{3/2}$ / P
δ = boundary layer thickness (related to gas velocity)
C$_G$, C$_S$ = precursor conc. at bulk of gas and at substrate surface
(conc. gradient – driving force for diffusion)
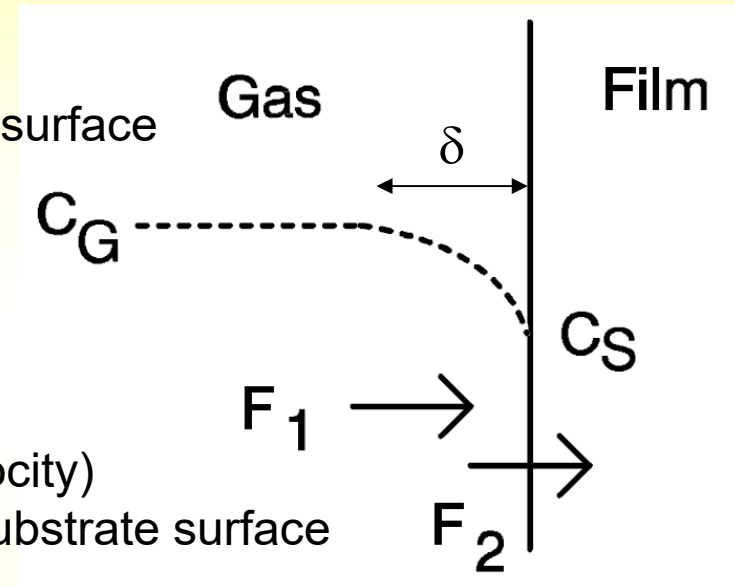
F$_2$ = flux consumed in film-growth reaction (rate of chemical reaction)

F$_2$ = k$_S$ · C$_S$

k$_S$ = surface-reaction rate constant: k$_S$ = A exp ($-$E$_a$/kT)

Steady state
F$_1$ = F$_2$ = F

CVD Kinetics

Steady state
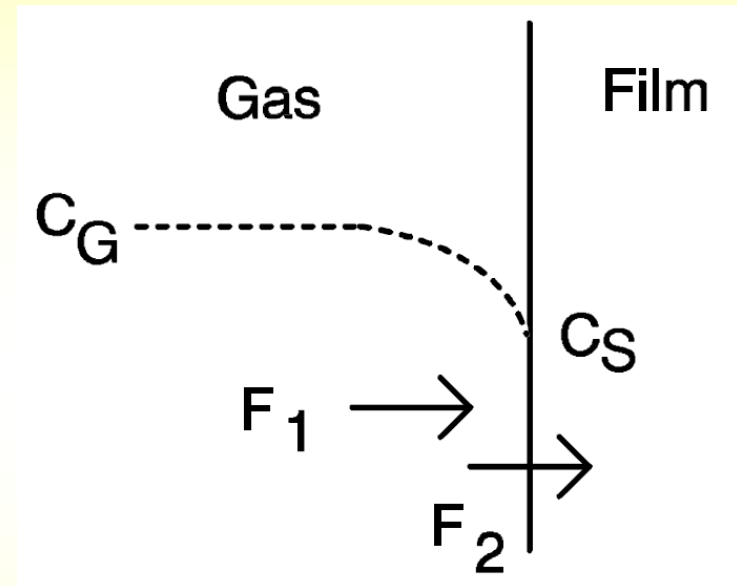**$F_1 = F_2 = F$**

Growth Rate Model

$F_1 = F_2$ **(rate of transport = rate of reaction)**

$h_G \cdot (C_G - C_S) = k_S \cdot C_S$

$C_S = C_G / (1 + k_S/h_G)$

$F = k_S\, h_G\, C_G / (k_S + h_G)$

Growth rate (thickness growth rate)

$dy / dt = F / \rho$

y = film thickness
ρ = atomic density of film

$$\frac{dy}{dt} = C_G \frac{1}{\rho} \frac{1}{\dfrac{1}{k_S} + \dfrac{1}{h_G}}$$

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Growth Rate

Growth rate is determined by:

$$\frac{dy}{dt} = C_G \frac{1}{\rho} \frac{1}{\dfrac{1}{k_S} + \dfrac{1}{h_G}}$$

a) Concentration C_G of a precursor in bulk of gas mixture

b) By the smaller of h_G and k_S

$k_S \ll h_G$ = Surface reaction limited $dy/dt \sim \exp(-E_a/kT)$

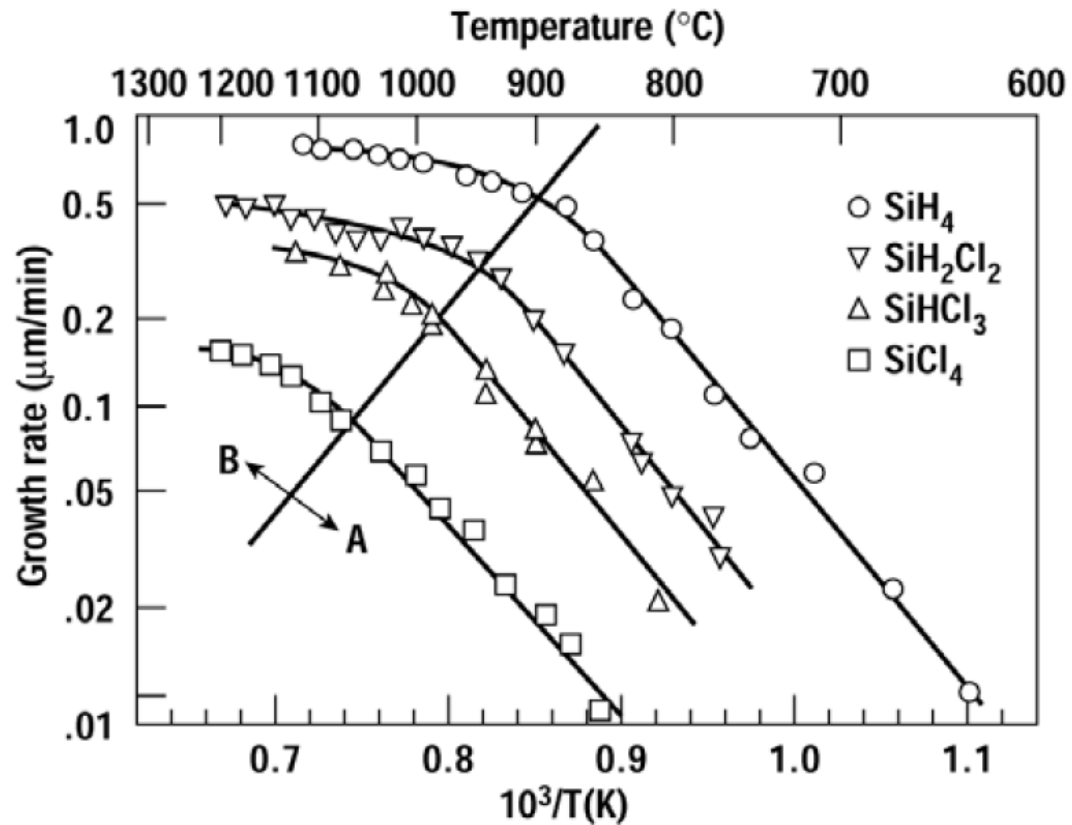$h_G \ll k_S$ = Mass transport limited $dy/dt \sim T^{3/2}$

When temperature is low, surface reaction rate is slow, and excess of reactants is available = the reaction is **surface reaction limited**

Above a certain temperature all source gas molecules react immediately = the reaction is then in **mass-transport limited** regime (also diffusion limited and supply limited regime)

Deposition rate vs. Temperature

[log scale] Rate

$\ln (dy / dt)$

Mass transport limited $\mathbf{R} \propto \mathbf{T}^{3/2}$

$$\frac{dy}{dt} = C_G \frac{1}{\rho} \frac{1}{\dfrac{1}{k_S} + \dfrac{1}{h_G}}$$

surface-reaction limited

$\mathbf{R} \propto \mathbf{e}^{-E_A/kT}$

0

$1/T$

high T ⟵   ⟶ low T

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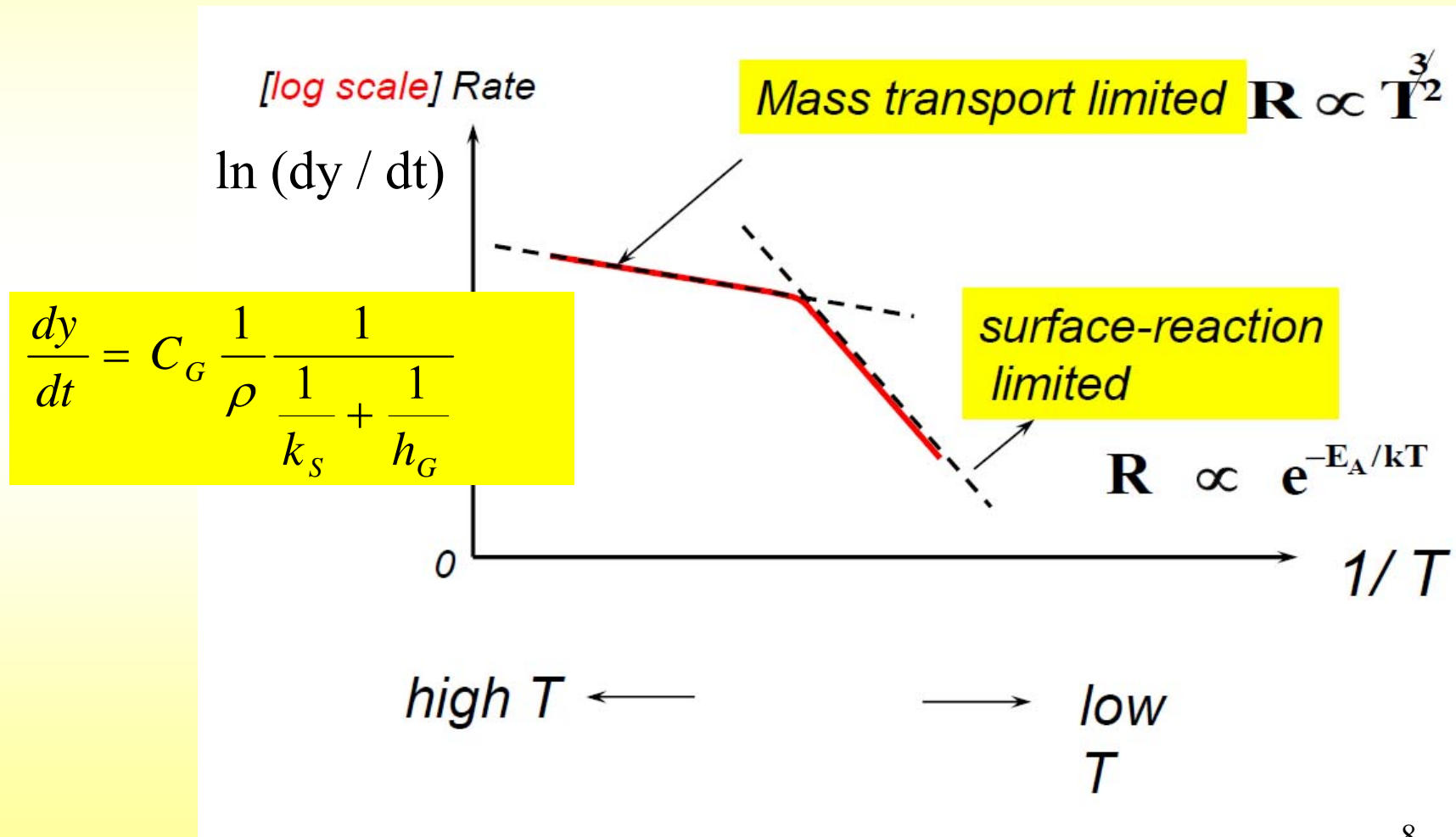
Deposition rate vs. Temperature

A = Surface reaction limited
B = Mass transport limited

Growth Rate Dependence on Flow Velocity

$$\delta_s(x) = \left(\frac{\mu x}{\rho U}\right)^{\frac{1}{2}}$$

μ = viscosity, ρ = density, U = velocity

$F_1 = h_G \cdot (C_G - C_S)$

h_G = mass-transfer coefficient

$h_G = D / \delta$
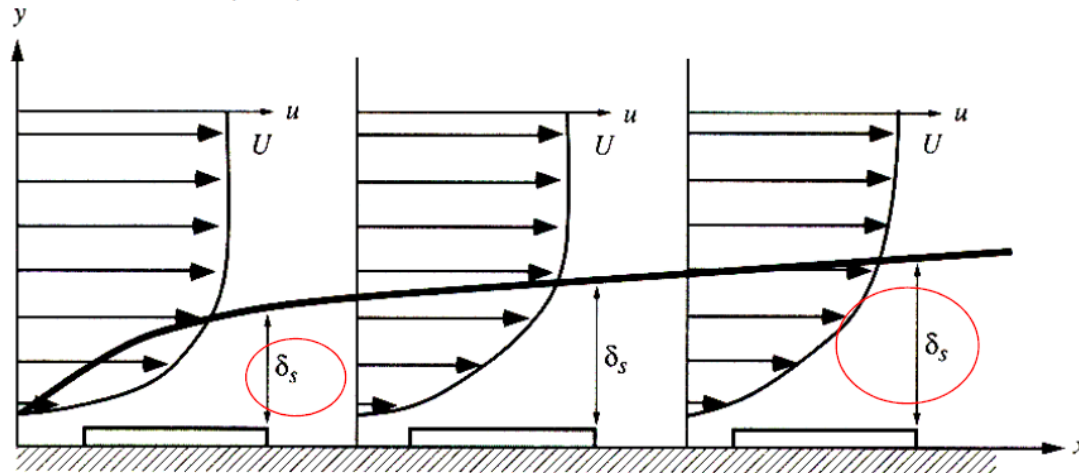
δ = boundary layer thickness

At constant T

Growth Rate dy/dt

mass transport limited

surface reaction limited

(Gas Flow Rate, U)$^{1/2}$

Low flow rate U
   ~ large boundary layer thickness δ
        ~ slow mass-transfer

10
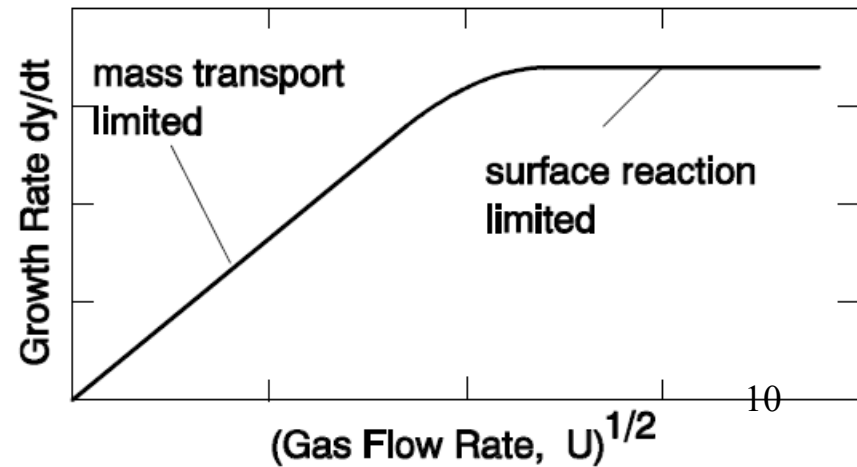
Precursor Volatility

Thermogravimetric Analysis
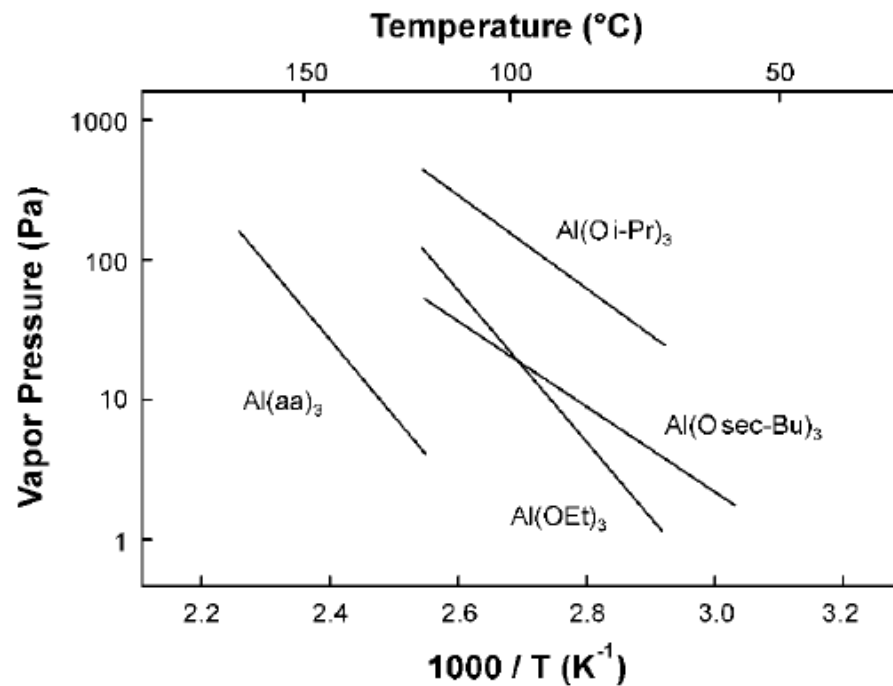
Clausius-Clapeyron Equation

$$\ln \frac{p_2}{p_1} = \frac{-\Delta H^0_{subl}}{R}\left(\frac{1}{T_2} - \frac{1}{T_1}\right)$$
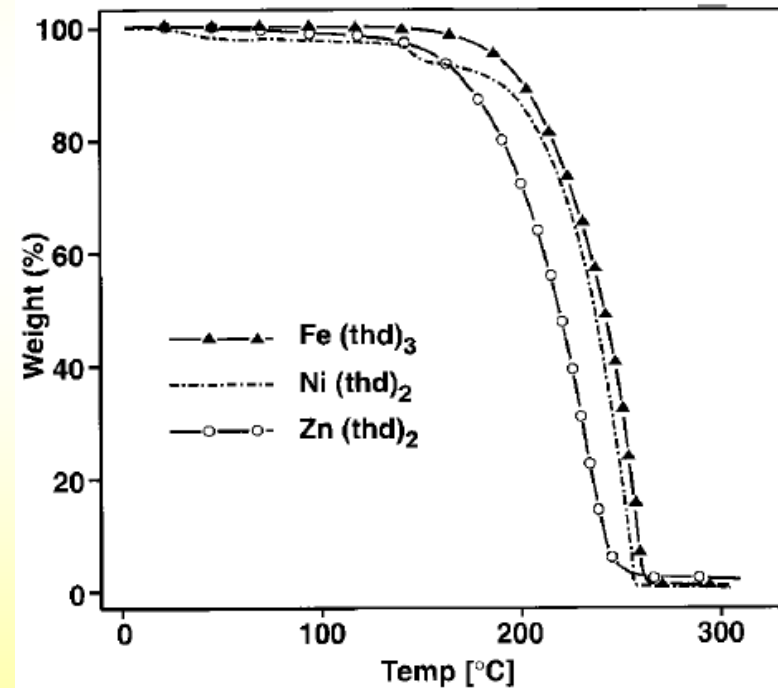
CVD of Al

Aluminum

2.27 $\mu\Omega$ cm, easily etched, Al dissolves in Si

GaAs + Al $\rightarrow$ AlAs + Ga
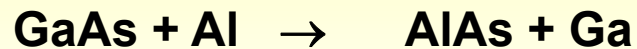
Gas diffusion barriers, Al on polypropylene, food packaging = chip bags, party balloons, high optical reflectivity = mirrors

TIBA = Triisobutylaluminum

β-Hydride Elimination

Clean deposition

below 330 ºC

β-Methyl Elimination

Carbon impurities

above 330 ºC

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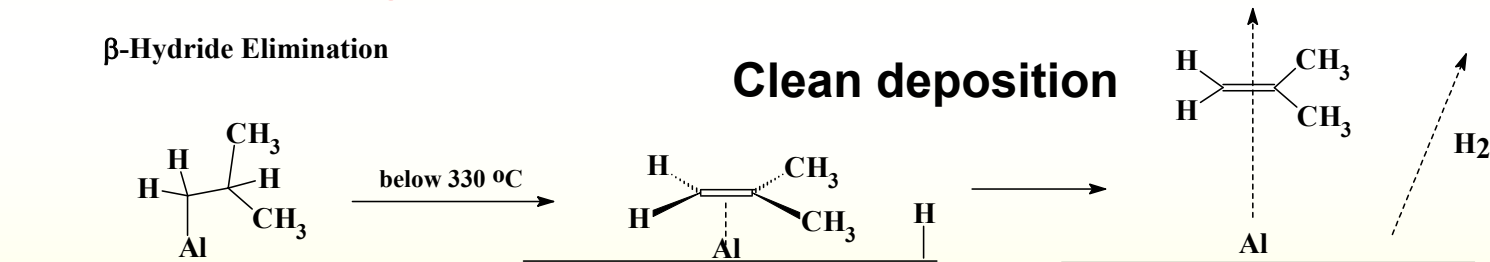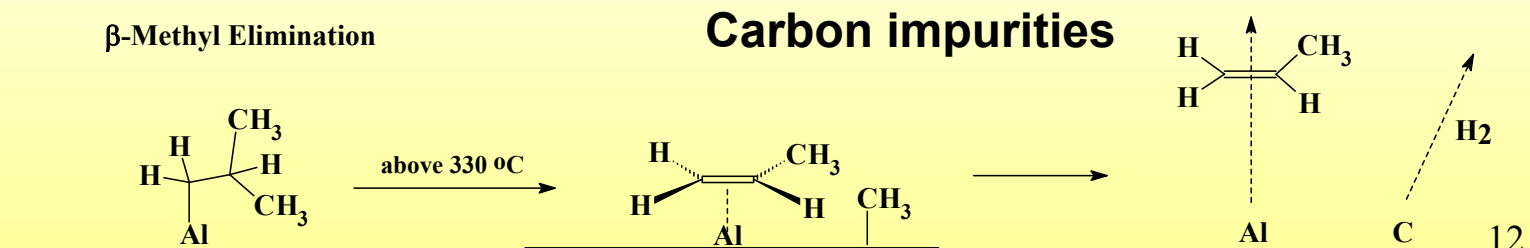
CVD of Al

Al deposits selectively on Al surfaces, not on SiO_2

Laser-induced nucleation
248 nm only surface adsorbates pyrolysed
193 nm gas phase reactions, loss of spatial selectivity control

TMA = Trimethylaluminum
Large carbon incorporation, Al_4C_3, RF plasma, laser

$$Al_2(CH_3)_6 \rightarrow \quad 1/2 \; Al_4C_3 \; + \; 9/2 \; CH_4 \quad under \; N_2$$

$$Al_2(CH_3)_6 \; + \; 3 \; H_2 \quad \rightarrow \quad 2 \; Al \; + \; 6 \; CH_4 \quad under \; H_2$$
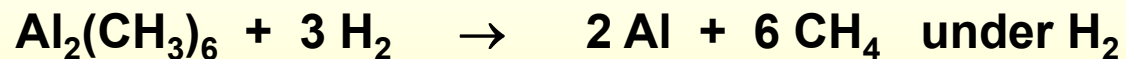
DMAH = Dimethylaluminum hydride
Ligand redistribution
$$[(CH_3)_2AlH]_3 \quad \rightarrow \quad (CH_3)_3Al \; + \; AlH_3$$

$$AlH_3 \rightarrow \; Al \; + \; H_2 \qquad at \; 280 \; °C, \; low \; carbon \; incorporation$$

13
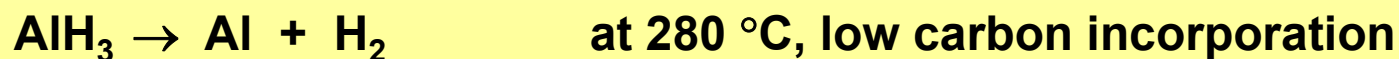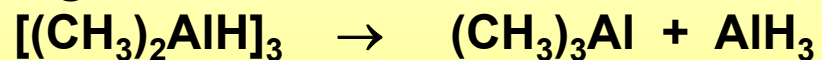
CVD of Al

TMAA = Trimethylammine-alane

Decomposition mechanism of TMAA on Al – below 100 °C

$$(CH_3)_3N\text{-}AlH_3 \quad \rightarrow \quad Al \quad + \quad (CH_3)_3N \quad + \quad 3/2\ H_2$$
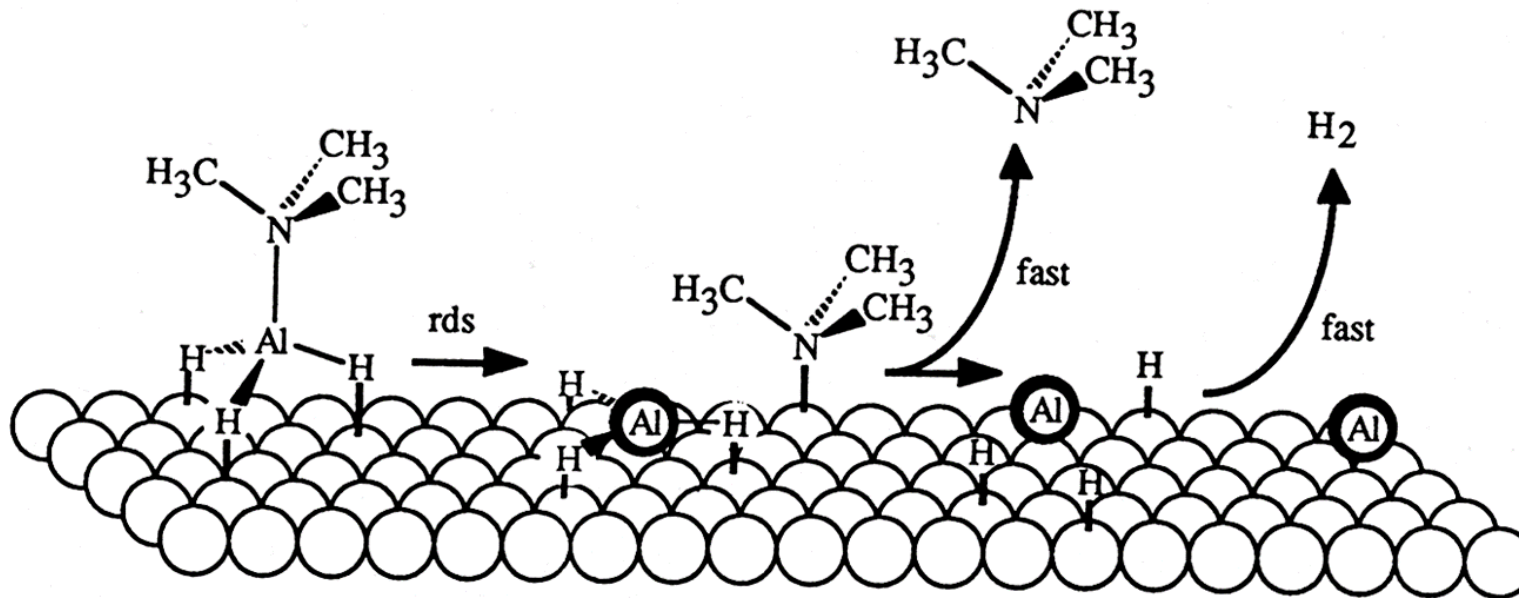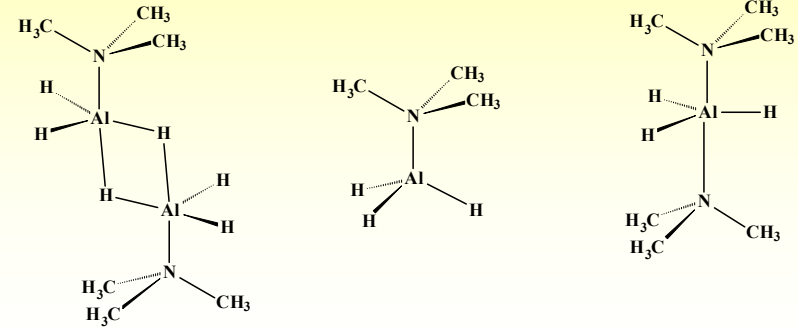
CVD of Al

Aluminoboranes

$(CH_3)_3N\text{-}BH_3$ + Al + 3/2 H_2
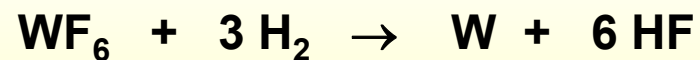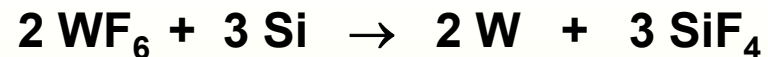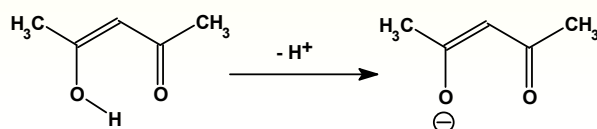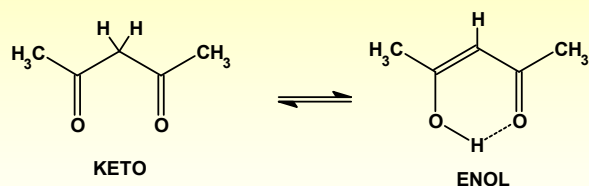
3/2 B_2H_6 + Al + 3/2 H_2

15
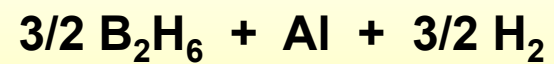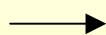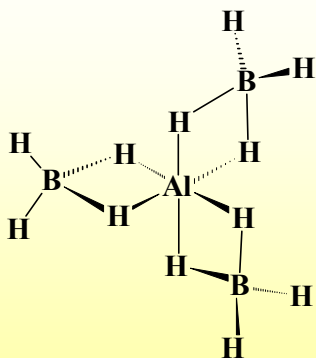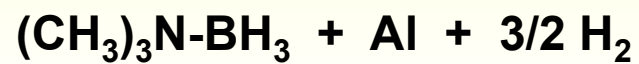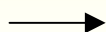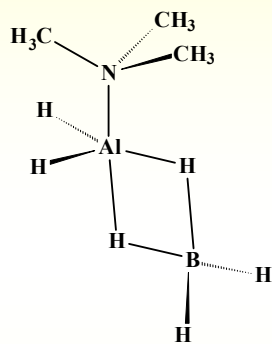
CVD of W

Tungsten

5.6 $\mu\Omega$ cm, a high resistance to electromigration, the highest mp of all metals 3410 °C

Tungsten hexafluoride (WF_6) is the heaviest known gas at room temperature and pressure, density = 13 g/L
**Purity 99.999%, extremely corrosive and toxic**

$$2\ WF_6 + 3\ Si \rightarrow 2\ W + 3\ SiF_4$$

$$WF_6 + 3\ H_2 \rightarrow W + 6\ HF$$

$$WF_6 + 3/2\ SiH_4 \rightarrow W + 3\ H_2 + 3/2\ SiF_4$$

$$W(CO)_6 \rightarrow W + 6\ CO$$

Tungsten hexafluoride

Diketonate Precursors

KETO ⇌ ENOL

Mononuclear

R$_1$	R$_2$	Name	Abbreviation
CH$_3$	CH$_3$	Pentane-2,4-dionate (acetylacetonate)	acac
CH$_3$	CF$_3$	1,1,1-trifluoropentane-2,4-dionate (trifluoroacetylacetonate)	tfac
CF$_3$	CF$_3$	1,1,1,5,5,5-hexafluoropentane-2,4-dionate (hexafluoroacetylacetonate)	hfac
CH$_3$	C(CH$_3$)$_3$	1,1-dimethylhexane-3,5-dionate	dhd
C(CH$_3$)$_3$	C(CH$_3$)$_3$	2,2,6,6-tetramethylheptane-3,5-dionate	thd
CH$_3$	CH$_2$CH(CH$_3$)$_2$	6-methylheptane-2,4-dionate	mhd
C(CH$_3$)$_3$	CH$_2$CH(CH$_3$)$_2$	2,2,7-trimethyloctane-3,5-dionate	tmod
C$_6$H$_5$	C$_6$H$_5$	1,3-diphenylpropane-1,3-dionate (dibenyzoylmethanate)	dbm

Polynuclear

17
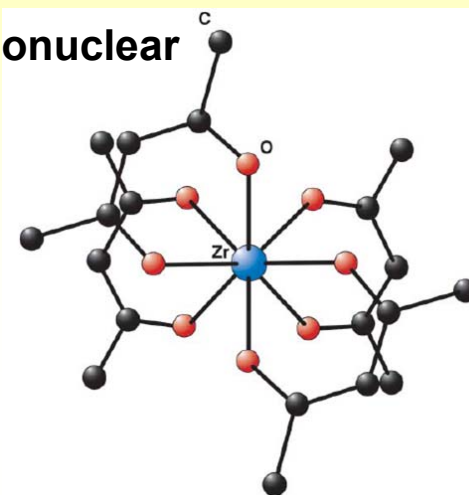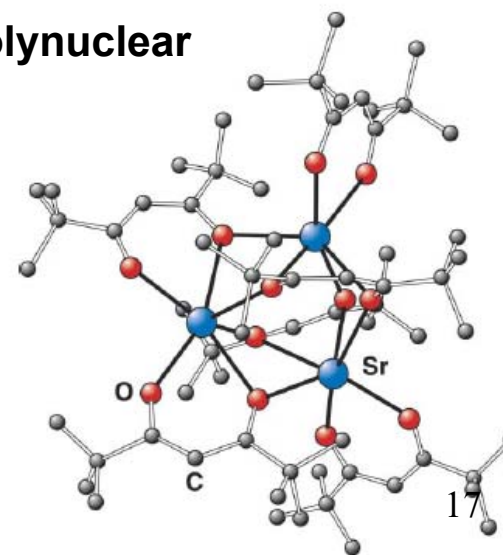
CVD of Cu

Copper(II) hexafluoroacetylacetonate

Excellent volatility
CF$_3$ groups (low polarizibility)
Vapor pressure of 0.06 Torr at r. t.
Low decomposition temperature
Stability in air, low toxicity
Commercial availability
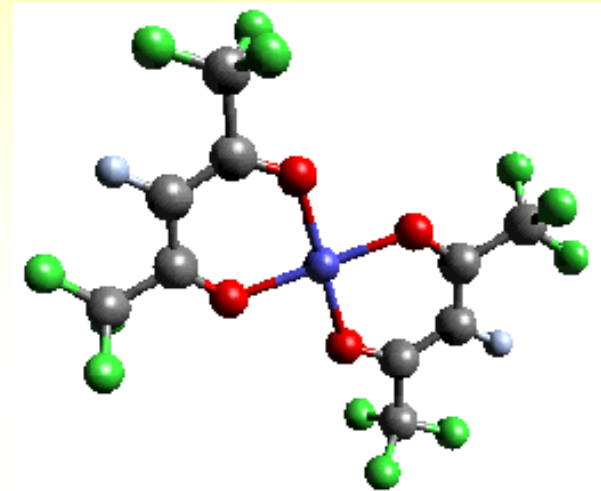
Deposition on metal surfaces (Cu, Ag, Ta)
The first step can already occur at -150 °C
= a dissociation of the precursor molecules on the surface

An electron transfer from a metal substrate to the single occupied HOMO, which has an anti-bonding character with respect to copper d$_{xy}$ and oxygen p orbitals, weakens the Cu-O bonds and facilitates their fission
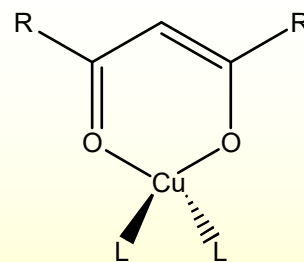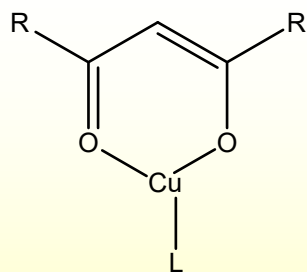
CVD of Cu

Copper(II) hexafluoroacetylacetonate

Scheme I

19
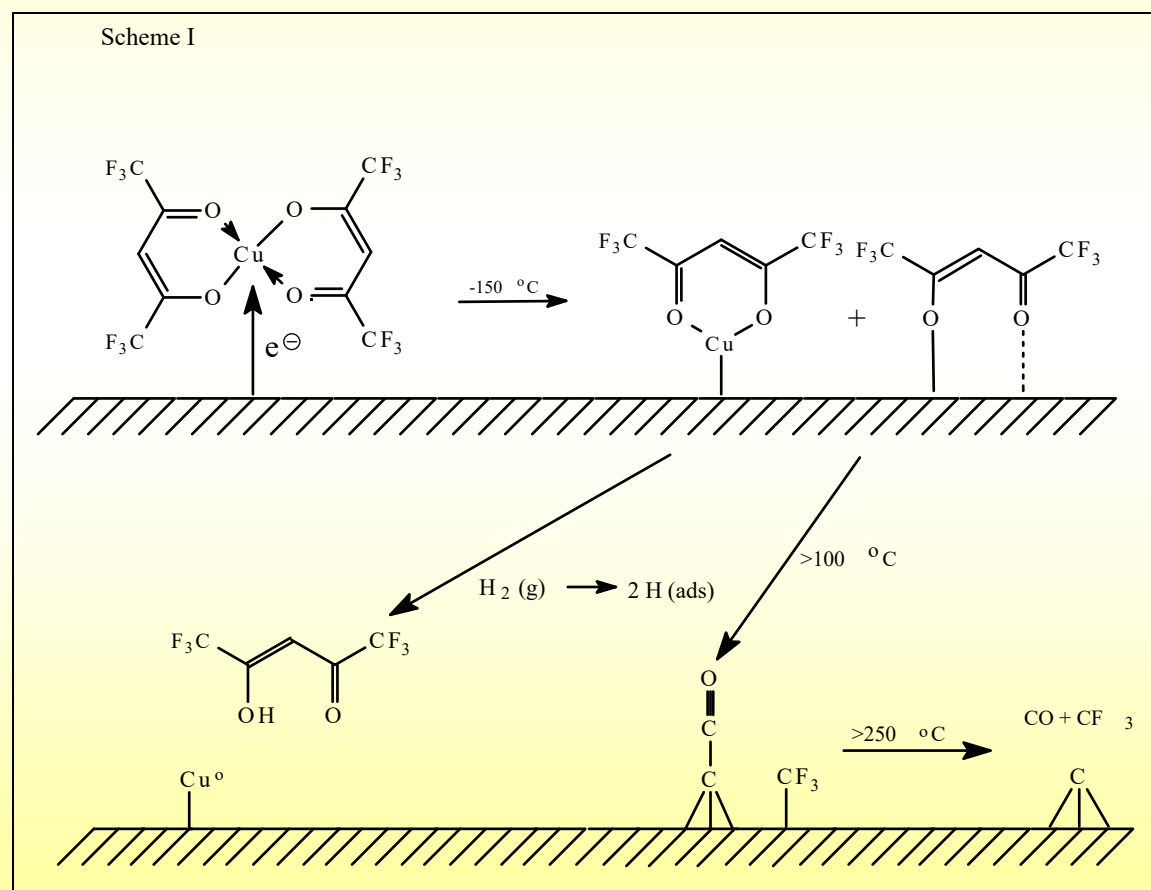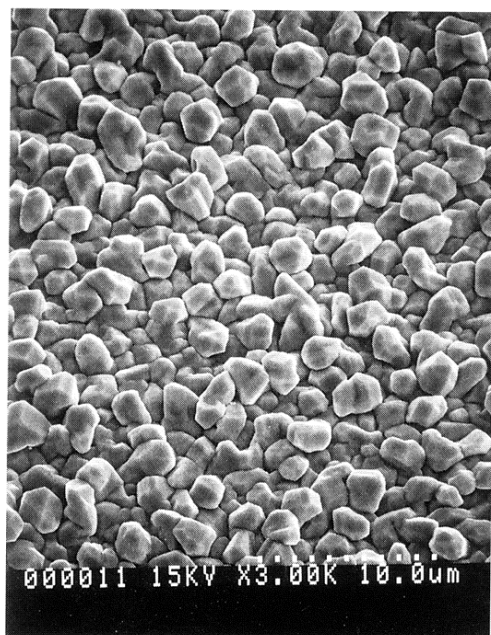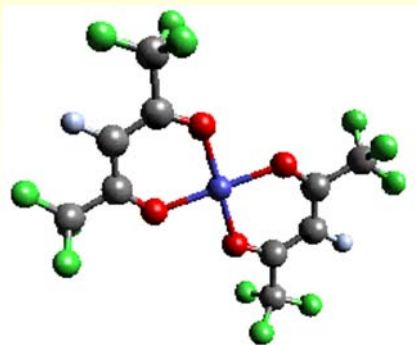
CVD of Cu

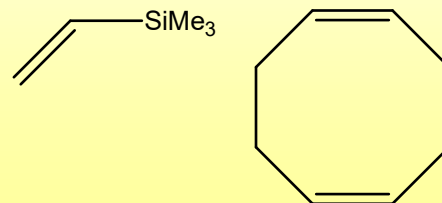**Growth rate of Cu films deposited from Cu(hfacac)$_2$ with 10 torr of H$_2$**
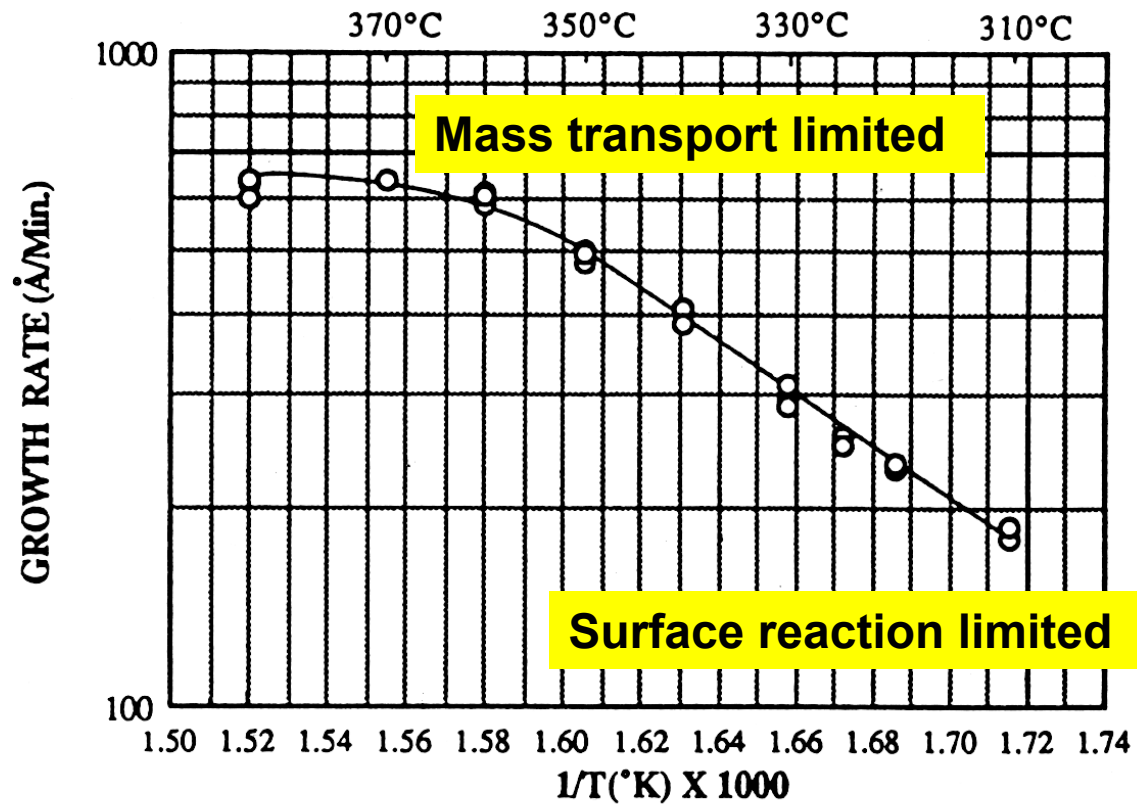
CVD of Cu

Cu(I) Precursors

Disproportionation to Cu(0) and Cu(II)

2 Cu(diketonate)L$_n$ $\rightarrow$ **Cu + Cu(diketonate)$_2$ + n L**

L: PMe$_3$, PEt$_3$, CO, CN^{t}Bu,

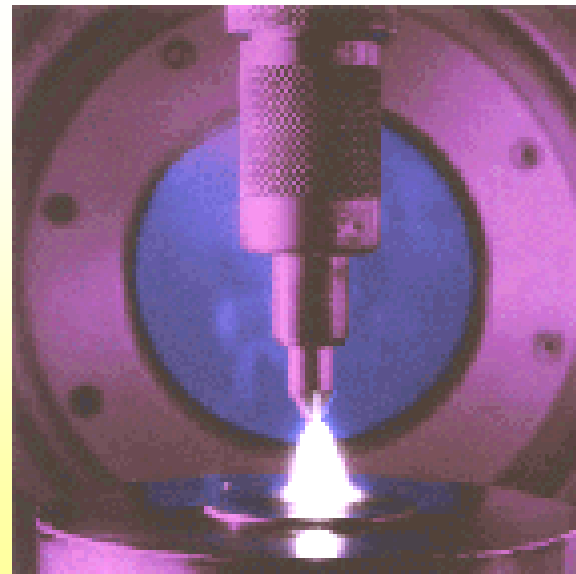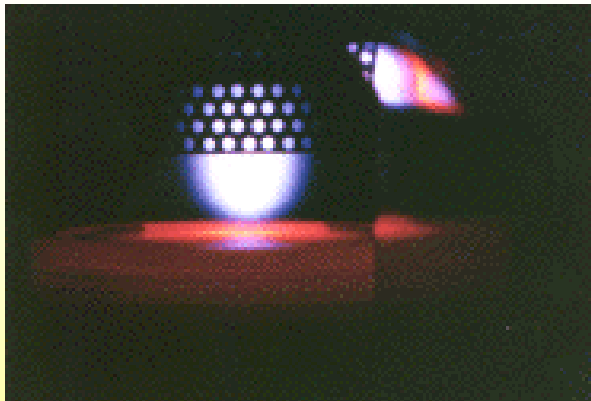# CVD of YF$_3$ from Y(hfacac)$_3$ Complex

22
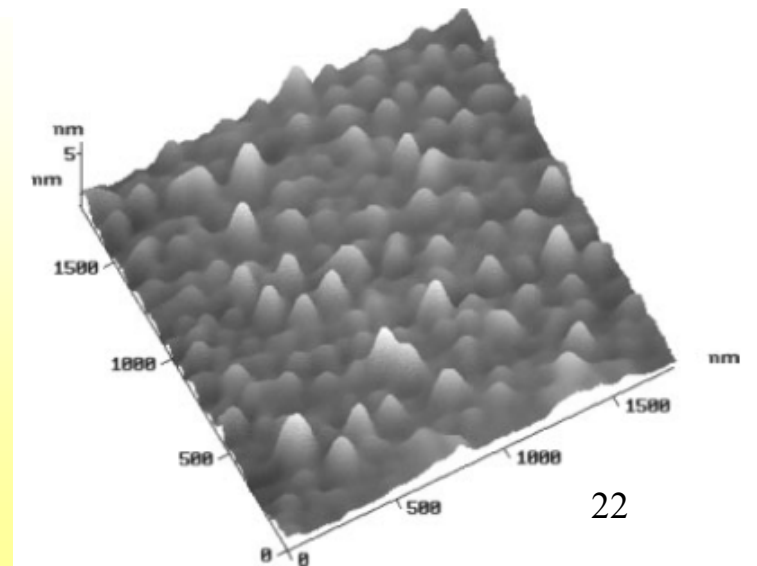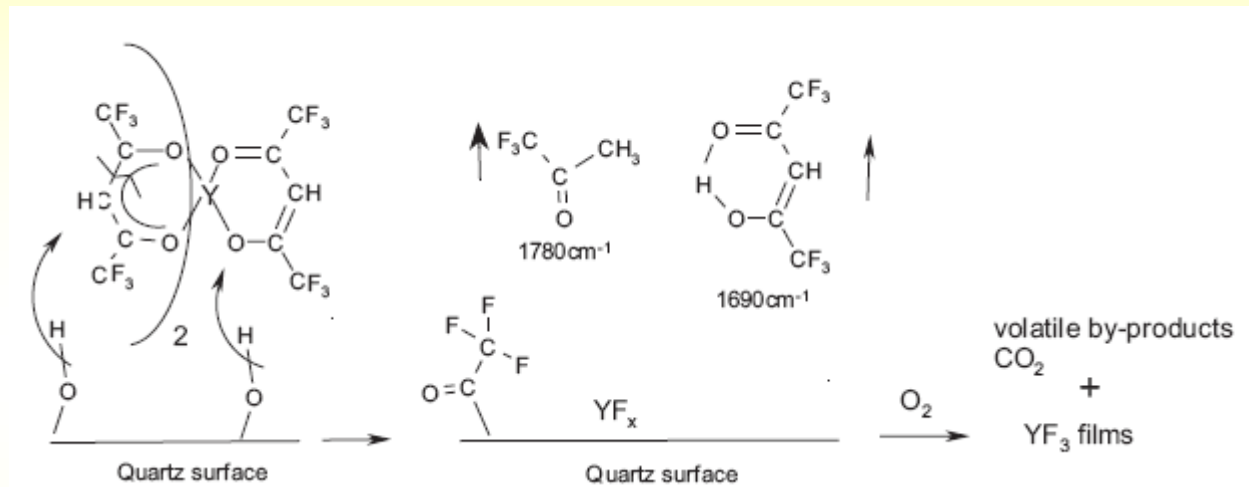
CVD of Diamond

Diamond films

Activating gas-phase carbon-containing precursor molecules:

- **Thermal (*e.g.*, hot filament)**

- **Plasma (D.C., R.F., or microwave)**

- **Combustion flame (oxyacetylene or plasma torches)**

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CVD of Diamond

Experimental conditions:

Temperature 1000 - 1400 K

The precursor gas diluted in an excess of H_2

Typical CH_4 mixing ratio ~1-2 vol%

Deposited films are polycrystalline

Film quality:

- **the ratio of sp^3 (diamond) to sp^2-bonded (graphite) carbon**

- **the composition (*e.g.*, C-C versus C-H bond content)**

- **the crystallinity**

Combustion methods: high rates (100-1000 µm/hr), small, localised areas, poor quality films

Hot filament and plasma methods: slower growth rates (0.1-10 µm/hr), high quality films

CVD of Diamond

Hydrogen atoms generated by activation (thermally or via electron bombardment)
H-atoms play a number of crucial roles in the CVD process:

H abstraction reactions with hydrocarbons, highly reactive radicals: CH_3
(stable hydrocarbon molecules do not react to cause diamond growth)
Radicals diffuse to the substrate surface and form C-C bonds to propagate the diamond lattice

H-atoms terminate the 'dangling' carbon bonds on the growing diamond surface - prevent cross-linking and reconstructing to a graphite-like surface

Atomic hydrogen etches both diamond and graphite - under CVD conditions, the rate of diamond growth exceeds its etch rate whilst for graphite the converse is true

This is the basis for the preferential deposition of diamond rather than graphite

CVD of Diamond

Diamond initially nucleates as individual microcrystals,

which then grow larger until they coalesce into a continuous film

Diamond laser window
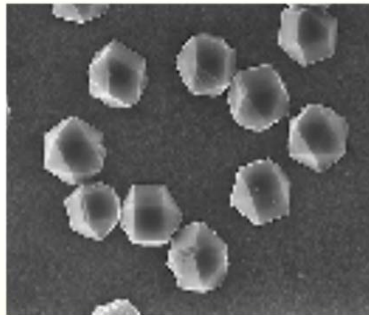
Enhanced nucleation by ion bombardment:

- Damage the surface - more nucleation sites

- Implant ions into the lattice

- Form a carbide interlayer - glue, promotes diamond growth, aids adhesion

CVD of Diamond

Substrates: metals, alloys, and pure elements:

Little or no C Solubility or Reaction: Cu, Sn, Pb, Ag, and Au, Ge, sapphire, diamond, graphite

C Diffusion: Pt, Pd, Rh, Fe, Ni, and Ti
The substrate acts as a carbon sink, deposited carbon dissolves into the metal surface, large amounts of C transported into the bulk, a temporary decrease in the surface C concentration, delaying the onset of nucleation

Carbide Formation: Ti, Zr, Hf, V, Nb, Ta, Cr, Mo, W, Fe, Co, Ni, Y, Al
B, Si, SiO_2, quartz, Si_3N_4 also form carbide layers
SiC, WC, and TiC

Applications of Diamond Films

Thermal management - **a heat sink for laser diodes, microwave integrated circuits, active devices mounted on diamond can be packed more tightly without overheating**

Cutting tools - **an abrasive, a coating on cutting tool inserts**
CVD diamond-coated tools have a longer life, cut faster and provide a better finish than conventional WC tool bits

Wear Resistant Coatings - **protect mechanical parts, reduce lubrication gearboxes, engines, and transmissions**

Optics - **protective coatings for infrared optics in harsh environments**
ZnS, ZnSe, Ge: excellent IR transmission but brittle
the flatness of the surface, roughness causes attenuation and scattering of the IR signal

Electronic devices - **doping, an insulator converted into a semiconductor**
p-doping: B_2H_6 incorporates B into the lattice
doping with atoms larger than C very difficult, n-dopants such as P or As, cannot be used for diamond, alternative dopants, such as Li

CVD of ZnO
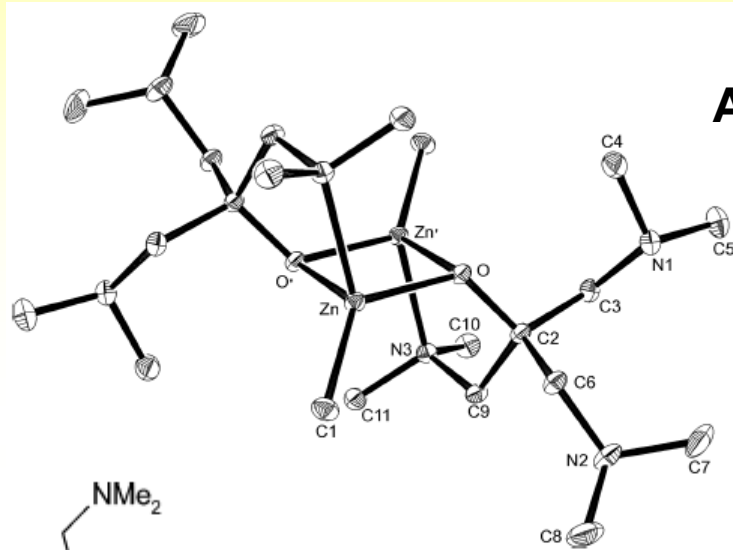
Aminoalcoholates

500 °C, 1 h

TG
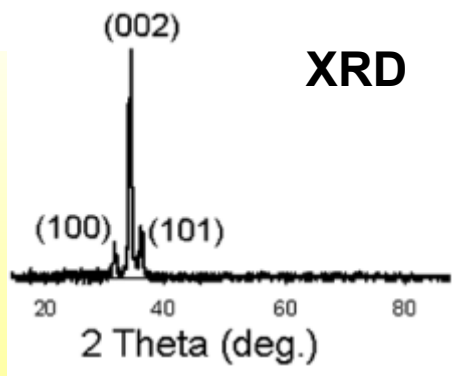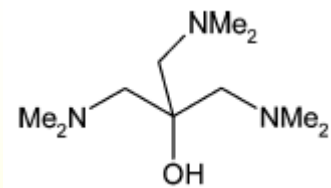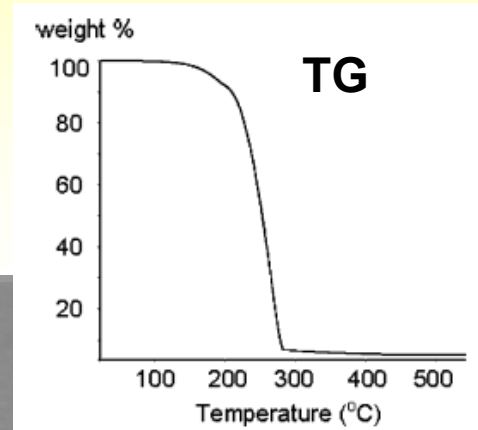
XRD

Hexagonal ZnO PDF 79-0208
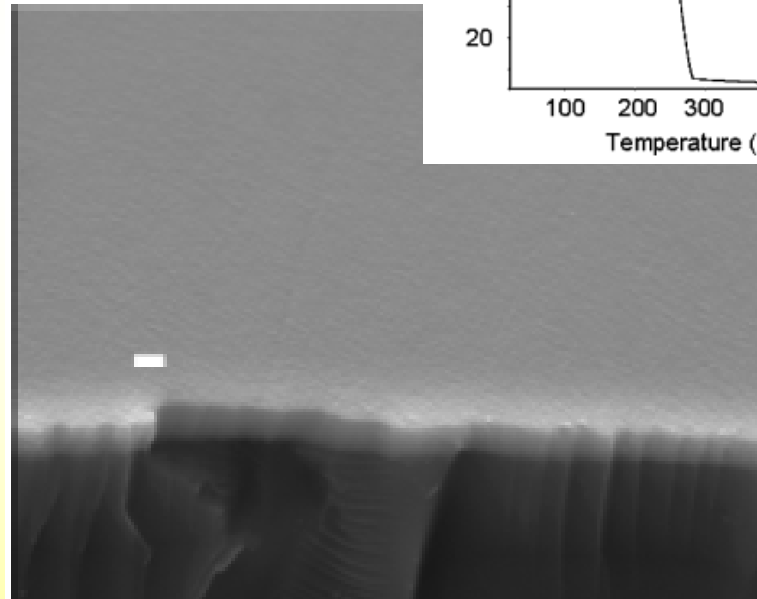
SEM of the ZnO film
Bar = 1 μm

Laser-Enhaced CVD

$$Si(O_2CCH_3)_4 \rightarrow SiO_2 + 2\ O(OCCH_3)_2$$
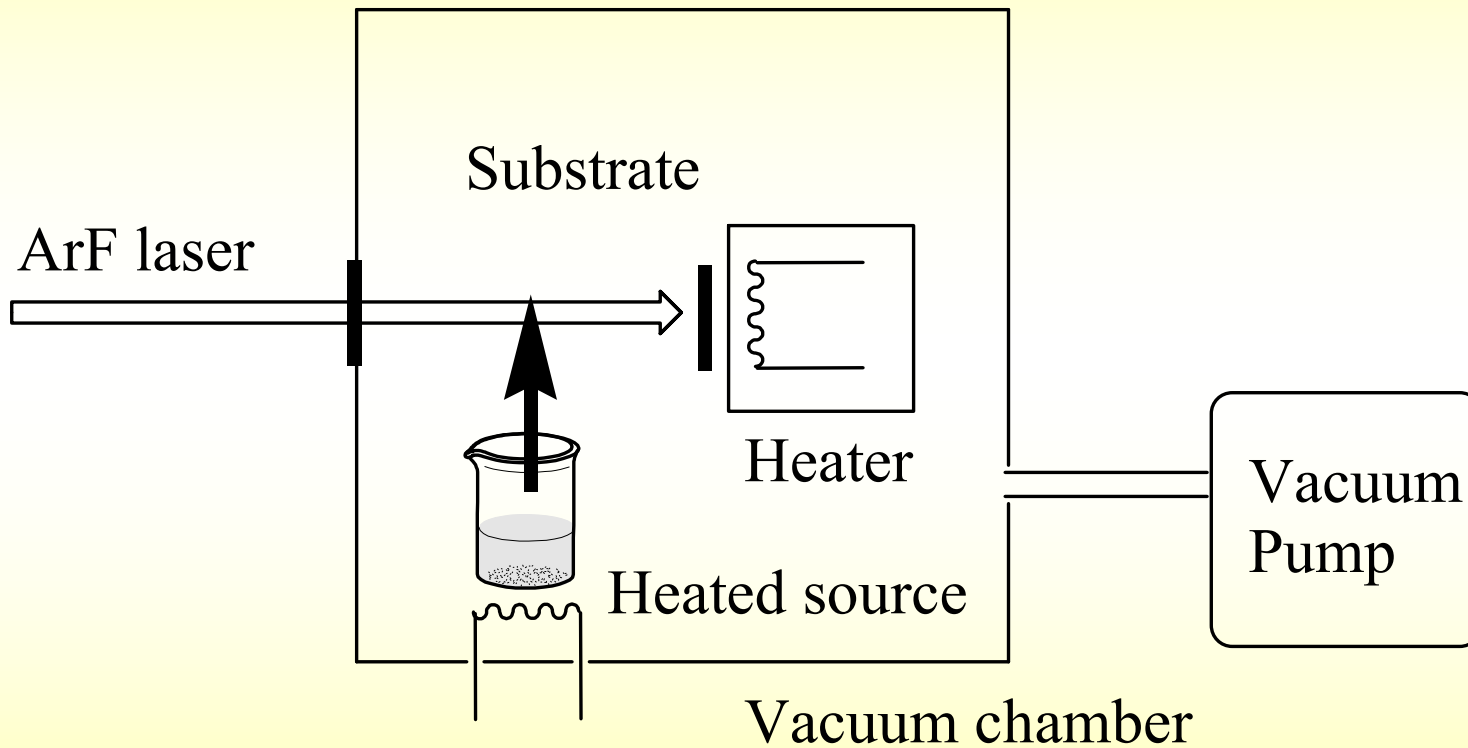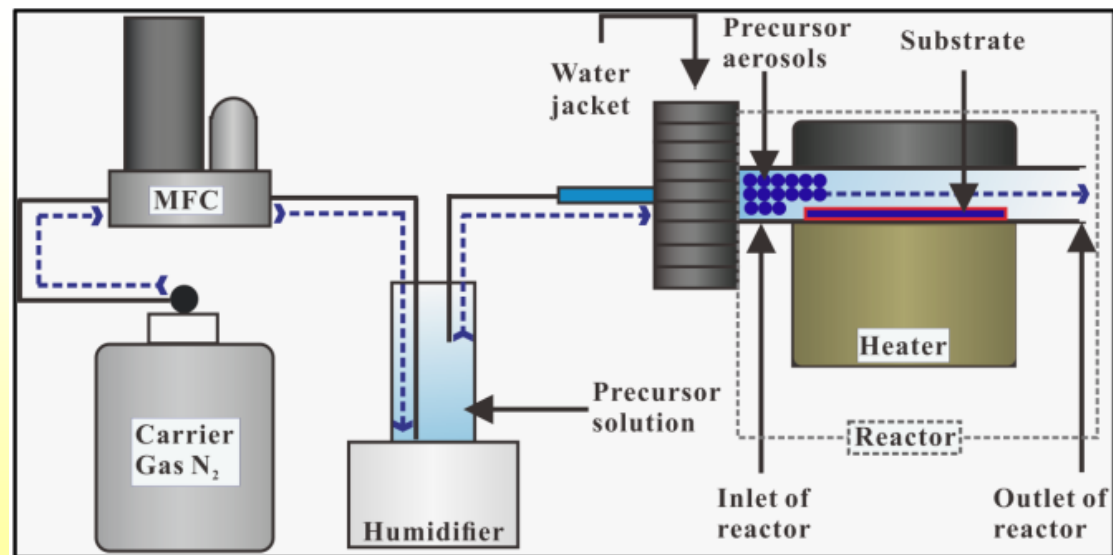
Aerosol-Assited CVD

Carried out at atmospheric pressure
Non-volatile precursors
Formation of droplets in ultrasonic nebulizer / atomizer
Precursor aerosol delivered by the stream of carrier gas to reactor
Solvent evaporation

Pulsed Injection MOCVD

Low-pressure hot-wall reactor - vacuum 10 Torr
A pulsed liquid injection system - electromagnetic injector
A metal-organic precursor dissolved in a solvent (DME)
Precise micro-doses (several μL, frequency 2 Hz)
A hot (200 °C) evaporation zone - flash evaporation of micro-doses
A mixture of precursor and solvent vapors carried into the deposition zone with a Ar:O_2 (4:1) gas mixture
Depositions on sapphire-C substrates at 350 - 900 °C

$Sn(OtBu)_2(PyTFP)_2$ (1)

$Sn(OtBu)_2(DMOTFP)_2$ (2)

$Sn(Bu)_2(acac)_2$
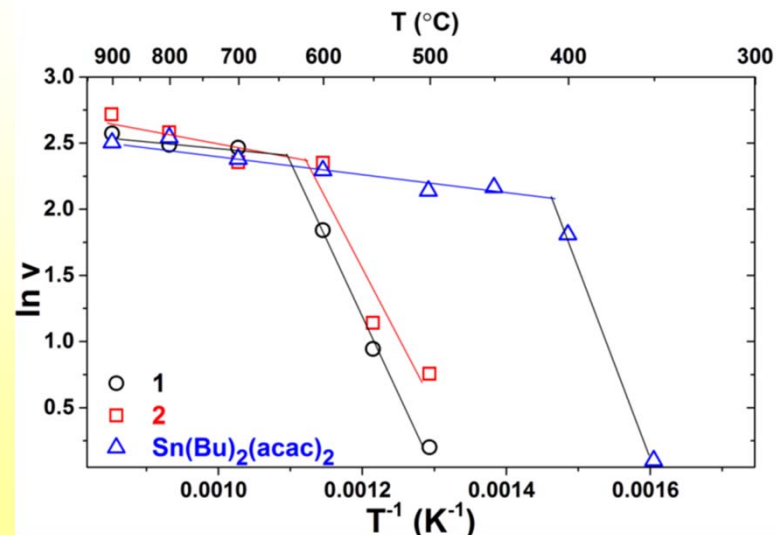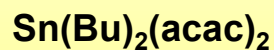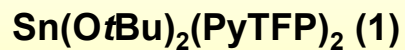
Atomic Layer Deposition ALD

A method for the deposition of thin films

**Film growth by cyclic process
4 steps:**

1) **Exposition by 1st precursor**
2) **Cleaning of the reaction chamber**
3) **Exposition by 2nd precursor**
4) **Cleaning of the reaction chamber**

(a) $Al(CH_3)_3$ pulse

(b) purge

(d) purge

(c) H_2O pulse
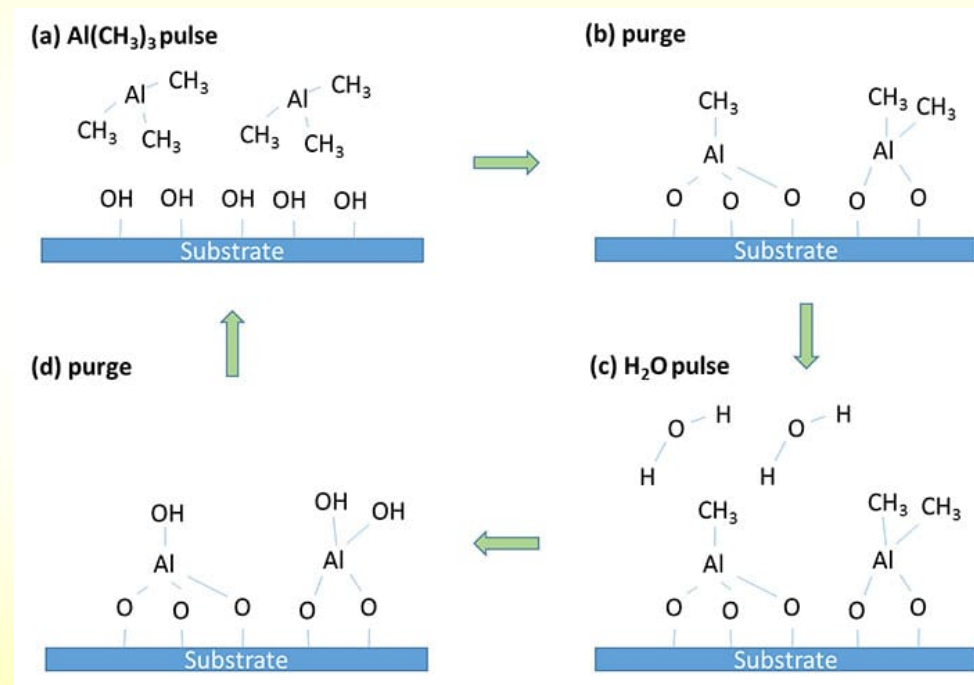
Cycle repetitions until desired film thickness is reached

1 cycle: 0.5 s – several sec.   thickness 0.1- 3 Å

Atomic Layer Deposition ALD

Requires high reactivity

Self-Limiting growth mechanism

Formation of a monolayer

Control of film thickness and composition

Carried out at room temperature

Reactor walls inactive – no reactive layer

Separate loading of reactive precursors –
    no gas-phase reactions

Deposition on large surface area

Highly conformal coverage of surface

Precursor transport to the reaction zone
    does not have to be highly uniform
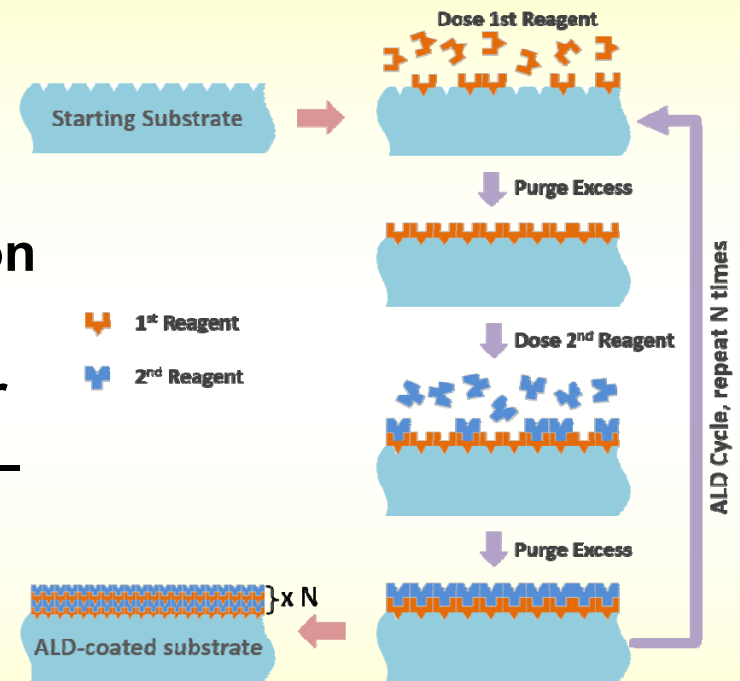
Comparison of ALD and CVD

ALD

- Highly reactive precursors
- Precursors react separately on the substrate
- Precursors must not decompose at process temperature
- Uniformity ensured by the saturation mechanism
- Thickness control by the number of reaction cycles
- Surplus precursor dosing acceptable

CVD

- Less reactive precursors
- Precursors react at the same time on the substrate
- Precursors can decompose at process temperature
- Uniformity requires uniform flux of reactant and T
- Thickness control by precise process control and monitoring of T, flow, time
- Precursor dosing important

ALD vs. CVD vs. PVD Comparison
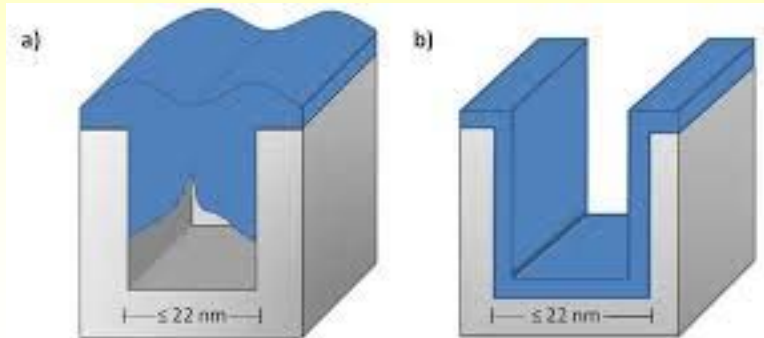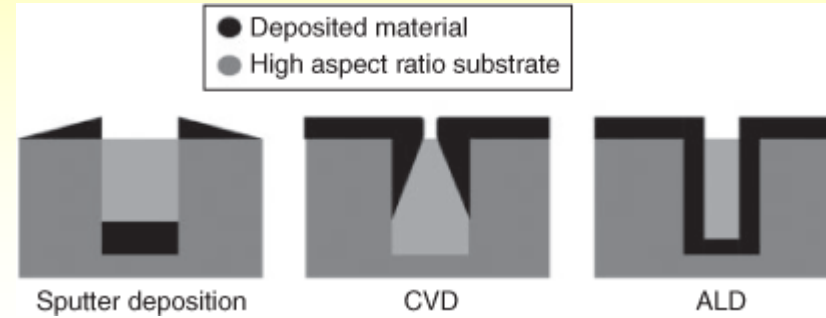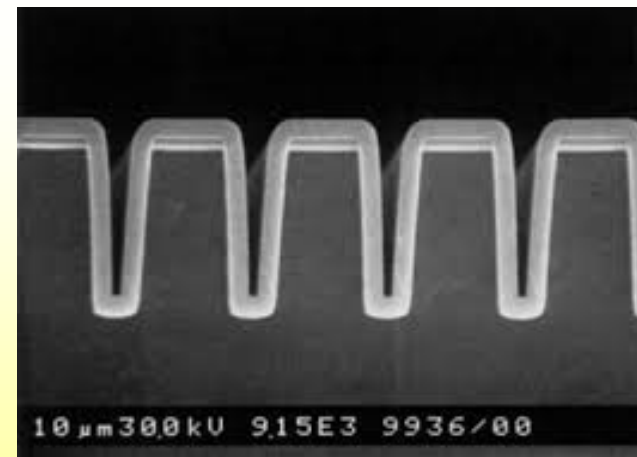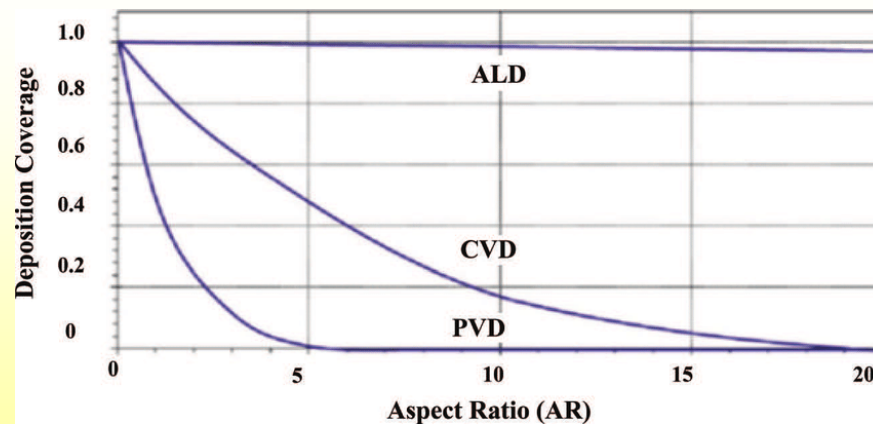
AR = the ratio of width to depth

ALD Precursor Properties

Selection of suitable combination of precursors
Molecular size influences film thickness
Gases, volatile liquids, solids with high vapor pressure

Typical precursors:

Metallic - halogenides (chlorides), alkyls, alkoxides, organometallics
    (cyclopentadienyl complexes), alkyl amides

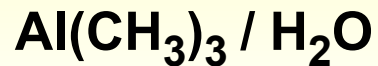**Nonmetallic** - water, hydrogen peroxide, ozone, hydrides, ammonia,
    hydrazine, amines

Thermally stable

Must react with surface centers (hydroxyl groups on oxide surface)

Thermodynamics – Kinetics – Mechanisms

Examples of ALD

High-permitivity Oxides

$Al(CH_3)_3$ / H_2O

$ZrCl_4$ / H_2O

$HfCl_4$ / H_2O
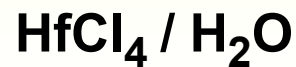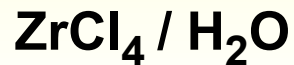
DRAM capacitors

$(Ba,Sr)TiO_3$ – Sr and Ba cyclopentadienyl compounds
and water as precursors

Nitrides of transition metals

TiN - $TiCl_4$ and NH_3

TaN - $TaCl_5$ / Zn / NH_3

WN - WF_6 and NH_3

WC_xN_y
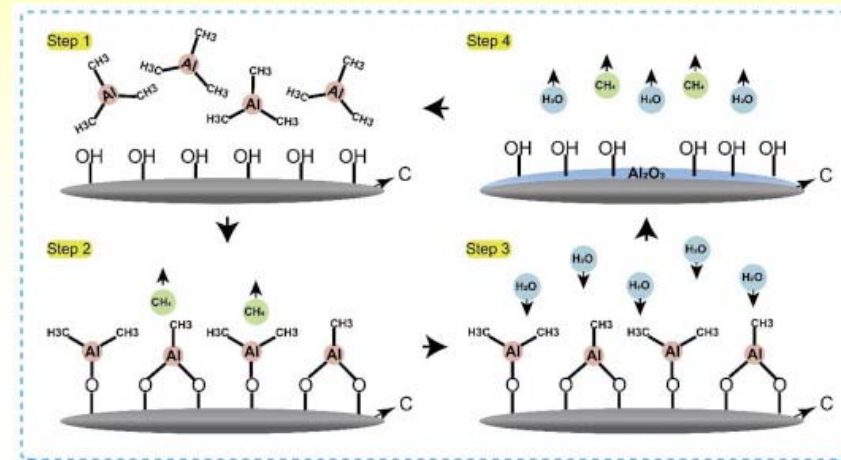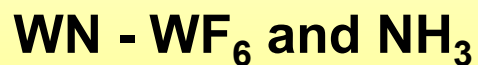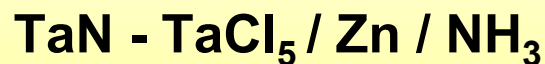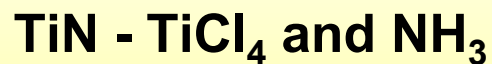
Examples of ALD

Metallic films

Difficult by ALD: metal surface has no reaction sites, low reactivity with reducing agents

W - WF_6 and Si_2H_6

Ru, Pt - organometallic precursors and oxygen applies to all precious metals capable of catalytic dissociation of O_2

Ni, Cu – metal oxide reduction by hydrogen radicals formed in plasma

Al – direct reduction of $AlMe_3$ by H radicals from plasma

ALD of SiO$_2$ and Al$_2$O$_3$ Films

Precursors:

- **Trimethylalane**
- **Tris(tert-butoxy)silanol**
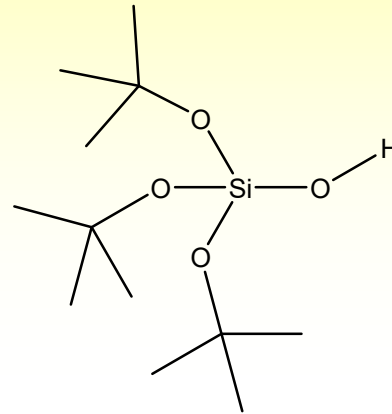
Deposition of amorphous SiO$_2$ and nanolaminates of Al$_2$O$_3$

32 monolayers in 1 cycle

Applications:

- **microelectronics**
- **optical filters**
- **protective layers (against diffusion, oxidation, corrosion)**
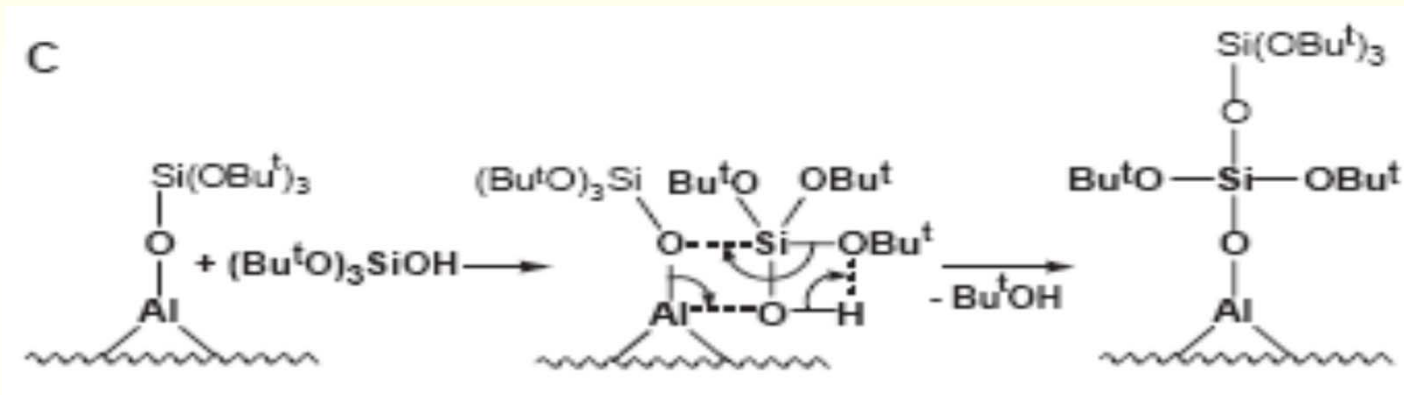
ALD of SiO$_2$ and Al$_2$O$_3$ Films

Step A
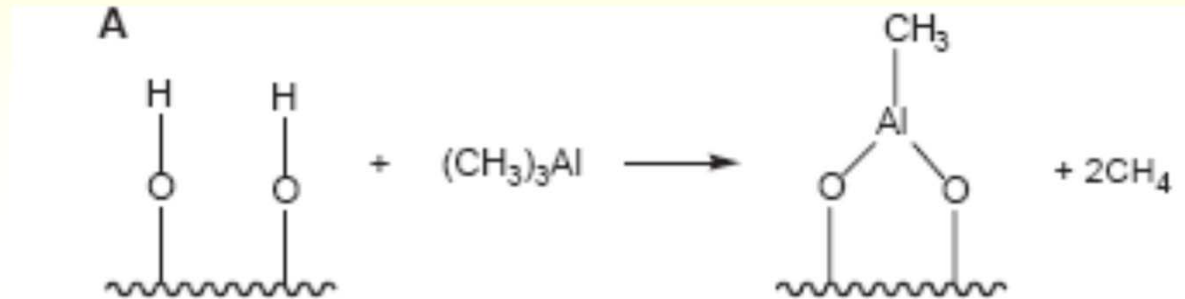
Step B
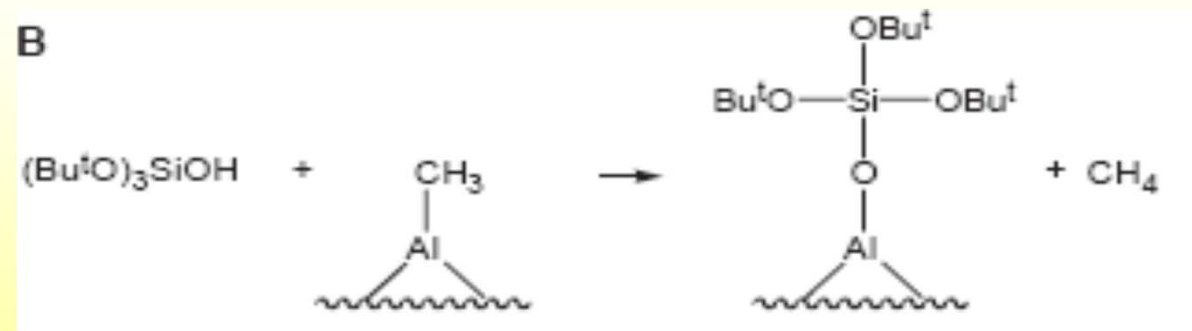
ALD of SiO$_2$ and Al$_2$O$_3$ Films

C, D: alkoxide - siloxide exchange
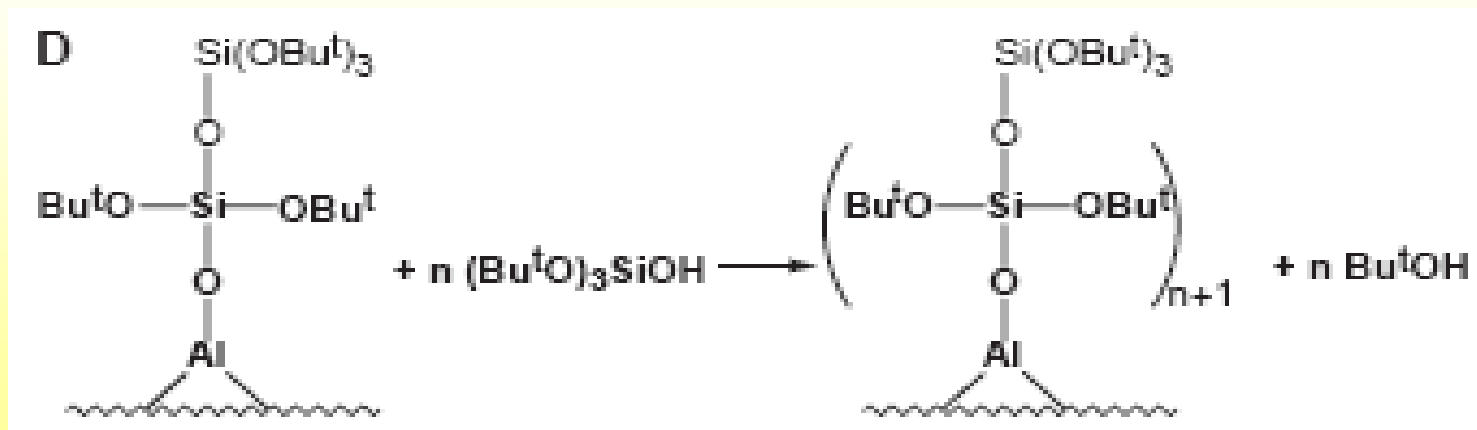
ALD of SiO$_2$ and Al$_2$O$_3$ Films

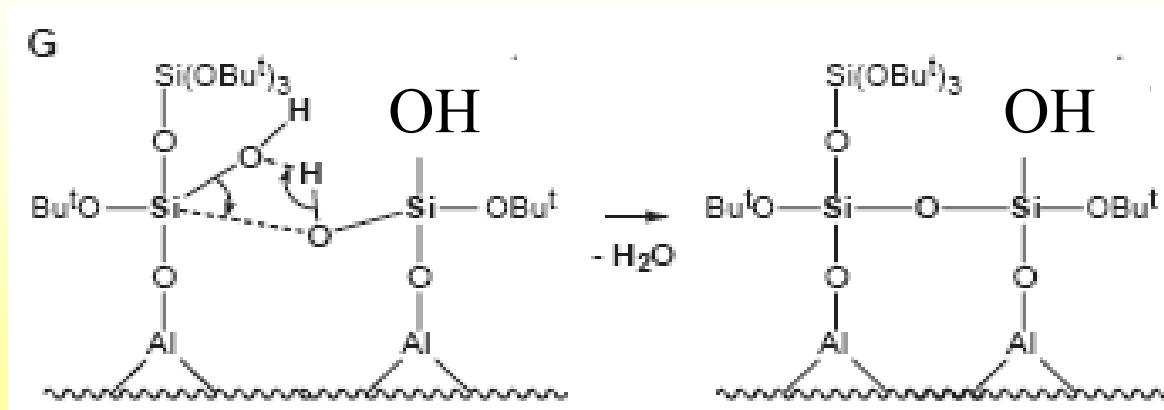E: elimination of isobutene = formation of -OH
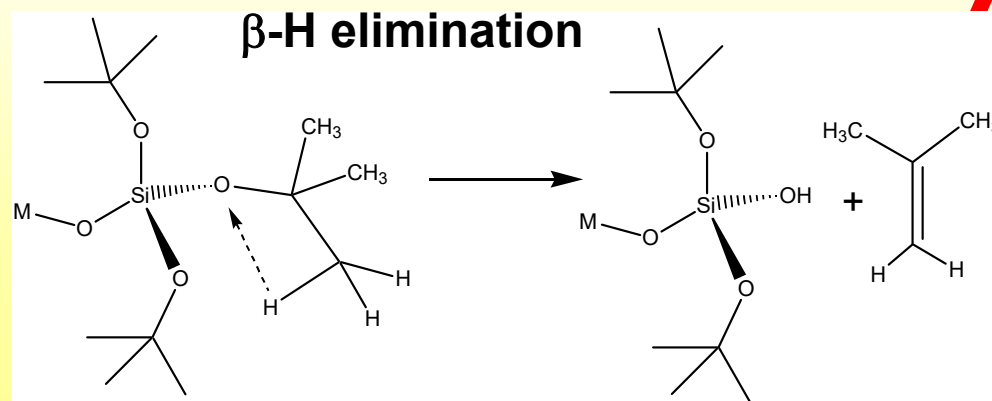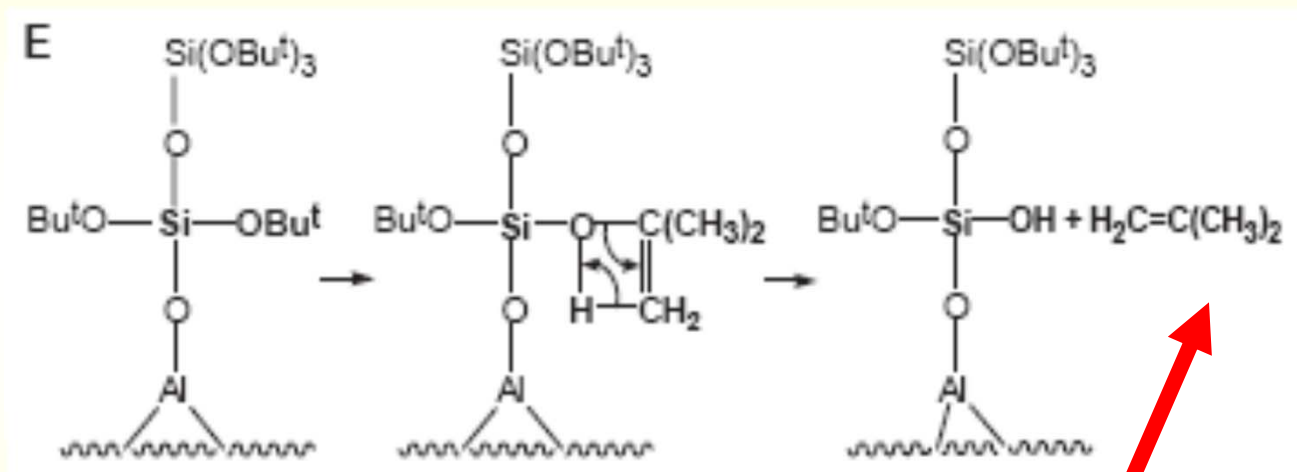
ALD of SiO$_2$ and Al$_2$O$_3$ Films
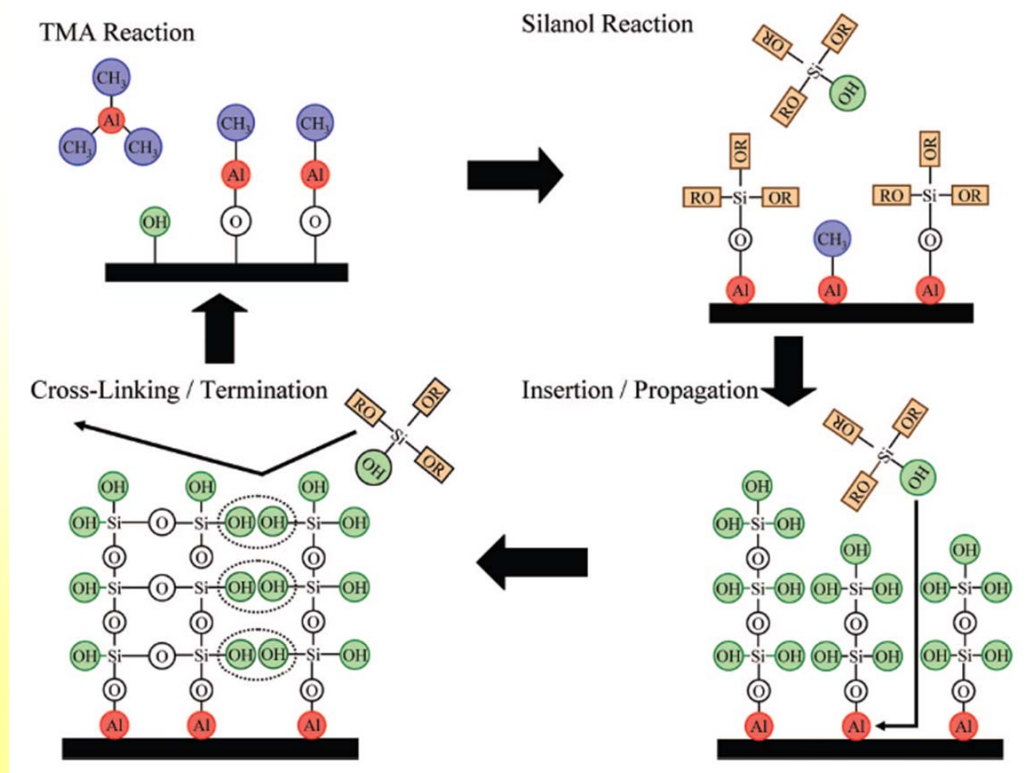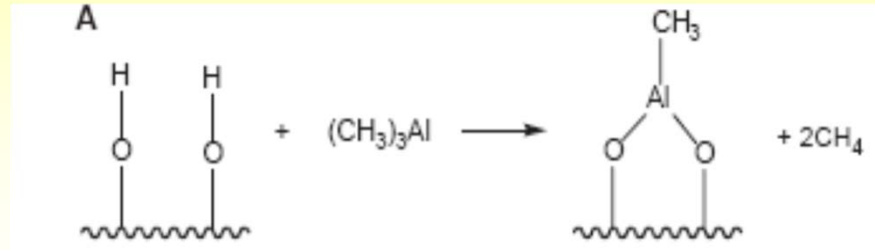
F: elimination of butanol = condensation
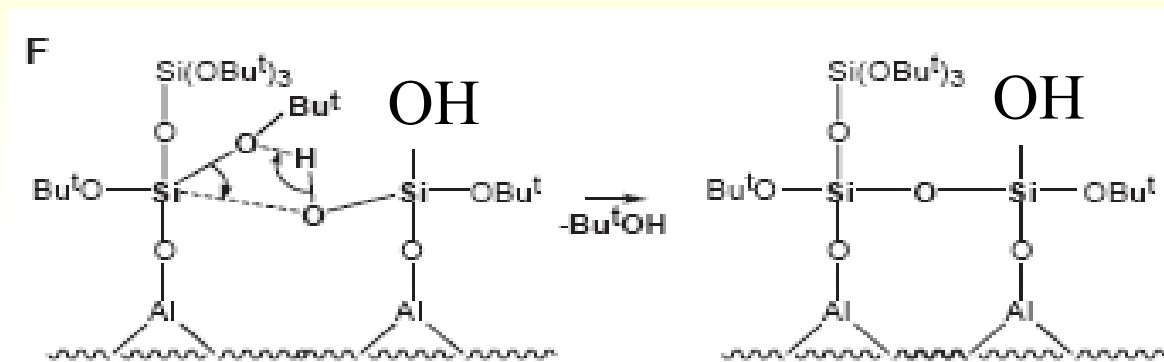
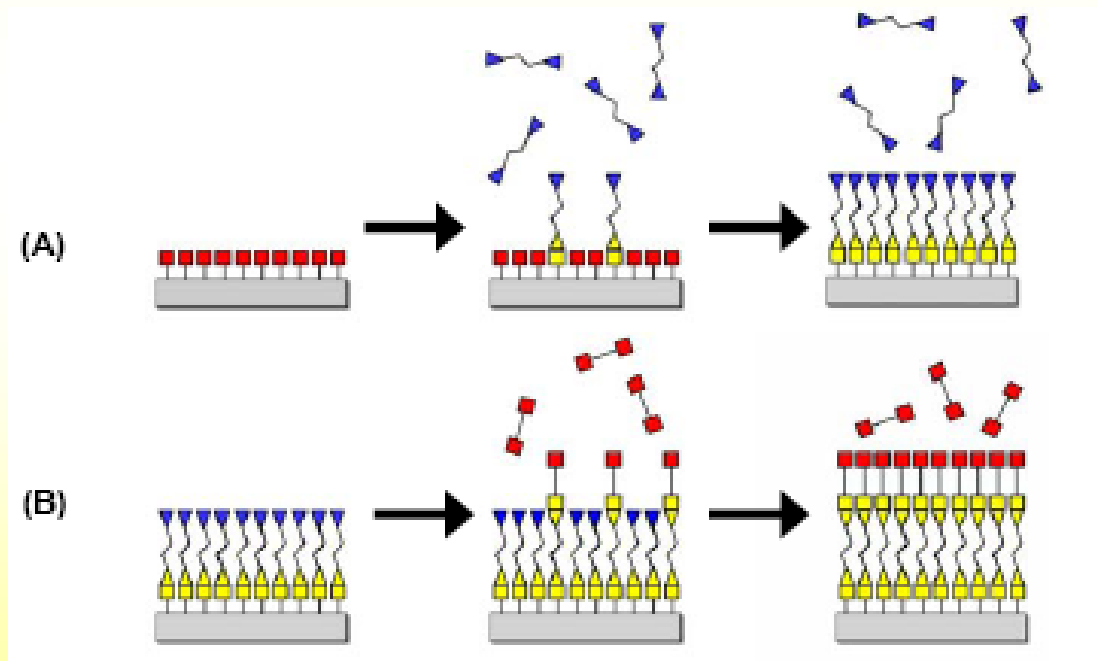
G: elimination of water = condensation

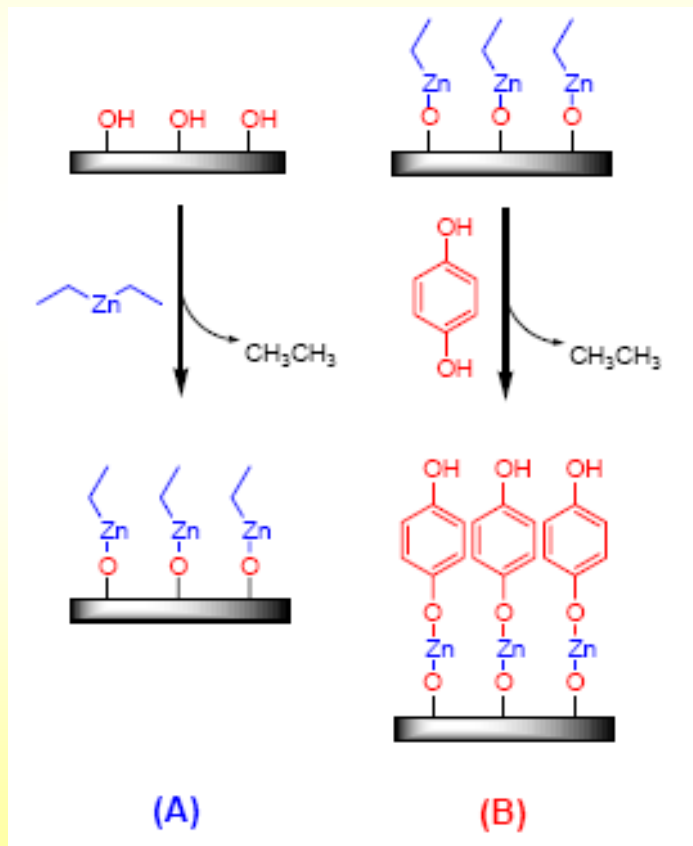# ALD of SiO$_2$ and Al$_2$O$_3$ Films

Repeat Step A

Molecular Layer Deposition MLD

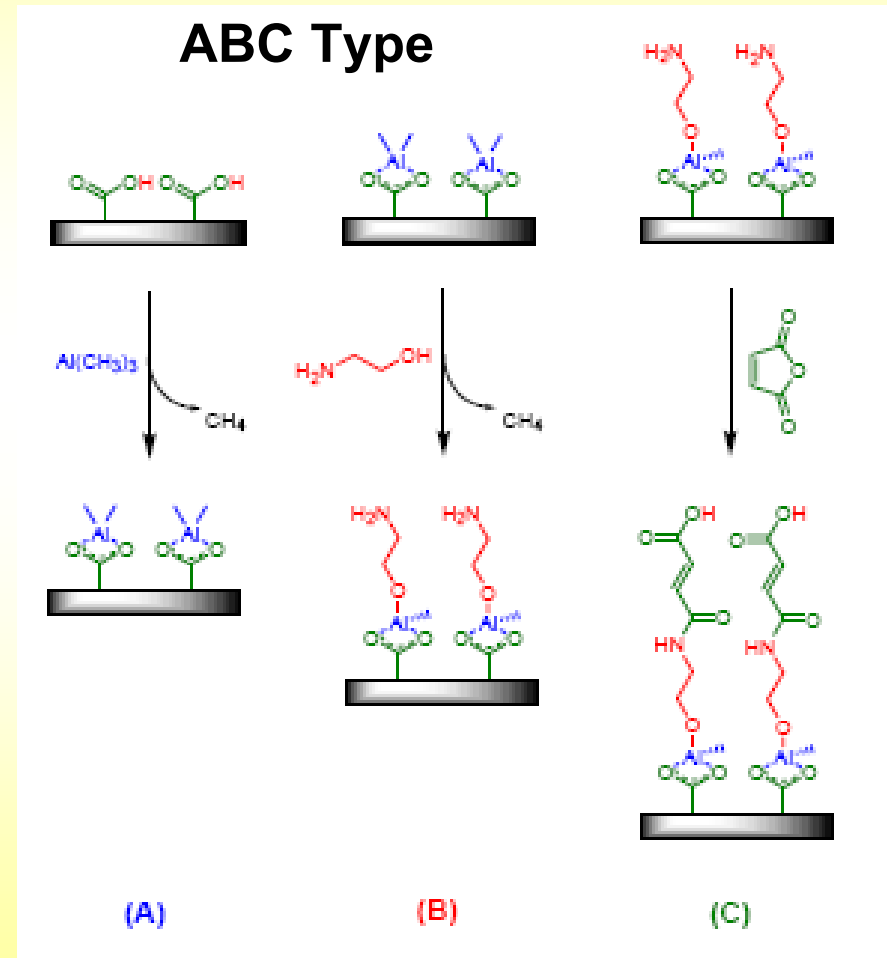**Sequential, self-limiting reactions A and B for MLD growth using two homobifunctional reactants**
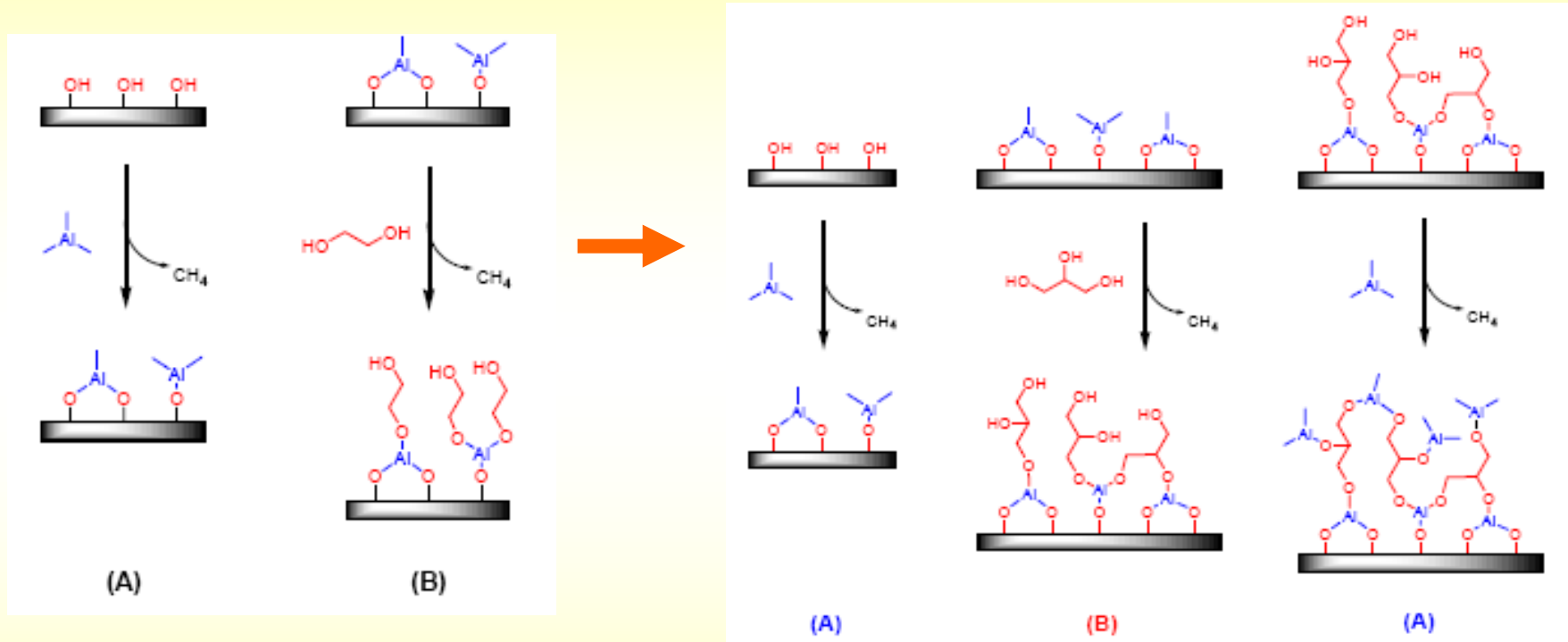
Molecular Layer Deposition MLD

AB Type

ABC Type

Diols vs. Polyols

(A)   (B)   →   (A)   (B)   (A)

Homobifunctional precursors can react twice with the AlCH$_3$* surface species, double reactions lead to a loss of reactive surface sites and decreasing growth rate

AB Lewis Acid-Lewis Base Reactions